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Temperature Evolution of the Double Umbrella Magnetic Structure in Terbium Iron Garnet

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1. Introduction

The rare earth iron garnets RE₃Fe₅O₁₂ (REIG hereafter) have been discovered at Grenoble (France) (Bertaut & Forrat, 1956; Bertaut et al., 1956) then independently at Murray Hill (USA) (Geller & Gilleo, 1957a, 1957b). These most studied ferrimagnetic materials have a general formula {RE³⁺₃}[Fe³⁺₂](Fe³⁺₃)O₁₂ where RE³⁺ can be any trivalent rare earth ion or the Yttrium Y³⁺. The crystal structure is described by the cubic space group Ia $\overline{3}$ d-(O_h¹⁰) No. 230. Three type of brackets are used to indicate the different coordinations of the cations with respect to the oxygen O²⁻ ions situated in the general positions x, y, z of the sites 96h(1) (Fig. 1). The RE³⁺ ions are located in the dodecahedral sites {24c}(222) whereas the two Fe³⁺ ions are distributed in the octahedral [16a]($\overline{3}$) and tetrahedral (24d)($\overline{4}$) sites.

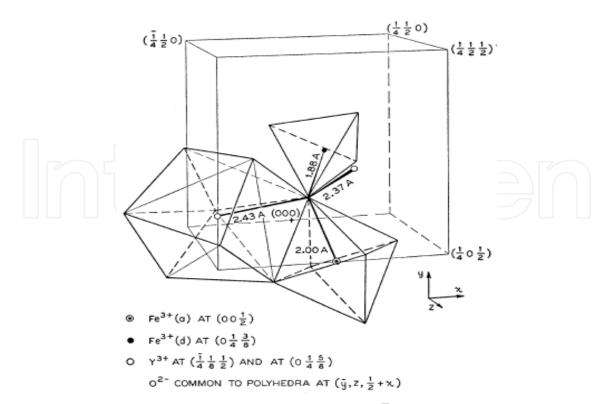


Fig. 1. Crystallographic sites of YIG in the space group Ia $\overline{3}$ d (Geller & Gilleo, 1957b)

It is a rather loose structure with a volume of 236.9 Å³ per formula unit, which has the great technical advantage that it is possible to accommodate a very large variety of cations in the garnet structure. Thus, it is feasible to achieve an enormous range of control of the magnetic properties in the garnet structure system. The largest of REIG that can be formed is SmIG with a lattice parameter of 12.529 Å and the smallest is LuIG with a lattice parameter of 12.283 Å. The REIG have became these famous magnetic compounds by illustrating the Néel theory of ferrimagnetism (Néel, 1948). The strongest superexchange interactions between the two iron sublattices Fe³⁺[a]-Fe³⁺(d) are antiferromagnetic and the magnetic moment vectors m_a and m_d are antiparallel. They make YIG, an ideal ferrimagnet with the Néel temperature (T_N) equal to 560 K (Pauthenet, 1958a, 1958b). However, substitution of a magnetic rare earth ion for the diamagnetic Y³⁺ in YIG introduces a third sublattice in the crystallographic site {c} in which m_{RE} are the magnetic moment vectors. In this three sublattices model only weaker and negative antiferromagnetic interactions RE³⁺{c}-Fe³⁺(d) exist. If M_a , M_d and M_{RE} are the magnetizations of each sublattice, the total bulk ferrite magnetization of REIG is given by the following equation

$$M(\text{REIG}) = |M_{\text{RE}} - (M_{\text{d}} - M_{\text{a}})| \tag{1}$$

For the heavier RE³⁺ ions (Eu³⁺, ..., Yb³⁺), M_{RE} is antiparallel to the net resultant of the iron magnetizations $M_{\text{Fe}} = M_{\text{d}} - M_{\text{a}}$. We can consider that if we have $M_{\text{Fe}} \approx M(\text{YIG})$ then the interactions between the rare earth ions are negligible and the equation (1) becomes in a first approximation

$$M(\text{REIG}) = |M_{\text{RE}} - M_{\text{Fe}}| \approx |M_{\text{RE}} - M_{\text{YIG}}|$$
(2)

