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Chapter

Perovskite Ferroelectric

Paramjit Kour and Sudipta Kishore Pradhan

Abstract

The spectrums of properties exhibited by ferroelectric materials are dielectric, ferroelectric, piezoelectric and pyroelectric effect. This is the makes these materials to have a wide range of useful application. Infrared detectors are used pyroelectric effect of ferroelectric materials. It is used in nonvolatile memories due to have ferroelectric hysteresis. Its piezoelectric properties make them useful for actuator, radio frequency filter, sensor, and transducer. Ferroelectric capacitors are used, their good dielectric behavior. According to the necessity of the system they are available in different form such as single crystals, ceramics, thin film, and polymer, composite. The diversity of properties ferroelectric materials always attracted the attention of engineers and researchers. Size reduction of this material from micro to nanoscale established an enormous consideration to develop nanotechnology. Its vast use of different filed imposed the in detail research in adding to the development of processing and characterization method. This chapter will put some light on some fundamental principle of ferroelectricity, the list of perovskite materials and their application.

Keywords: PZT, BaTiO₃, CaTiO₃, PT, PLZT, Kb

1. Introduction

Like ferromagnetic materials, the functional properties of ferroelectric materials find wide range of applications, ranging from actuators and sensors to memory or optical devices. A ferroelctric class of materials cannot define in a single sentence. So before we define the ferroelectric materials, we should classify dielectric materials. Dielectric is belonging to a class of insulating materials that on the application of an electric field shows dielectric polarization [1]. Here the center of symmetry plays a significant role for their properties. Crystal structure with a center of symmetry have such an arrangement of atoms around a point or center that by the inversion, we can get the same arrangement of atoms in the crystal. Dielectric materials belong to a group of non Centro symmetric crystal structure. In 432 point group the entire non Centro symmetric point group shows piezoelectric properties [2]. The properties due to which voltage obtains form charge a separation in the face due to the mechanical stress and vice versa. Both direct and inverse piezoelectric effects have a wide range of application in electronic devices [3]. Barium titanate is an example of non-centrosymmetric piezoelectric material used in microphone and transducer [2–4]. In non centrosymmetric crystals there is an axis of symmetry, called polarity. These piezoelectric polar crystals are shown pyroelectricity. With changes in temperature there is a charge separation. The cells of polar structure have efficient dielectric polarization, so often called a spontaneous polarization. Either by stress or by a change in temperature the dipole moment of these polar

structures is change. The is a charge separation in the surface, results in the spontaneous polarization. So the polar dielectric spontaneous polarization direction and magnitude can be modified with the applied stress. Zinc oxide, which belongs to family of polar dielectric shows wurtzitecrystal structure [5]. In this structure, between hexagonally packed oxygen ions layer, Zn2+ions are at the tetrahedral site.

Between these layers dielectric, whose spontaneous polarization depends on the direction and magnitude of the applied stress. Adequate amount of stress can change the direction of spontaneous polarization. But the exclusion of the stress does not bring back the original magnitude and direction of the spontaneous polarization. These families of polar dielectric are called ferroelectric. **Figure 1** shows the way the Centro symmetric, acentric, polar and ferroelectric, dielectrics are related to each other. Many review articles shown the history of ferroelectric [6–12]. Many great scientist open the path of discovery of the ferroelectric.

Pyroelctricity was studied by Brewster. Piezoelectricity was discovered by J. P. Curie. Debye, Boltzmann, Pockelsetc helps in conceptualizing the polarization in the dielectric. It was E. Schoridgener, who coined the term ferroelectricity, but Joesph Valasek known for the discovery of ferroelectric. In 1920 the Rochelle salt (sodium potassium tartrate) shows spontaneous polarization which can be switch with the magnitude and direction of the applied field. This is the first manifestation of the ferroelectricity in a crystal. That is the trademark of ferroelectricity. Logically effectively the term ferroelectricity is defined as the switchable polarization between two or more stable state by the application of electric field. There exist some exceptions. Some semiconductor materials show ferroelectric properties. They do not posses electric polarization. In some ferroelectrics materials the spontaneous polarization cannot be switched with the electric field. This is either due to they are too conducting or reach the electrical break down first. This ferroelectric property first observed in Rochelle salt. But later it is observed in oxides, polymer, ceramics, and liquid crystal. When it is about ferroelectric property, the perovskite structure materials have a special importance. So in this chapter, we will discuss some perovskite ferroelectrics which are used in various electronics devices. According to the structure, there are five types of structure. (i) Organic polymer (ii) Charge ordered ferroelectrics (iii) Magnetic order ferroelectric (iv) Corner-sharing oxygen octahedral (v) Hydrogen bonded radical compound (v) Ceramic polymer composites [11]. Among these ferroelectric group a mostly used ferroelectrics are the corner sharing oxygen octahedral oxides.

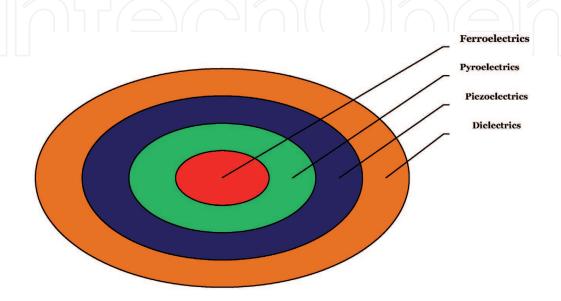


Figure 1.
Ven diagram of ferroelectric fit into different materials.

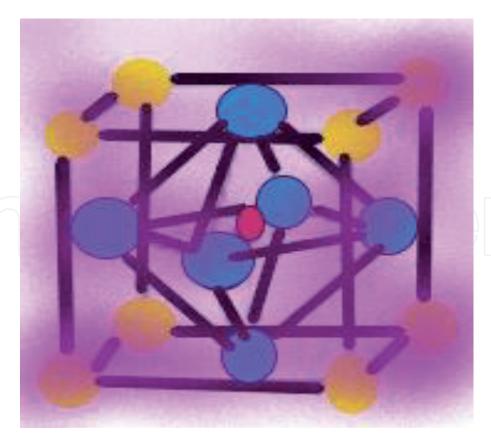


Figure 2.Shows the schematic diagram of perovskite structure.

Basically, the corner sharing oxygen octahedral oxide structures is represented as $A^{a+}B^{b+}O_3$ [12]. **Figure 2** shows the schematic figure of O^{2-} ions corner sharing oxygen octahedral. The B^{b+} cation is seated within every octahedron. The b of the cation has the value lies in between 3 to 6. A^{a+} ions lie in the gap among the octahedral with its value of a is in between 1 to 3. A nonpolar lattice has been observed in prototype forms by the overlap of A^{a+} , B^{b+} , and O^{2-} ions geometric center. The total polarity of the lattice is obtained by displacement of A and B ions with respect to the O^{2-} ions. Due to change in temperature, phase transition takes place. This will result in displacement of ion results in a change in lattice structure. Spontaneous polarization will produce due to the displacement of ions in the arrangement of dipoles if there is no recompense pattern in the diploe.

2. Perovskite ferroelectrics

It is a family of a subgroup of corner sharing oxygen octahedral material's exhibiting ABO₃ structure. This family contains some mostly used piezoelectric and ferroelectric ceramics. Some member which is used in different field of Technology are strontium titanate (STO), barium titanate (BaTiO₃), (SrTiO₃), lead titanate (PbTiO₃), barium strontium titanate (BST), PZT, potassium niobate (KN) (KNbO₃) etc. Some of these perovskite are discussed in detail as follows.

2.1 Barium titanate ceramics

Barium titanate which a member of perovskite family exhibit good piezoelectric, ferroelectric and high dielectric constant. This FE ceramics are used in first piezoelectric transducer. But now it is mostly used in multilayer capacitors (MLCs) due to having a high dielectric constant. It is also used in positive temperature coefficient

Resistance (PTCR), sensor, PTC Thermistors, IR detectors, RAM, sonars and in electro optic devices. By doping Barium titanate which is insulator in pure form can be changed into a semiconductor. Barium titanate basically white powder of inorganic compound. With a decrease in temperature the octahedral TiO_6 undergoes distortion gives rise to five structural phase transition from hexagonal, cubic, tetragonal, orthorhombic, and rhombohedra. Very large spontaneous polarization and high dielectric constant are observed in this phase due to the distorted octahedral. At Curie point, i.e. above 120° C the distorted octahedral of TiO_6 comes to equilibrium result in an isotropic cubic structure [13, 14] so only this phase does not exhibit FE properties.

Capacitor, MLC etc. used first BaTiO₃ as the dielectric ceramic with large dielectric constant and dielectric loss for manufacturing. The factor that affects the dielectric properties of the materials is not only its structure but also its synthesis route which will reflect in its size of the grain, density, purity etc. [15]. The Applied frequency of the electric field, temperature and do pants also affects the dielectric properties of the materials [16–23].

The dielectric constant of Barium titanate prepared by any method increase depends on the grain size and distribution of grains [17, 24]. At room temperature the frequency dependent dielectric constant decreases at low frequency and then increases slightly and become constant at high frequency. Temperature dependent dielectric constant decreases at higher temperature [25]. Ions of different size can be can be added to the perovskite structure. So doping at both A and B site of this perovskite structure is used to tolerate its electrical properties [26]. At A site mono, die and trivalent acceptor do pants are substituted to produce P-type semiconductor, whereas N type semiconductor are obtained by donor dopant of tri, tetra and pentavalent ions at the B site of the perovskite. The concentration of the dopants also affects the electrical properties of Barium titanate. Increase in concentration of Donor dopant makes the semiconducting ceramics to an insulating one. Barium titanate ceramics have application in various engineering fields.

2.1.1 Multilayer capacitor (MLCs)

Mostly $BaTiO_3$ is used in capacitor due to having a high dielectric constant. In thin dielectric form packed in a minimum space not only with high capacitance but also mechanically tough [27]. It is used as passive component in the circuit for large scale integration (LSI). It is also cost effective one with the use of internal electrode made of, nickel (Ni) and copper (Cu) [28].

2.1.2 (s2) PTC thermistor

The high resistivity of BaTiO₃ makes it as good candidates for PTC thermistor. Doping at A and B site of BaTiO₃ convert them as a semiconducting material. Below the Curie temperature these semiconductor materials have low resistance. Above crystal particle boarder barrier layer persuades by the surface state. So FE characteristics with high dielectric constant exhibit by this crystal boundary of high resistance. So below Curie temperature the potential barrier is small. Low resistivity has been exhibited in these materials due to effortlessly penetration of electron. The height of potential barrier increases above the Curie temperature makes electron difficult to pass through it results in increasing the resistivity of the material. In various electronics circuits this semiconducting BaTiO₃ materials are used as constant temperature heaters or switching devices. For temperature or parameter related to temperature can be detected, measured and control of temperature with

the use of this PTC thermistor. Among all the available sensor materials PTC has the highest temperature coefficient of resistance.

2.1.3 Nanogenerator (NG)

High piezoelectric effect with biocompatibility make them use as Nanogenerator. Piezoelectric potential is induced in between the two electrodes of NG, this is due to mechanical stress. Commercial devices can be work using this generated electrical energy without an external energy source.

2.2 Strontium titanate (SrTiO₃)

Another member of perovskite family is strontium ferrite STO (SrTiO₃) is a complex oxide. It exhibits cubic structure at room temperature. O²⁻ ions are bonded with six folded coordinate to Ti⁴⁺ ions and with twelve folded coordinate to Sr²⁺ ion. Each Sr²⁺ ion lies in between four TiO₆ octahedral. To decrease in temperature, it undergoes a phase transition. The first transition below room occur at -168°C. The opposite rotation of adjacent oxygen octahedral at this term turns the cubic structure to tetragonal structure. To further decrease in temperature at -236°C the changeable phonon modes turns the tetragonal structure to orthorhombic structure. The orthorhombic structure is change to rhombohedra structure at -263°C [29]. As Curie–Weiss law suggests the dielectric constant increases with the phase transition below room temperature. At this temperature due to quantum fluctuations leads to the quantum PE [30]. The charge storage capacity is high, chemical stability, optical transparency in the visible region with good insulating properties makes it use in modern electronics applications such as phase shifters, high-voltage capacitors, delay lines, filters, tunable oscillators etc. [31–33]. It is used in cancer treatment and in thermo – electric generators due to they have insolubility and high melting point properties [34, 35]. It shows photoconductivity when exposed to light due to having an direct gap and indirect band gap of 3.75 eV and 3.25 eV respectively [36]. Its conductivity increases with the contact of light to the crystal. Its conductivity persists for several days, with small decay [37, 38].

The first oxide, which is superconducting below 0.35 K is the strontium oxide (STO) [39]. It can be used as a tremendous substrate for superconductors with a high operating temperature and for oxide-based thin films. It is used as single-crystal substrate due to the enhanced electrical conductivity of niobium doped STO for the growth of perovskite oxides. It is also used for other ferroelectric and magnetic oxide substrate due to its large lattice parameter. Variation of temperature and pressure parameter can lead to increase oxygen vacancies in both crystal and thin films of SrTiO₃. So it becomes more conducting due to stimulate free electrons in the conduction band and also opaque. This used as gate-dielectric material due to the growth of high quality growth of epitaxial SrTiO₃ layers on silicon without forming silicon oxide Furthermore, it allow the incorporation of other perovskite oxides based thin-film on top of silicon [40]. It is also used for its piezoelectricity, ferroelectricity and Pyro electricity application in nanoscale [41–44]. It is used for various technological applications such as in super capacitor, nonvolatile memory, tunable microwave capacitors, ultralow-temperature scanning microscopies, high-density dynamic random access memories, soft phonon devices, oxygen sensors [45, 46].

2.3 Barium strontium titanate (BST)

The model of $BaTiO_3$ (BTO) has been used to develop Barium strontium titanate ($Ba_xSr_1_xTiO_3$ perovskite. This perovskite also undergoes phase change at it

Curie temperature depend upon the ratio of Ba:Sr. If the ratio decreases the curie temperature decreases. The dielectric constant at the Curie temperature of BST is more than BTO. BST is simply ferroelectric with spontaneous polarization below the Curie temperature. Nearly $T_{\rm c}$ the tenability of BST has extremely high in the FE phase. So in this phase find application for non volatile application. Above the Curie temperature the pyroelectric BST finds its applications in tunable microwave device associate with low dielectric loss and high dielectric constant. It also finds application in phase shifters, tunable filters and tunable antennas due to its composition dependent curie temperature with permittivity depends electric field.