The magnetizations M_a and M_d are still given by the N.M.R values found in YIG (Gonano et al., 1967). Below T_N which is nearly the same for all REIG compounds (554 ± 6) K (Pauthenet, 1958a, 1958b) the magnetization of the rare earth ions M_{RE} can dominate the magnetization M_{Fe} . If the temperature is decreasing, a rapid increasing of M_{RE} is observed because of the large magnetic moment m_{RE} . In heavy rare earth iron garnets, there exists a compensation temperature (T_{comp}) or inversion temperature (T_I) (Herpin, 1968) at which the bulk ferrite magnetization vanishes. For TbIG, T_{comp} is equal to (243.5 ± 0.5 K) and (249.0 ± 0.5 K) for the single crystal and powder samples respectively (Lahoubi et al., 1985; Lahoubi, 1986). In the vicinity of T_{comp} , the magnetic behavior is equivalent to that observed in the antiferromagnet compounds with the existence of the so-called field induced phase transitions which have been studied previously theoretically and experimentally (Zvezdin, 1995). In the Néel model, the RE³⁺ magnetic behavior is described by the pure free ion Brillouin function assuming that the superexchange interactions are represented by the isotropic Weiss molecular field coefficients.

The optical and magneto-optical (MO) properties of REIG and their substituted compounds have also received a substantial interest due to their strong Faraday and Kerr effects. The REIG had their first industrial use in bubble memories more than twenty years ago. Today, these MO materials are the key elements of several technical applications. There are used in Faraday rotators, optical isolators, holographic storage and magnetic field sensors. These applications can be enhanced by using photonic crystals with REIG and such research has yielded promising results. Recently, an additional interest to the REIG has been caused by the prospects for developing materials based on these ferrimagnets for hardware components in

the next-generation spintronic devices. The fundamental properties of these REIG and their applications have been enormously reviewed previously and only a few authors are reported here (Dahlbäck, 2006; Geller, 1977; Guillot, 1995; Kazei et al., 1991). At low temperature when both crystal field acting on the RE³⁺ ions and the RE³⁺{c}-Fe³⁺(d) magnetic exchange anisotropies become important (Nekvasil & Veltrusky, 1990), the Néel model must be replaced by a new spin configuration in which the RE³⁺ magnetic moments cease to be antiparallel to the Fe³⁺ magnetization M_{Fe} . So, neutron diffraction experiments have been performed previously to study, the non collinear magnetic structures of the RE³⁺ moments (RE = Dy, Er, Ho, Tb, Tm and Yb) which appear at liquid-helium temperatures (Bertaut et al., 1970; Guillot et al., 1982; Herpin et al., 1960; Hock et al., 1990, 1991; Lahoubi et al., 1984; Lahoubi, 1986, 2012; Pickart et al., 1970; Tchéou et al., 1970a, 1970b). Neutron diffractions experiments have been also made to follow the temperature dependence of these "*umbrella*" magnetic structures in HoIG (Guillot et al., 1984), ErIG (Hock et al., 1991), DyIG (Lahoubi et al., 2009, 2010).

In the present chapter, we will present the temperature evolution of the Tb³⁺ magnetic ordering in TbIG using powder neutron diffraction experiments combined with magnetic field magnetization measurements on single crystal (Lahoubi et al., 1984, 1985; Lahoubi, 1986; Lahoubi et al., 1997; Lahoubi, 2012). The experimental techniques are described in the section 2. The principle of the non-polarized neutrons diffraction with the preliminarily experiment at 614 K are introduced in the main section 3. The neutron diffraction results obtained at high and low temperatures are discussed using the predictions of the symmetry analysis and compared with the data of magnetization measurements in the sections 4 and 5 respectively. The "Representation Analysis" of Bertaut (Bertaut, 1968, 1971, 1972) is applied to the paramagnetic space group Ia $\overline{3}$ d for determining all possible "*umbrella*" magnetic structures in this "cubic description". The method of the so-called "symmetry lowering device" (Bertaut, 1981) is required in the treatment for the determination of the best subgroups of Ia $\overline{3}$ d when the temperature is decreasing below T_N until liquid-helium temperatures. The "basis vectors of irreductible representations" of the distorted space group R 3 c are chosen in the "rhombohedral description". The thermal variations of the parameters of the "double umbrella" magnetic structure constitute the section 6 which will be followed by a conclusion in the section 7.