Due to its high value of capacitance make it useful to construct high capacitance capacitor. It has uses in tunable microwave devices tunable capacitor, phase shifters, tunable transformers. BST varactors are a good replacement of the presently used semiconductor varactor and mechanical tuners. It is not only the drawback of large size with small tuning speed of mechanical tunners, but also small power handling capability of semiconductor varactor. BST also used in band pass and low pass tunable filter.

Semiconductor based phase shifter are used in fighter aircraft radar and cellular telephone base stations are associated with high loss at microwave frequencies with low power use ability. BST is the best replacement of these semiconductor based phase shifter associate with small loss, inexpensive and with better power handling properties. It is used in micro strip antenna. In tunable microwave application thin film of BST is used. The reduced size with small weight makes it compatible with microwave circuit. Dielectric constant not decreases sharply with the variation of the thickness of the film [47–49]. When it's used in metal-insulator-metal capacitor shows high dielectric constant, low dielectric loss, high leakage current density, high charge storage density makes it's used in dynamic random access memory (DRAM).

2.4 Lead titanate (PbTiO₃)

Another member of perovskite family is the inorganic lead salt of titanic acid compound, i.e. Lead titanate (PbTiO₃). Yellow powder of lead titanate is water insoluble. It shows a high Curie point of 490°C. It shows second order phase transition due to which it changes from cubic pyroelectric phase to ferroelectric tetragonal phase. At room temperature it shows the tetragonal structure belongs to P4 mm space group. It undergoes large volume change when cooled below the Curie temperature. It is not easy to formulate it in the bulk form. There is formation of crack during manufacture due to strain. To reduce this strain various dopants are used to modify the lead titanate. Ferroelctric lead titanate find its uses in, resonators, actuators, IR sensors, ultrasonic transducers and MLCs etc. [50-52]. Various process such as melting, Co precipitation, decomposition, hydrothermal, sol-gel, chemical vapor deposition, molecular beam epitaxy (MBE), molten salt methods, solid state method and sputtering is used to prepare thin films, single crystals and ceramic powders of PbTiO₃ [53–74]. For advance electromechanical devices, it is the capable building blocks. It can be prepared in micro tube, Nano sized powder and nanowire by the use of hydrothermal method [68, 75–79]. It is one of the most used materials for the fabrication of sensors, memory capacitors, optoelectronics devices etc. [76, 79].

2.5 Lead zirconate titanate (PZT)

Lead zirconate titanate with perovskite structures is an inorganic intermetallic compound. It is a solid solution of lead zirconate and lead titanate represented as [Pb (Zr_xTi_{1-x}) O_3 , $0 \le x \le 1$]. PZT has Pb²⁺ ions at A site with random occupation of B

site by Ti⁴⁺ and Zr⁴⁺ ions. It undergoes phase change with composition, but also with temperature. Above its Curie temperature it shows the pyroelectric effect with cubic structure. It undergoes a structural change from PE cubic phase either to FE rhombohedral phase or tetragonal phase. The spontaneous polarization in these structures is oriented in $\langle 100 \rangle$ and $\langle 111 \rangle$ set of directions for tetragonal and rhombohedral phase respectively. The morphotropic phase boundary is around at 52/48 of Zr/Ti ratio separating FE tetragonal and orthorhombic phases. At this boundary PZT has shown maximum dielectric and piezoelectric constants. Rhombohedral phase with 8 possible domains and tetragonal phase with 6 domains with total 14 domains are equally favorable at this composition. Piezoelectric PZT ceramics can be tailored according to application with ions having valence diverse from the host ions in the lattice. Especially PZT at MPB is modified to form soft and hard PZT. Acceptor ions are used either by the use of Al³⁺, Fe³⁺ at B site and Na⁺ and K⁺ at A site of the perovskite to produce oxygen vacancies in the lattice [80, 81]. Donor ions PZT produced domains wall motion fromed by Nd³⁺, La³⁺ at A site where as Nb⁵⁺, Sb⁵⁺ at B site in the lattice [82–85]. Hard PZT formed by donor doping shows low dielectric constant, small electrical losses, small piezoelectric co-efficient with high coercive field. So it is difficult to pole and dipole the sample and make them useful for rough applications. Whereas soft PZT has shown larger losses, high dielectric constant and piezoelectric co-efficient. So they can easily dope and dipole. They are used in FeRAM, actuator of STM/AFM type, ultrasound transducer, capacitor with high dielectric constant, IR sensor, etc. High value of piezoelectric coefficient of FE PZT make them useful for micro sensor, micro actuator used micro electromechanical system (MEMS) devices [86-88]. Nano rods, wire and hollow tube of PZT are now used for different application [89–98]. PZT Nanogenerator are used for piezoelectric effect to produce piezoelectric energy in the microscale. The intrinsic polar crystal structure of PZT nanofibers shows high piezoelectric voltage constant. High aspect ratio of nanostructure PZT overcomes extremely brittle nature observed in bulk PZT and its thin films and used for generation of energy in alternating loads [89, 90].

2.6 Lead magnesium niobate (PMN)

Single crystal of Pb ($Mg_{1/3}Nb_{2/3}$) O_3 (PMN) materials belongs to the FE relaxor materials and are used in [99], and the work in the electrostrictive actuators with high-strain and capacitor with high dielectric constant [100–102]. The Perovskite type structure of Pb (B_1 , B_2) O_3 formula with lower valence B_1 such as Zn^{2+} , Mg^{2+} , Fe^{3+} and Ni^{2+} and higher valency B_2 is such as Ta^{5+} , Nb^{5+} and W^{5+} . On cooling below the Curie temperature relaxor PMN ferroelectrics shows a wide dispersive and the diffused phase transition. Heterogeneity in composition on a microscopic scale leads to the diffuse phase transition in the relaxor ferroelectric.

In micoregion the stoichiometry is not obeyed by disorder B-site leads to change in FE transition temperatures leads to broadening dielectric peak. Strong frequency dependent dielectric constant has been observed in the relaxor ferroelectric. Curie temperature changes linearly with frequency with high dielectric loss below this temperature. Some relaxor shows second order phase transition. This relaxor remnant polarization is starting to decrease from Curie temperature and becomes zero with the increasing in temperature [103–106].

2.7 Potassium niobate (KNbO₃)

KN (KNbO₃) is also exhibited perovskite structure. It shows interesting ferroelectric properties at low temperature. It shows three phase transitions simple displacive type at low temperature with different symmetry and PE phase at