2. Experimental techniques

The neutron diffraction experiments were measured on polycrystalline sample of TbIG owing to some severe extinctions which appear when a high quality single crystal is used (Bonnet et al., 1979). The first set of patterns have been recorded previously at the Centre d'Etudes Nucleaires de Grenoble (CENG) CEA Grenoble, France (Lahoubi et al., 1984; Lahoubi, 1986; Lahoubi et al., 1997) using the famous "Position Sensitive Detector" (PSD) detector (Convert et al., 1983; Roudaut et al., 1983). For the study, ten temperatures are chosen in the cryostat: 4.2, 20, 54, 68, 80, 109 ± 2, 127 ± 5, 160, 208 ± 2 and around T_{comp} at 244 ± 10 K. In the furnace, the temperatures are: above T_N (T = 614 K) and below T_N (T = 283, 400 and 453 K). The time of counting for some temperatures has been in the order of ten hours. The patterns were recorded with a wavelength λ equal to 2.49 Å and filters to avoid $\lambda/2$ contaminations are used. The second set of patterns has been collected recently on the high flux diffractometer D1B at the Institut Laue–Langevin Grenoble, France. The value of the wavelength λ is equal to 2.52 Å and four temperatures 5, 13, 20 and 160 K have been chosen below room temperature. The resolution of the multidetector is 0.2°. For the magnetic study, the magnetization of a flux-

grown single crystal of TbIG was measured in dc magnetic fields produced either by a superconducting coil up to 80 kOe or a Bitter coil up to 150 and 200 kOe. The first series of experiments was performed at the Louis Néel Laboratory of Grenoble, France (currently Néel Institut) and the second series at the Service National des Champs Intenses (SNCI) of Grenoble (currently LNCMI). In the 4.2–300 K temperature range, the external magnetic field was applied parallel to the crystallographic directions <111>, <110> and <100> successively. The spherical sample with 5.5 mm of diameter and 0.4313 g of weight is oriented along these three main crystallographic directions by the X-ray Laüe technique with an error less than 1°. The isothermal magnetizations $M_{\rm T}(H)$ as a function of internal magnetic field H (the external magnetic field $H_{\rm ex}$ minus the demagnetizing magnetic field of the sphere $H_{\rm d}$) are analyzed by the least-squares refinement technique. The measured spontaneous magnetizations $M_{\rm S}^{\rm mes}(T)$ are reported in ($\mu_{\rm B}/$ mol) where one mole is equal to 2(TbIG) formula units.

3. Principle of neutron diffraction

We present here only the case of the diffraction of the non-polarized (or unpolarized) neutrons. This method was used firstly in the determination of the magnetic structure in MnO (Shull & Smart, 1949). A multitude of others followed after: approximately thousands of magnetic structures have now been solved. The use of polarized neutrons ten years after has been the next progress in magnetic neutron scattering (Nathans et al., 1959). In this second method, the incident neutron beam is polarized either up or down, and the neutron intensities scattered by the sample are compared for the two possible states of the incident polarization. Compared to the non-polarized neutrons experiments, an interference term between the nuclear and the magnetic amplitudes adds or subtracts to the intensities depending on the direction of the polarization. For small magnetic amplitudes, the enhancement of sensitivity is remarkable and we know in the case of ferromagnetic and ferrimagnetics to determine the form factor of the magnetic atoms and to reconstruct the spin (or magnetization) density within the cell. Such investigations are now very common: several hundred spin density investigations have already been performed. In YIG (Bonnet et al., 1979), the polarized neutrons were used in addition to the study of the covalency effects. A book edited recently (Chatterji, 2006) is mostly devoted to the application of polarized neutron scattering from magnetic materials.

3.1 Non-polarized neutrons diffraction: Determination of the diffracted intensities

The scattering of neutron by a magnetic atom is composed essentially of two terms: a nuclear neutron scattering and a magnetic neutron scattering. The first term is due to a "*neutron-nucleus*" nuclear interaction giving the nuclear diffraction which yields information on the spatial arrangement of the nuclei of the atoms in crystal. In the second term, the neutron has a magnetic moment which can interact with the unpaired electrons of the magnetic atoms through the "*dipole-dipole*" interaction conducting to the magnetic diffraction (Bacon, 1975). The nuclear and the magnetic neutron scattering are incoherent in our non-polarized neutrons diffraction. For a magnetic material in the paramagnetic state obtained at $T > T_N$, there is no magnetic contribution and only the nuclear diffraction exists with the diffracted nuclear intensity (I_N). The total diffracted intensity $I(T > T_N)$ is equal to