high-temperature [107]. Oxygen behaves as rigid body in the octahedral and vibrates about Nb atoms [108]. Liberation of oxygen octahedral leads to the irregular anisotropy exhibited by oxygen atom due to mean square displacements. Its structure shows two subshell obtained from the splitting of oxygen octahedron. Six niobium atoms are from the third nearest sub shell. 24 oxygen atoms form a fourth adjacent octahedral shell consists of four sub shell of six atoms. Fifth shell is made of 12 niobium atoms. Neutron diffraction study predicts the structural change with temperature KNbO₃. It shows three phase transition from cubic to tetragonal, tetragonal–orthorhombic and orthorhombic–rhombohedral at $T \approx 418$ °C, 225°C and 10°C respectively [109, 110], Transverse optic mode exhibited by KNbO₃ is softened with lessening temperature obtain from Raman, IR and inelastic neutron scattering [111–113]. Soften mode frequency obtain from dielectric measurement is good agreement with values calculated Cochran from $\omega^2 \propto (T - T_0)$ with $T_0 \approx 370$ °C Curie-Weiss temperature [114]. KNbO₃ Curie-Weiss constant was found to be about 2.8×10⁵ K. Displacive model is used to calculate this shows good agreement with the theoretical value. Large electromechanical coupling factor with zero temperature coefficients at room temperature exhibited by KN crystal is used for piezoelectric application [115, 116]. Surface acoustic wave (SAW) filter prepared using KNbO₃ find its application in mobile phones and television receivers [115, 116]. The crystal symmetry of KN crystal shows 49.5° rotation about the y axis by the x-cut [117, 118]. In high quality fiber shape these crystals show small lattice defects [119]. Different melting temperature hinders to grow high quality and large size KN crystal [115]. Both Bridgman (BM) technique and Top-seeded solution growth (TSSG) method is used, however the Bridgman (BM) technique is the easier one to grow bulk shape KN crystals [120–124]. Phase diagram suggests line compounds are formed when these crystals are developed from high-temperature solutions [125]. A peritectic transformation is shown when it grows from molten stoichiometric composition. KN in nanorod form are used capacitor and nano (NG) [126].

2.8 Sodium niobate (NaNbO₃)

This is also the member of perovskite family, but with anti ferroelctric properties. Six phase transition in between -200 to 650°C range affects its structural, dielectric, and optical properties. Phase transition at 200,360 and 480°C has been observed due to off center displacement of Nb ion with tilting of oxygen octahedral. Its stable cubic structure of the *Pm3m* space group has been observed at high temperature, i.e. >640°C [127, 128]. The orthorhombic structure of the phase *Pbcm* space group has been observed at room temperature is antiferroelctric one. At 360°C the antiferroelcetric orthombic structure with *Pbma* space group undergoes phase transition to antiferroelectric orthorhombic structure with a *Pnmm* space group associated with maximum dielectric constant [129–134]. NaNbO₃ single crystal exhibit low-frequency relaxation processes [135]. Distinct discontinuity is observed in mean relaxation time and relaxation parameter at Curie temperature Tc. At high temperature, low frequency relaxation increases due to crystalline structure disorder. This leads formation of local dipole in the polar region [136]. Stimulating electrical and mechanical properties of Sodium niobate-based ceramics make them a useful candidate for many technological applications [136–147]. At attainable electric fields this well-known antiferroelectric shows FE properties. Cost effective lead-free nanowire based NaNbO₃ piezoelectric has a high-output [148]. So it find application in hologram and optical data storage having high density [137–143]. It also used as nanocapacitors, NGs and in the memories of nanoscale [148–150]. For large-scale lead-free piezoelectric NG may be NaNbO₃ nanowires is one of useful candidate [151].

3. Conclusions

Several reports on ferroelectric materials and their use for different piezoelectric application has been studied in the last few years. As discussed in this chapter the effect of perovskite structure affect its ferroelectric properties. Doping in these perovskite structures also responsible for the enhanced of its properties by tailoring its crystal structure. These materials now investigated in the composite, nanowire and the nanorods form to make the device mechanically robust and more compact. This is a vital field for the research, as a key element in the digital world.

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References

- [1] von Hippel, A., Ferroelectricity, domain structure, and phase transitions of barium titanate, Rev. Mod. Phys., Vol. 22, 222-237 (1950).
- [2] Valasek, J. Piezo-electric and allied phenomena in Rochelle salt, Phys. Rev., 17, 475-481 (1921).
- [3] W. Kanzig, Ferroelectrics and Antiferroelectrics, Academic Press, New York (1957).
- [4] Hao, YM., Lou, SY., Zhou, SM. et al. Structural, optical, and magnetic studies of manganese-doped zinc oxide hierarchical microspheres by self-assembly of nanoparticles. Nanoscale Res Lett 7, 100 (2012).
- [5] Kanzig, W. History of ferroelectricity, 1938-1955. Ferroelectrics, 74, 285-291 (1987).
- [6] Cross, L.E. and Newnham, R.E. History of Ferroelectrics, Ceramics and Civilization, High-Technology Ceramics—Past, Present, and Future, vol. 3, American Ceramic Society, Westerville, OH, pp. 289-305 (1987).
- [7] Busch, G, How i discovered the ferroelectric properties of KH₂PO₄. Ferroelectrics, 71, 43-47 (1987).
- [8] Fousek, J., Ferroelectricity: remarks on historical aspects and present trends. Ferroelectrics, 113, 3-20 (1991).
- [9] Feldman, C., Formation of thin films of BaTiO₃ by evaporation. Review of Scientific Instruments, 26(5), 463-466 (1955).
- [10] Haertling, G.H, Ferroelectric ceramics: history and technology. J. Am. Ceram. Soc., 82 (4), 797-818 (1999).
- [11] Jin, Wenchao and Wang, Zhao and Huang, Hao and Hu, Xiaokang and He, Yahua and Li, High-performance

- piezoelectric energy harvesting of vertically aligned Pb(Zr,Ti)O₃ nanorod arrays, RSC Adv, 8, 7422-7427 (2018).
- [12] Seung, W., Yoon, H.J., Kim, T.Y., Ryu, H., Kim, J., Lee, J.H., Lee, J.H., Kim, S., Park, Y.K., Park, Y.J. and Kim, S.W., Boosting power-generating performance of triboelectric nanogenerators via artificial control of ferroelectric polarization and dielectric properties. Advanced Energy Materials, 7(2), 1600988 (2017).
- [13] Hench, L.L. and West, L.K. (1990) Principles of Electronic Ceramics, John Wiley and Sons, Inc., 244-247.
- [14] Kingery, W.D., Bowen, H.K., and Uhlmann, D.R. (1976) Introduction to Ceramics, John Wiley and Sons, Inc., pp. 926-927.
- [15] Guo, L., Luo, H., Gao, J., Guo, L., and Yang, J. Microwave hydrothermal synthesis of barium titanate powders. Mater. Lett., 60, 3011 (2006).
- [16] Arya, P.R., Jha, P., Subbanna, G.N., and Ganguli, A.K. Polymeric citrate precursor route to the synthesis of nano-sized barium lead titanates. Mater. Res. Bull., 38, 617 (2003).
- [17] Boulos, M., Guillement-Fritsch, S., Mathieu, F., Durand, B., Lebey, T., and Bley, V. Hydrothermal synthesis of nanosized BaTiO₃ powders and dielectric properties of corresponding ceramics. Solid State Ionics, 176, 1301 (2005).
- [18] Xu, H. and Gao, L. Tetragonal nanocrystalline barium titanate powder: preparation, characterization, and dielectric properties. J. Am. Ceram. Soc., 86, 203 (2003).
- [19] Vinothini, V., Singhand, P., and Balasubramanian, M. Synthesis of barium titanate nanopowder using

- polymeric precursor method. Ceram. Int., 32, 99 (2006).
- [20] Duran, P., Gutierrez, D., Tartaj, J., and Moure, C. Densification behaviour, microstructure development and dielectric properties of pure BaTiO₃ prepared by thermal decomposition of (Ba, Ti)-citrate polyester resins. Ceram. Int., 28, 283 (2002).
- [21] Stojanovic, B.D., Jovalekic, C., Vukotic, V., Simoes, A.Z., and Varela, J.A. Ferroelectric properties of mechanically synthesized nanosized barium titanate. Ferroelectrics, 319, 65 (2005).
- [22] Buscaglia, V., Viviani, M., Buscaglia, M.T., Nanni, P., Mitoseriu, L., Testino, A., Stytsenko, E., Daglish, M., Zhao, Z., and Nygren, M. Nanostructured barium titanate ceramics. Powder Technol., 148, 24 (2004).
- [23] Simon-Seveyrat, L., Hajjaji, A., Emziane, Y., Guiffard, B., and Guyomar, D. Re-investigation of synthesis of BaTiO₃ by conventional solid-state reaction and oxalate coprecipitation route for piezoelectric applications. Ceram. Int., 33, 35 (2007).
- [24] Kim, H.T. and Han, Y.H. Sintering of nanocrystalline BaTiO₃. Ceram. Int., 30, 1719 (2004).
- [25] Benlahrache, M.T., Barama, S.E., Benhamla, N., and Achour, A. Influence of polarization electric field on the dielectric properties of BaTiO₃-based ceramics. Mater. Sci. Semicond. Process., 9, 1115 (2006)
- [26] Buscaglia, M.T., Buscaglia, V., Viviani, M., Nanni, P., and Hanuskova, M. Influence of foreign ions on the crystal structure of BaTiO₃. J. Eur. Ceram. Soc., 20, 1997 (2000).
- [27] Wang, H.-L. Structure and dielectric properties of perovskite-barium titanate (BaTiO₃). MatE, 115 (2002).

- [28] Kuo, D.-H., Wang, C.-H., and Tsai, W.-P. Donor- and acceptor-cosubstituted BaTiO₃ for nonreducible multilayer ceramic capacitors. Ceram. Int., 32, 1-5 (2006).
- [29] Lytle, F.W. X-ray diffractometry of low-temperature phase transformations in strontium titanate. J. Appl. Phys., 35, 2212 (1964).
- [30] Muller, K.A. and Burkard, H. SrTiO₃: an intrinsic quantum paraelectric below 4 K. Phys. Rev. B, 19 (7), 3593-3602 (1979).
- [31] Kolodiazhnyi, T. and Petric, A. The applicability of Sr-deficient n-type SrTiO₃ for SOFC anodes. J. Electroceram., 15 (1), 5-11 (2005).
- [32] Rehn, L.E. Ion beams in hightemperature superconductivity research. Nucl. Instrum. Methods Phys. Res., Sect. B, 64, 161 (1992).
- [33] Ramirez, P. Colossal magnetoresistance. J. Phys.: Condens. Matter, 9, 8171-8199 (1997).
- [34] Power Sources for Remote Arctic Applications, U.S. Congress, Office of Technology Assessment (1994), Washington, DC, OTA-BP-ETI-129.
- [35] Standring, W.J.F., Selnæs, Q.G., Sneve, M., Finne, I.E., Hosseini, A., Amundsen, I., and Strand, P. (2005) Assessment of Environmental, Health and Safety Consequences of Decommissioning Radioisotope Thermal Generators (RTGs) in Northwest Russia.
- [36] Van Benthem, K., Elsässer, C., and French, R.H. Bulk electronic structure of SrTiO₃: experiment and theory. J. Appl. Phys., 90, 6156 (2001).
- [37] Marianne C. Tarun, Farida A. Selim, and Matthew D. McCluskey Persistent Photoconductivity in Strontium Titanate, Phys. Rev. Lett., 111, 187403 (2013).

- [38] Tamarra Kemsley, Light Exposure Increases Crystal's Electrical Conductivity 400-fold, Nature World News, Retrieved 18 November (2013).
- [39] Koonce, C.S. and Cohen, M.L. Superconducting transition temperatures of semiconducting SrTiO₃. Phys. Rev., 163 (2), 380 (1967).
- [40] McKee, R.A., Walker, F.J., and Chisholm, M.F., Crystalline oxides on silicon: the first five monolayers. Phys. Rev. Lett., 81 (14), 3014 (1998).
- [41] Urban, J.J., Yun, W.S., Gu, Q., and Park, H. Synthesis of single crystalline nanorods composed of Barium titanate and Strontium titanate. J. Am. Chem. Soc., 124 (7), 1186-1187 (2002).
- [42] Márquez-Herrera, A., Ovando-Medina, V.M., Corona-Rivera, M.A., Hernandez-Rodriguez, E., Zapata-Torres, M., Campos-Gonzalez, E., Guillen-Cervantes, A., Zelaya-Angel, O., and Meléndez-Lira, M. A novel solvothermal route for obtaining strontium titanate nanoparticles. J. Nanopart. Res., 15, 1525 (2013).
- [43] Kiat, J.M., Bogicevic, C., Gemeiner, P., Al-Zein, A., Karolak, F., Guiblin, N., Porcher, F., Hehlen, B., Yedra, L.L., Estradé, S., Peiró, F., and Haumont, R., Structural investigation of strontium titanate nanoparticles and the core-shell model. Phys. Rev. B, 87, 024106 (2013).
- [44] Goto, T., Nakashima, K., Fujii, I., Kuroiwa, Y., Makita, Y., Ryu, M., Suzuki, T., and Wada, S. Preparation of barium titanate/strontium titanate multilayer complex nanoparticles using nanocube substrates. IOP Conf. Ser. Mater. Sci. Eng., 18, 092030 (2011).
- [45] Voigts, F., Damjanovic, T., Borchardt, G., Argirusis, C., and Maus-Friedrichs, WSynthesis and characterization of strontium titanate nanoparticles as potential high temperature oxygen sensor material. J. Nanomater., 6, 63154 (2006).

- [46] AZO Nano Strontium Titanate (SrTiO₃) Nanoparticles-Properties and Applications (2016).
- [47] Shaw, T.M., Suo, Z., Huang, M., Liniger, E., Laibowitz, R.B., and Baniecki, J.D. The effect of stress on the dielectric properties of barium strontium titanate thin films. Appl. Phys. Lett., 75, 2129-2131 (1999).
- [48] Maria, J.-P., Parker, C.B., Kingon, A.I., and Stauf, G. Thickness, strain, and temperature-dependent properties of barium strontium titanate thin films. IEEE International Symposium on Applications of Ferroelectrics, pp. 151-154 (2002).
- [49] Basceri, C., Streiffer, S.K., Kingon, A.I., and Waser, R. The dielectric response as a function of temperature and film thickness of fiber-textured (Ba, Sr)TiO₃ thin films grown by chemical vapor deposition. J. Appl. Phys., 82 (5), 2497-2504 (1997).
- [50] Suchicital, C.T.A. and Payne, D.A. Flux growth of single crystal lead titanate. J. Cryst. Growth, 104, 211 (1990).
- [51] Jaffe, B., Cook, W.R. Jr., and Jaffe,H. (1971) Piezoelectric Ceramics,Academic Press, London, New York.
- [52] Meyer, R. Jr., Newnham, R., Alkoy, S., Ritter, T., and Cochran, J. Jr., Prefocused lead titanate >25 MHz single-element transducers from hollow spheres. IEEE Trans. Ultrason. Ferroelectr. Freq. Control, 48 (2), 488-493 (2001).
- [53] Kobayashi, J. Growing of ferroelectric PbTiO₃ crystals. J. Appl. Phys., 29, 866 (1958).
- [54] Sun, B.N., Huang, Y., and Payne, D.A. Growth of large PbTiO₃ crystals by a self-flux technique. J. Cryst. Growth, 128, 867 (1993).