$$I(T > T_{\rm N}) = I_{\rm N} \tag{3}$$

206

In the ordered magnetic state ($T < T_N$) a magnetic intensity (I_M) is added to the intensity I_N . We can measure finally the total diffracted intensity $I(T < T_N)$ with the following equation

$$I(T < T_{\rm N}) = I_{\rm N} + I_{\rm M} \tag{4}$$

The diffracted magnetic intensity $I_{\rm M}$ is then obtained by the difference

$$I_{\rm M}(T < T_{\rm N}) = I - I_{\rm N} \tag{5}$$

This method is applied if we have a good counting statistics and also if I_M is not too lower then I_N . We will recall briefly the following useful expressions of the nuclear and magnetic diffracted intensities for the case of a polycrystalline sample.

3.1.1 Nuclear intensity IN

In the paramagnetic state, the nuclear intensity $I_N(H)$ can be calculated through the nuclear structure factor $F_N(H)$ with the following expressions

$$I_{\rm N}(\boldsymbol{H}) = P(\boldsymbol{H})|F_{\rm N}(\boldsymbol{H})|^2 \tag{6}$$

$$F_{\rm N}(H) = \sum_{j=1}^{\rm N} b_{j} \exp(2\pi i H \cdot r_{j}) \exp(-B_{j} |H|^{2} / 2)$$
(7)

with P(H), the multiplicity of the reflector plane, and H, the corresponding scattering vector which is concentrated in the Bragg peaks (hkl) of the reciprocal lattice, b_j and B_j being respectively the neutron scattering length and the Debye-Waller factor of the jth atom in the jth position vector r_j among N, the total number of atoms in the crystallographic unit cell. The nuclear structure factor appears as a scalar factor.

3.1.2 Magnetic intensity IM

When the materials are magnetically ordered, the magnetic diffracted intensity $I_M(H)$ has the same form that found before for $I_N(H)$ but we consider only the perpendicular component to H of the magnetic structure factor $F_M(H)$. We can write these equations

$$I_{M}(H) = ||F_{M}(H)|^{2} - |H \cdot F_{M}(H)|^{2} / |H|^{2}|$$
(8)

$$F_{\rm M}(H) = \eta \sum_{j=1}^{N} S_{jj}(H) \exp(2\pi i H \cdot r_j) \exp(-B_j |H|^2 / 2)$$
(9)

with $f_j(H)$, the magnetic form factor of the jth spin S_j , and $\eta = |\gamma| e^2/2m_0 c = +0.02696$ cm, the magnetic diffusion length ($\gamma = -1.91348$). The magnetic structure factor has a vector form.

3.1.3 Observed Iobs and calculated Ical intensities

The observed diffracted intensity I_{obs} is integrated on number of counts. The value (in barns) is corrected by the Lorentz factor L(H) and normalized by the scale factor K where the definitions are expressed respectively by the following equations

$$L(\mathbf{H}) = 1 / \sin\theta \sin(2\theta); K = I_{cal} / I_{obs}$$
(10)

The value of *K* is obtained by the refinement of the nuclear structure. For each I_{obs} , an absolute error ΔI_{obs} is found. She is associated to the sum of a statistical error with the appreciated one on the continu fund noise. The corresponding observed relative error is noted $\Delta I_{obs}/I_{obs}$. Using the least square method, we can define the reliability factor *R*

$$R = \sum |I_{\rm obs} - I_{\rm cal}| / \sum |I_{\rm obs}|$$
(11)

For each calculated diffracted intensity I_{cal} , a calculated relative error $\Delta I_{cal}/I_{cal}$ is attributed. The non-polarized neutron diffraction is then based in the comparison between I_{cal} and I_{obs} .