- [55] Ganesa Moorthy, S., Joseph Kumar, F., Balakumar, S., and Subramanian, C. Growth of lead titanate single crystals and their characterization. J. Korean Phys. Soc., 32, S1214–S1216 (1998).
- [56] Grabmaier, B.C. PbTiO₃ grown from the melt. Ferroelectrics, 13, 501 (1976).
- [57] Oka, K., Unoki, H., Yamaguchi, H., and Takahashi, H. Crystal growth of PbTiO₃ by the top-seeded solution-growth method. J. Cryst. Growth, 166, 380 (1996).
- [58] Blum, J.B. and Gurkovich, S.R. Sol-gel-derived PbTiO₃. J. Mater. Sci., 20, 4479 (1985).
- [59] Tangboriboon, N., Jamieson, A.M., Sirivat, A., and Wongkasemjit, S. A novel route to perovskite lead titanate from lead and titanium glycolates via the sol–gel process. Appl. Organomet. Chem., 20 (12), 886-894 (2006).
- [60] Lanki, M., Nourmohammadi, A., and Feiz, M.H. (2012) Lead partitioning in sol–gel derived lead titanate nanopowders. Applications of Ferroelectrics Held Jointly with 2012 European Conference on the Applications of Polar Dielectrics and 2012 International Symposium Piezoresponse Force Microscopy and Nanoscale Phenomena in Polar Materials (ISAF/ECAPD/PFM), 2012 International Symposium, 2012, pp. 1-4.
- [61] Fox, G.R., Adair, J.H., and Newnham, R.E. Effects of pH and H₂O₂ upon co precipitated PbTiO₃ powders. J. Mater. Sci., 25, 3634 (1990).
- [62] Fang, J., Wang, J., Ng, S.C., Chew, C.H., and Gan, L.M., Preparation and characterisation of ultrafine lead titanate (PbTiO₃) powders. J. Mater. Sci., 34, 1943-1952 (1999).
- [63] Fang, J., Wang, J., Gan, L.-M., and Ng, S.-C., Comparative study on phase development of lead titanate powders. Mater. Lett., 52, 304-312 (2002).

- [64] Kim, M.J. and Matijevic, E. Preparation and characterization of 'uniform submicrometer lead titanate particles. Chem. Mater., 1, 363 (1989).
- [65] Watson, D.J., Randall, C.A., Newnham, R.E., and Adair, J.H. (1988) in Ceramics Transactions, Ceramic Powder Science II, vol. 1 (eds G.L. Messing, E.R. Fuller Jr., and H. Hausner), The American Ceramic Society Inc., Westerville, OH, p. 154.
- [66] Blakea, G.R., Armstronga, A.R., Sastreb, E., Zhoua, W., and Wrighta, P.A. The preparation of a novel layered lead titanate and its conversion to the perovskite lead titanate PbTiO₃. Mater. Res. Bull., 36, 1837-1845 (2001).
- [67] Morita, T. and Cho, Y. A hydrothermally deposited epitaxial lead titanate thin film on strontium ruthenium oxide bottom electrode. Appl. Phys. Lett., 85, 2331 (2004).
- [68] Morita, T. and Cho, Y. Piezoelectric property of an epitaxial lead titanate thin film deposited by the hydrothermal method. Appl. Phys. Lett., 88, 112908 (2006).
- [69] Yoon, S.-.G., Park, J.-.D., Choi, J.-.H., and Kim, H.-.G. Preparation, properties, and characterization of thin ferroelectric films of lead titanate. J. Vac. Sci. Technol., A, 9, 281 (1991).
- [70] Yoon, S.G. and Kim, H.G. Characterization and electrical properties of chemical vapour deposited ferroelectric lead titanate films on titanium. IEEE Trans. Ultra son. Ferroelectric. Freq. Control, 37 (5), 333-338 (1990).
- [71] Rispens, G. and Noheda, B. (2007) Ultra-thin lead titanate films grown by molecular beam epitaxy. Integr. Ferroelectric., 92, 30-39 (2006).
- [72] Jacobsen, H., Jung, T., Ortner, K., Schiffmann, K.I., Quenzer, H.V., and

- Wagner, B. Development of a piezoelectric lead titanate thin film process on silicon substrates by high rate gas flow sputtering. Sens. Actuators, A, 133 (1), 250-258 (2007).
- [73] Arendt, R.H. and Rosolowski, J.H.(1979) Molten salt synthesis of leadzirconate titanate solid solution powder.US Patent 4 152 282, May 1, 1979.
- [74] Shrout, T.R., Papet, P., Kim, S., and Lee, G.-S. Conventionally prepared submicrometer lead-based perovskite powders by reactive calcination. J. Am. Ceram. Soc., 73 (7), 1862 (1990).
- [75] Zhao, L., Steinhart, M., Yu, J., and Gösele, U. Lead titanate nano- and microtubes. J. Mater. Res., 21 (3), 685-690 (2006).
- [76] Hu, Y., Gu, H., Sun, X., You, J., and Wang, JPhotoluminescence and Raman scattering studies on PbTiO₃ nanowires fabricated by hydrothermal method at low temperature. Appl. Phys. Lett., 88 (19), 193120 (2006).
- [77] Shimada, T., Tomoda, S., and Kitamura, T. Ab initio study of ferroelectricity in edged PbTiO₃ nanowires under axial tension. Phys. Rev. B, 79, 024102 (2009).
- [78] Leong, M., Bayerl, D.J., Shi, J., and Wang, X. Evolution of lead titanate nanostructures from nanoparticle self-assembly. Sci. Adv. Mater., 4, 832-836 (2012).
- [79] Nwe, H.H., Maung, Y.M., Win, T.T., and Kyaw Soe, K.K. Hydrothermal synthesis of nano-sized PbTiO₃ powder and epitaxial film for memory capacitor application. Am. J. Mater. Sci. Technol., 1, 22-27 (2012).
- [80] Weston, T.B., Webster, A.H., and McNamara, V.M., Lead zirconate-lead titanate piezoelectric ceramics with iron oxide additions. J. Am. Ceram. Soc., 52, 253 (1969).

- [81] Uchida, N. and Ikeda, T. Studies on $Pb(Zr-Ti)O_3$ ceramics with addition of Cr_2O_3 . Jpn. J. Appl. Phys., 6, 1292 (1967).
- [82] Kulcsar, F. Electromechanical properties of lead titanate zirconate ceramics modified with certain three-or five-valent additions. J. Am. Ceram. Soc., 42, 343 (1959).
- [83] Kulcsar, F. Electromechanical properties of lead titanate zirconate ceramics modified with tungsten and thorium. J. Am. Ceram. Soc., 48, 54 (1965).
- [84] Atkin, R.B., Holman, R.L., and Fularth, R.M. Substitution of Bi and Nb ions in lead zirconate-titanate. J. Am. Ceram. Soc., 54, 113 (1971).
- [85] Banno, H. and Tsunooka, T. Piezoelectric properties and temperature dependences of resonant frequency of WO₃-MnO₂-modified ceramics of Pb(Zr-Ti)O₃. Jpn. J. Appl. Phys., 6, 954 (1967).
- [86] Polla, D.L. and Francis, L.F. Ferroelectric thin films in microelectromechanical systems applications. MRS Bull., 21, 59-65 (1996).
- [87] Zhang, Q.Q., Gross, S.J., Tadigadapa, S., Jackson, T.N., Djuth, F.T., and Trolier-McKinstry, S. Lead zirconate titanate films for d33 mode cantilever actuators. Sens. Actuators, A, 105, 91-97 (2003).
- [88] Kim, J.H., Wang, L., Zurn, S.M., Li, L., Yoon, Y.S., and Polla, D.L. Fabrication process of PZT piezoelectric cantilever unimorphs using surface micromachining. Integr. Ferroelectric. 15, 325-332 (1997).
- [89] Swallow, L.M., Luo, J.K., Siores, E., Patel, I., and Dodds, D. A piezoelectric fibre composite based energy harvesting device for potential wearable applications. Smart Mater. Struct., 17 (2), 025017 (2008).

- [90] Chen, X., Xu, S., Yao, N., and Shi, Y., 1.6 V nanogenerator for mechanical energy harvesting using PZT nanofibres. Nano Lett., 10 (6), 2133-2137 (2010).
- [91] Xu, S., Hansen, B.J., and Wang, Z.L., Piezoelectric nanowire-enabled power source for driving wireless microelectronics. Nat. Commun., 93, 1-5 (2010).
- [92] Wu, W., Bai, S., Yuan, M., Qin, Y., Wang, Z.L., and Jing, T., Lead zirconate titanate nanowire textile nanogenerator for wearable energy-harvesting and self-powered devices. ACS Nano, 6 (7), 6231-6235 (2012).
- [93] Gu, L., Cui, N., Cheng, L., Xu, Q., Bai, S., Yuan, M., Wu, W., Liu, J., Zhao, Y., Ma, F., Qin, Y., and Wang, Z.L. Flexible fiber nanogenerator with 209 V output voltage directly powers a lightemitting diode. Nano Lett., 13 (1), 91-94 (2013).
- [94] Zhu, G., Wang, A.C., Liu, Y., Zhou, Y., and Wang, Z.L. Functional electrical stimulation by nanogenerator with 58 V output voltage. Nano Lett., 12, 3086-3090 (2012).
- [95] Lee, K.Y., Kumar, B., Seo, J.S., Kim, K.H., Sohn, J.I., Cha, S.N., Choi, D., Wang, Z.L., and Kim, S.W. P-type polymer-hybridized high-performance piezoelectric nanogenerators. Nano Lett., 12, 1959-1964 (2012).
- [96] Park, K.I., Jeong, C.K., Ryu, J., Hwang, G.T., and Lee, K.J. Flexible and large-area nanocomposite generator based on lead zirconate titanate particles and carbon nanotubes. Adv. Energy Mater., 3, 1539-1544 (2013).
- [97] Scott, J.F., Fan, H.J., Kawasaki, S. et al. Terahertz emission from tubular Pb(Zr, Ti)O₃ nanostructures. Nano Lett., 8 (12), 4404-4409 (2008).
- [98] Bai, S., Xu, Q., Gu, L., Ma, F., Qin, Y., and Wang, Z.L. (2012) Single

- crystalline lead zirconate titanate (PZT) nano/micro-wire based self-powered UV sensor. Nano Energy, 1 (6), 789-795.
- [99] Smolenskii, G.A. and Agranovskaya, A.I. Dielectric polarization of a number of complex compounds. Sov. Phys. Solid State (Eng. Transl.), 1 (10), 1429-1437 (1960).
- [100] Cross, L.E., Jang, S.J., and Newnham, R.E. Large electrostrictive effects in relaxor ferroelectrics. Ferroelectrics, 23, 187-192 (1980).
- [101] Nomura, S. and Uchino, K. Recent applications of PMN-based electrictors. Ferroelectrics, 50, 197-202 (1983).
- [102] Kahn, M., Burks, D., Burn, I., and Schulze, W. in Electronic Ceramics (ed. L.M. Levinson), Marcel Dekker, New York, pp. 191-274 (1988).
- [103] Cross, L.E. Relaxor ferroelectrics. Ferroelectrics, 76, 241 (1987).
- [104] Smolenski, G.A. and Agranovskaya, A.I. Dielectric polarization and losses of some complex compounds. Sov. Phys. Tech. Phys., 3, 1380 (1958).
- [105] Shrout, T.R. and Halliyal, A. Preparation of lead-based ferroelectric relaxors for capacitors. Am. Ceram. Soc. Bull., 66, 704 (1987).
- [106] Shrout, T.R. and Dougherty, J.P. Lead Based Pb(B₁B₂)O₃ Relaxors vs BaTiO₃ Dielectrics for Multilayer Capacitors, Ceramic Transactions, Ceramic Dielectrics: Composition, Processing, and Properties, J. Am. Ceram. Soc., 8, 3 (1990).
- [107] Matthias, B.T., New ferroelectric crystals. Phys. Rev. B, 75, 1771 (1949).
- [108] Hewat, A.W. Cubic-tetragonalorthorhombic-rhombohedral ferroelectric transitions in perovskite potassium niobate: neutron powder

- profile refinement of the structures. J. Phys. C: Solid State Phys., 6, 2559-2572 (1973).
- [109] Frenkel, A.I., Stern, E.A., and Yacoby, Y., Pressure-induced changes in the local structure of KNbO₃. AIP Conference Proceedings, 436, 238 (1998).
- [110] Perry, C.H., Hayes, R.R., and Tornberg, N.E. (1971) in Proceedings of the 2nd International Conference on Light Scattering of Solids (ed. M. Balkanski), Paris: Flammarion Sciences.
- [111] Fontana, M.D., Metrat, G., Servoin, J.L., and Gervais, F. Soft ferroelectric mode in KNbO₃. Ferroelectrics, 38, 797 (1981).
- [112] Fontana, M.D., Kugel, G.E., Vamvakas, J., and Carabatos, C. Persistence of tetragonal raman lines in cubic KNbO₃. Solid State Commun., 45, 873 (1983).
- [113] Fontana, M.D., Dolling, G., Kugel, G.E., and Carabatos, C. Inelastic neutron scattering in tetragonal KNbO₃. Phys. Rev. B, 20, 3850 (1979).
- [114] Jona, F. and Shirane, G. (1962) Ferroelectric Crystals, MacMillan, New York.
- [115] Yamanouchi, K., Wagatsuma, Y., Odagawa, H., and Cho, Y. Single crystal growth of KNbO₃ and application to surface acoustic wave devices. J. Eur. Ceram. Soc., 21, 2791-2795 (2001).
- [116] Yamanouchi, K. and Odagawa, H. Research of super-high electromechanical coupling surface acoustic wave substrates. Jpn. J. Appl. Phys., 40, 3726-3728 (2001).
- [117] Nakamura, K. and Kawamura, Y. Orientation dependence of electromechanical coupling factors in KNbO₃. IEEE Trans. Ultrason. Ferroelectr. Freq. Control, 47, 750-755 (2000).