3.2 Preliminarily neutron diffraction at 614 K

In the pattern (not show) recorded at T = 614 K higher than T_N (568 ± 2 K) (Pauthenet, 1958a, 1958b), only the nuclear contribution exists and the reflections (hkl) are indexed with the general extinction rule of the cubic space group Ia $\overline{3}$ d, h + k + 1 = 2n. Attention was paid to the thermal agitation of the jth atom in the different sites by introducing the corresponding isotropic Debye-Waller factors B_i. For the determination of the calculated nuclear intensities $(I_N)_{cal}$, the previous scattering lengths (Bacon, 1972) are used: b(O) = 0.580; b(Fe) = 0.95; b(Tb)= 0.76 (in units of 10^{-12} cm/atom). We can determine the number of refinement cycles and to choose the parameters with a sufficient number of iterations. The previous best values of the parameters x = -0.0279, y = 0.0555, z = 0.1505 found previously at T = 693 K (Tchéou et al., 1970c) for the general positions (96h) of the oxygen atoms are used as constant parameters and the refinement is made only on the scale factor *K* and after the corresponding isotropic Debye-Waller factors B_j. A good agreement with a reliability factor of order 10 % is found for the lattice parameter $a = 12.470 \pm 0.004$ Å and $K = 0.42 \pm 0.02$ with $B_h(O) = 0.88$; $B_d(Fe) =$ $B_a(Fe) = 0.82$ and $B_c(Tb) = 0.81$. We observe that the observed intensities of the reflections (400) and (420) are lower than the corresponding calculated intensities and the refinement of the temperature parameters B_i has a little influence on the values of the observed intensities (for example when $B_c(Tb)$ change from 0.80 to 0.30).

4. Neutron diffraction study at high temperature

The neutron diffraction patterns below T_N are reported in Fig. 2 for 453 and 400 K. Both patterns at (283 K) and at 5 K (D1B) are presented for a useful comparison in Fig. 3. A magnetic intensity $I_M(hkl)$ is superimposed to the nuclear intensity $I_N(hkl)$: we have then, $I(hkl) = I_N(hkl) + I_M(hkl)$. In these patterns, we have indexed all the reflections (hkl) in the chemical cell with the same extinction rule characterized by a wave vector k = 0. In these temperatures, the magnetic structure factors $F_M^{(a)}(hkl)$, $F_M^{(d)}(hkl)$ and $F_M^{(c)}(hkl)$ associated to each magnetic sublattice of TbIG are used to describe the collinear ferrimagnetic state of the Néel model along the easy axis [111] found by magnetization measurements (Lahoubi et al., 1985 and refs. herein). Two types of reflections (hkl) appear in the patterns: (h = odd, k = odd, l = even) and h = even, k = even, l = even). For the related reflections (hkl) with (h = even, k = odd, l = odd) and (h = odd, k = even, l = odd) and cyclic permutations (c.p.) of h, k, l must be done in the expressions of the magnetic structure factors. A complete description can be found in the previous paper on the neutron diffraction of HoIG (Guillot et al., 1984).

208

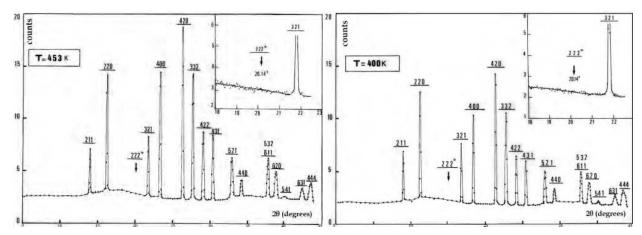


Fig. 2. Neutron diffraction patterns at 453 and 400 K

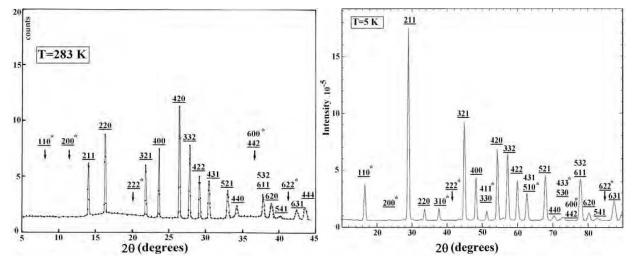


Fig. 3. Neutron diffraction patterns at 283 and 5 K (D1B)