- [118] Kimura, H., Tanahashi, R., Maiwa, K., Baba, H., Cheng, Z.X., and Wang, X.L. Potassium-sodium-rubidium niobate single crystals and electric properties. Int. J. Mod. Phys. B, 23, 3631-3636 (2009).
- [119] Takagi, T., Fujii, T., and Sakabe, Y. Growth and characterization of KNbO₃ by vertical Bridgman method. J. Cryst. Growth, 259, 296-301 (2003).
- [120] Fukuda, T. and Uematsu, Y. Preparation of KNbO₃ single crystal for optical applications. Jpn. J. Appl. Phys., 11, 163-169 (1972).
- [121] Fukuda, T., Uematsu, T., and Ito, T. Kyropoulos growth and perfection of KNbO₃ single crystal. J. Cryst. Growth, 24-25, 450-453 (1974).
- [122] Xing, W., Looser, H., Wuest, H., and Arend, HProgress in KNbO₃ crystal growth. J. Cryst. Growth, 78, 431-437 (1986).
- [123] Zgonik, M., Schlesser, R., Biaggio, I., Voit, E., Tscherry, J., and Günter, P. Materials constants of KNbO₃ relevant for electro and acousto-optics. J. Appl. Phys., 74, 1287-1297 (1993).
- [124] Wiesendanger, E. Dielectric, mechanical and optical properties of orthorhombic KNbO₃. Ferroelectrics, 6, 263-281 (1974).
- [125] Reisman, R. and Holtzberg, F. Phase equilibria in the system K_2CO_3 -Nb₂O₅ by the method of differential thermal analysis. J. Am. Chem. Soc., 77, 2115-2119 (1955).
- [126] Jung, J.H., Chen, C.-Y., Yun, B.K., Lee, N., Zhou, Y., Jo, W., Chou, L.-J., and Wang, Z.L. Lead-free KNbO₃ ferroelectric nanorod based flexible nanogenerators and capacitors. Nanotechnology, 23, 375401 (2012).
- [127] Mishra, S.K., Choudhury, N., Chaplot, S.L., Krishna, P.S.R., and

- Mittal, R. Competing antiferroelectric and ferroelectric interactions in NaNbO₃: neutron diffraction and theoretical studies. Phys. Rev. B, 76, 024110 (2007).
- [128] Mishra, S.K., Mittal, R., Pomjakushin, V.Y., and Chaplot, S.L. Phase stability and structural temperature dependence in sodium niobate: a high-resolution powder neutron diffraction study. Phys. Rev. B, 83, 134105 (2011).
- [129] Barth, T. (1925) Die Kristallstruktur von Perowskit und verwandten Verbindungen. Nor. Geol. Tidsskr., 8, 201.
- [130] Cross, L.E. and Nicholson, B.J. The optical and electrical properties of single crystals of sodium niobate. Philos. Mag., 46, 453-466 (1955).
- [131] Ahtee, M., Glazer, A.M., and Megaw, H.D. The structures of sodium niobate between 480 and 575°C and their relevance of soft-phonon modes. Philos. Mag., 26 (4), 995-1014 (1972).
- [132] Lefkowitz, I., Lukaszewicz, K., and Megaw, H.D. The high temperature phase of sodium niobate and the nature of transitions in pseudosymmetric structure. Acta Crystallogr., 20, 670-683 (1966).
- [133] Ishida, K. and Honjo, G. Soft mode and super lattice structure in NaNbO₃. J. Phys. Soc. Jpn., 34 (5), 1279-1288 (1973).
- [134] Glazer, A.M. and Megaw, H.D. Studies of the parameters and domains in the phase transition of NaNbO₃. Acta Crystallogr., A29, 489-495. (1973)
- [135] Konieczny, K. and Kajtoch, C. Low-frequency dielectric dispersion in NaNbO₃ single crystals. Ferroelectrics, 215, 65-73 (1998).
- [136] Konieczny, K. Dielectric relaxation in NaNbO₃ single crystal. Condens. Matter Phys., 2 (4), 655-660 (1999).

- [137] Xu, Y. (1991) Ferroelectric Materials and Their Applications, North Holland, Amsterdam.
- [138] Lines, M.E. and Glass, A.M. (1977) Principles and Applications of Ferroelectrics and Related Materials, Clarendon Press, Oxford.
- [139] Tejuca, L.G. and Fierro, J.L.G. (1993) Properties and Applications of Perovskite-Type Oxides, Marcel Dekker, New York. 176
- [140] Valdez, E., de Araujo, C.B., and Lipovskii, A.A. Second harmonic scattered light from a transparent glass-ceramic containing sodium niobate nanocrystals. Appl. Phys. Lett., 89, 031901 (2006).
- [141] Hollenstein, E., Davis, M., Damjanovic, D., and Setter, N. Piezoelectric properties of Li- and Ta-modified (K_{0.5}Na_{0.5}) NbO₃ ceramics. Appl. Phys. Lett., 87, 182905 (2006).
- [142] Falcão-Filho, E.L., Bosco, C.A.C., Maciel, G.S., Acioli, L.H., de Araújo, C.B., Lipovskii, A.A., and Tagantsev, D.K. Third-order optical nonlinearity of a transparent glass ceramic containing sodium niobate nanocrystals. Phys. Rev. B, 69, 134204 (2004).
- [143] Maciel, G.S., Rakov, N., de Araujo, C.B., Lipovskii, A.A., and Tagantsev, D.K. Optical limiting behavior of a glass–ceramic containing sodium niobate crystallites. Appl. Phys. Lett., 79 (5), 584 (2001).
- [144] Saito, Y., Takao, H., Tani, T., Nonoyama, T., Takatori, K., Homma, T., Nagaya, T., and Nakamura, M. Leadfree piezoceramics. Nature, 432, 84 (2004).
- [145] Cross, E. Materials science: lead-free at last. Nature, 432, 24 (2004).
- [146] Yuzyuk, Y.I., Simon, P., Gagarina, E., Hennet, L., Thiaudiere, D.,

Torgashev, V.I., Raevskya, S.I., Raevskii, I.P., Reznitchenko, L.A., and Sauvajol, J.L. Modulated phases in NaNbO₃: Raman scattering, synchrotron x-ray diffraction, and dielectric investigations. J. Phys. Condens. Matter, 17, 4977 (2005).

[147] Yuyuk, Y.I., Gagarina, E., Simon, P., Reznitchenko, L.A., Hennet, L., and Thiaudiere, D. Synchrotron x-ray diffraction and Raman scattering investigations of (Li_xNa_{1-x})NbO₃ solid solutions: evidence of the rhombohedral phase. Phys. Rev. B, 69, 144105 (2004).

[148] Xu, S., Yeh, Y.-W., Poirier, G., McAlpine, M.C., Register, R.A., and Yao, N. Flexible piezoelectric PMN-PT nanowire-based nanocomposite and device. Nano Lett., 13, 2393-2398 (2013).

[149] Yan, C., Nikolova, L., Dadvand, A., Harnagea, C., Sarkissian, A., Perepichka, D.F., Xue, D., and Rosei, F. Multiple NaNbO₃/Nb₂O₅ heterostructure nanotubes: a new class of ferroelectric/semiconductor nanomaterials. Adv. Mater., 22 (15), 1741-1745 (2010).

[150] Mendoza, M., Khan, M.A.R., Shuvo, M.A.I., Guerrero, A., and Lin, Y., Development of lead-free nanowire composites for energy storage applications., ISRN Nanomater., 151748, 1-8 (2012).

[151] Haertling, G.H. PLZT electrooptic materials and applications—a review. Ferroelectrics, 75, 25 (1987).