4.1 Results and discussion

We have observed during the refinement that the calculated parameters m_{Tb} , m_{a} and m_{d} of the three magnetic sublattices are depending highly of the choice of the magnetic form factors, $f_{\rm Ib}$ (hkl) and particularly those related to the two iron sublattices f_a (hkl) and f_d (hkl). We choose at first, the theoretical magnetic form factors determined in the Hartree-Fock description based on the free ion model for the Tb³⁺ ion, f_{Tb} (hkl) (Blume, et al., 1962) and those calculated (Watson & Fremann, 1961) $f_a(hkl)$ and $f_d(hkl)$ with the equality $f_a(hkl) = f_d(hkl)$ for the iron sublattices. Secondly, two previous experimental magnetic form factors f_a(hkl) and f_d(hkl) are also tentatively used. The first values (Bonnet, 1976; Bonnet et al., 1979) obtained by polarized neutrons experiments on YIG single crystal indicate that $f_a(hkl)$ and $f_d(hkl)$ are different that the free ion value and $f_a(hkl) > f_d(hkl)$. The second values of $f_a(hkl)$ and $f_d(hkl)$ with the relation $f_a(hkl) < f_d(hkl)$ have been found previously by powder neutron diffraction experiments (Guillot et al., 1983). In the study of the "umbrella structure" at low temperature on HoIG sample prepared by grinding of single crystals (Guillot et al., 1984), an evaluation of m_a and m_d using the Bonnet' f_a(hkl), f_d(hkl) determinations was made. At 4.2 K, these moments were found equal respectively to 4,01 and 4,26 μ_B . These values are smaller than the theoretical ground state ${}^{6}S_{5/2}$ saturated magnetic moment (5 μ_{B}). The observed reduction of the moments

is explained by covalent bonding for YIG (Bonnet, 1976; Fuess et al., 1976) or topological frustration for FeF3 (Ferey et al., 1986). When the proposed values (Guillot., et al 1983) are chosen in the refinement of the neutron diagrams at 453, 400 and 283 K, we observe that the calculated intensities of the reflections (211) and (220) which have a high magnetic contribution are lower than the observed intensities. Consequently, we shall consider in this work for the Fe³⁺ ions only the theoretical values of $f_a(hkl)$ and $f_d(hkl)$ (Watson & Fremann, 1961). In this condition, m_a and m_d are not considered as fitting parameters in the first cycle of the refinement and the N.M.R values are used (Gonano et al., 1967). Working in this hypothesis leads to the best values of $m_{\rm Tb}$ at each temperature. Finally, in the second cycle of the refinement based on twenty reflections, the parameters m_a and m_d are fitted by the selfconsistent calculation of $m_{\rm Tb}$. At 453 and 400 K, we obtain respectively for $m_{\rm Tb}$ the refined values $0.50 \pm 0.10 \mu_B$ and $0.60 \pm 0.10 \mu_B$ with a reliability factor R varying in the range of 11.8– 9.3 and 11.6-8.6 % if the refinement is makes only on the twelve first reflections (Lahoubi, 1986). The results lead to a good agreement between the bulk calculated magnetizations $M_{\rm S}^{\rm cal}$ (TbIG) and the observed spontaneous magnetization $M_{\rm S}^{\rm obs}$ (TbIG) (Pauthenet, 1958a, 1958b). The result at 283 K (1.15 μ_B with R = 8.5 %) (Lahoubi, 1986) is similar to that found previously (Bonnet, 1976; Fuess et al., 1976) but with a poor agreement for m_a and m_d which have been found lower by comparison with those determined by N.M.R (Gonano et al., 1967).

5. Neutron diffraction study at low temperature and symmetry analysis

The neutron patterns recorded at 5 (D1B), 13 (D1B), 20, 54, 68, 80, 109, 127, 160, 208 and 244 K are presented from Fig. 3 up to Fig. 9.

5.1 Results and discussion

At T = 5 K (Fig. 3), two types of reflection appear. In addition to the earlier reflections (hkl) observed previously, pure superstructure lines (hkl)^{*} forbidden by the nuclear space group Ia $\overline{3}$ d are present and we have $I = I_{\rm M}$ (hkl)^{*}: (110)^{*}, (310)^{*}, (411, 330)^{*}, (433, 530)^{*} and (510)^{*}.

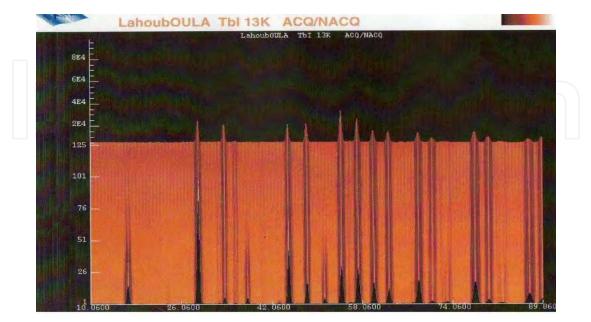


Fig. 4. Neutron diffraction pattern at 13 K (D1B)

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210

In contrast to previous findings at 1.5 K (Bertaut et al., 1970; Tchéou al., 1970a) and 4.2 K (Lahoubi et al., 1984), the small superstructure lines (200)* and (600, 442)* have been observed recently and confirmed only at 5 K (D1B) (Lahoubi, 2012) with a sensibility equal to 0.5 and 1% respectively, the line (110)* being chosen as a reference. The same order of magnitude (1/276) by comparison with the previous result (Hock et al., 1990) was found for the line (200)*. Above 5 K, they are not observed in the neutron diffraction pattern at 13 K (Fig. 4).

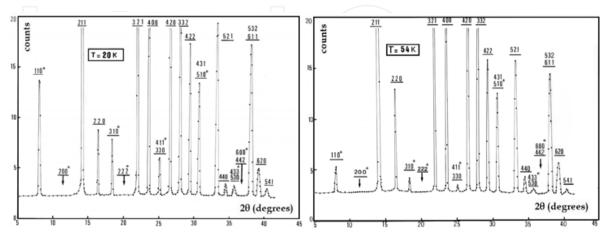
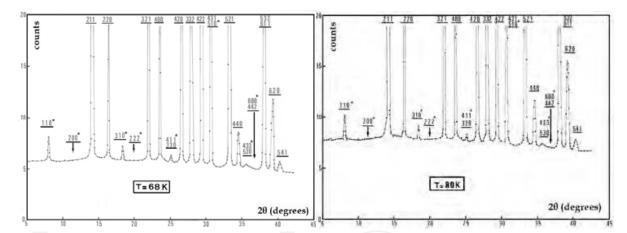
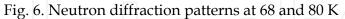


Fig. 5. Neutron diffraction patterns at 20 and 54 K





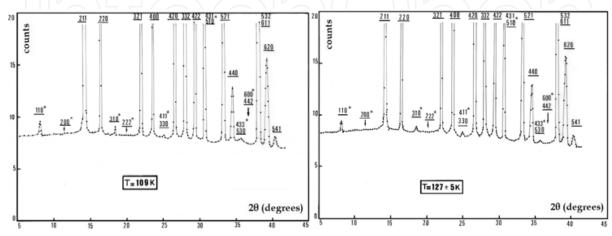


Fig. 7. Neutron diffraction patterns at 109 and 127 ± 5 K

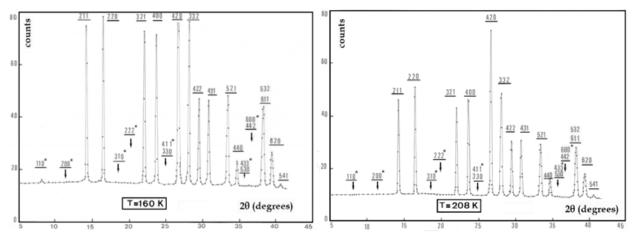


Fig. 8. Neutron diffraction patterns at 160 and 208 K

All the superstructure lines appear without any ambiguity from 5 K up to 127 ± 5 K. They are indexed with the same extinction rule (h^{*} + k^{*} + l^{*} = 2n) and imply the signature of a change of the collinear magnetic structure of the RE³⁺{24c} ions of the Néel model to a non collinear magnetic structure with the wave vector k = 0. At 160 and 208 K, the superstructure line (110)^{*} is resolved with the best sensibility (0.3 and 1% respectively) which is equal to the ratio of the peak to background normalized to the intensity of the line (211) (see the details for 20 up to 15° in the left of Fig. 9).

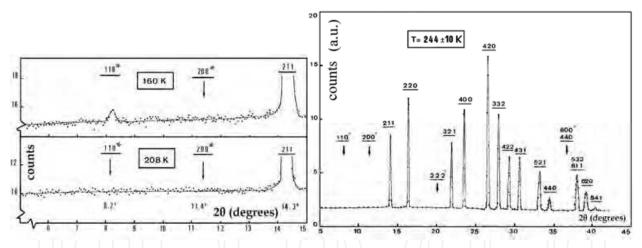


Fig. 9. Details at 160 and 208 K (left) and neutron diffraction pattern at 244 ± 10 K (right)

The chemical cell is equal to the magnetic cell, thus the primitive translation noted (1|1/2,1/2,1/2) of the crystallographic lattice (I) (Hahn, 1983) is a primitive translation of the magnetic lattice (I). Based on the numbered positions gathered on Table 1 we can write for the RE³⁺{24c} and Fe³⁺(24d) sublattices that $S_j = S_j + S_{j+12}$ (j = 1-3); for Fe³⁺[16a] sublattice, we have $S_p = S_p + S_{p+8}$ (p = 1-4) and $S_{p'} = S_{p'} + S_{p'+8}$ (p' = 5-8). It means that two spin vectors S_j , S_{j+12} , S_p , S_{p+8} and $S_{p'}$, $S_{p'+8}$ are coupled ferromagnetically. To discuss the corresponding magnetic structure factors $F_{M}^{(a)}(hkl)^*$, $F_{M}^{(d)}(hkl)^*$ and $F_{M}^{(c)}(hkl)^*$ associated to each magnetic sublattice in TbIG, the four earlier linear combinations of four spin vectors introduced by Bertaut (Bertaut, 1963) labeled *F*, *G*, *C* and *A* are used. These four magnetic modes (one ferromagnetic and three antiferromagnetics) form the "basis of irreductible representations".

Temperature Evolution of the Double Umbrella Magnetic Structure in Terbium Iron Garnet

{i =	1-6} (W	C _j , C' _j (j = 1-3) and olf et al., 1962; Wolf, earson et al., 1965)	Numb	tions $D_{j'}$ D'_j ($j = 1-3$) bered positions of bins in site (24d)($\overline{4}$)	Unprimed and primed (') notations respectively for p = 1-4 and p' = 5-8 with c.p.			
Loc U V	0-			$Fe^{3^{\prime}} \text{ ions in site } (24a)(4)$		Numbered positions of Fe ³⁺ ions in site $[16a](\overline{3})$		
W Nu	g _y (g _x) mbered	positions of RE ³⁺	6					
		${24c}(222)$						
C ₁ 1 3	100 011 011	(1) 1/8,0,1/4 (7) 7/8,0,3/4 (13) 5/8,1/2,3/4 (19) 3/8,1/2,1/4	D1	(1) 3/8,0,1/4 (7) 5/8,0,3/4 (13) 7/8,1/2,3/4 (19) 1/8,1/2,1/4		(1) 0,0,0 (9) 1/2,1/2,1/2		
C'1 2 4	100 011 011	(4) 3/8,0,3/4 (10) 5/8,0,1/4 (16) 7/8,1/2,1/4 (22) 1/8,1/2,3/4	D'1	(4) 1/8,0,3/4 (10) 7/8,0,1/4 (16) 5/8,1/2,1/4 (22) 3/8,1/2,3/4		(2) 0,1/2,1/2 (10) 1/2,0,0		
C ₂ 3 5	$ \begin{array}{c} 010 \\ \overline{1}01 \\ 101 \end{array} $	 (2) 1/4,1/8,0 (8) 3/4,7/8,0 (14) 3/4,5/8,1/2 (20) 1/4,3/8,1/2 	D ₂	 (2) 1/4,3/8,0 (8) 3/4,5/8,0 (14) 3/4,7/8,1/2 (20) 1/4,1/8,1/2 		(3) 1/2,0,1/2 (11) 0,1/2,0		
C'2 4 6	010 101 101	 (5) 3/4,3/8,0 (11) 1/4,5/8,0 (17) 1/4,7/8,1/2 (23) 3/4,1/8,1/2 	D'2	 (5) 3/4,1/8,0 (11) 1/4,7/8,0 (17) 1/4,5/8,1/2 (23) 3/4,3/8,1/2 		(4) 1/2,1/2,0 (12) 0,0,1/2		
C ₃ 5 1	001 110 110	 (3) 0,1/4,1/8 (9) 0,3/4,7/8 (15) 1/2,3/4,5/8 (21) 1/2,1/4,3/8 	D ₃	 (3) 0,3/4,3/8 (9) 0,3/4,5/8 (15) 1/2,3/4,7/8 (21) 1/2,3/4,3/8 	(')	 (5) 1/4,1/4,1/4 (13) 3/4,3/4,3/4 		
C' ₃ 6 2	001 110 110	 (6) 0,3/4,3/8 (12) 0,1/4,5/8 (18) 1/2,1/4,7/8 (24) 1/2,3/4,1/8 	D'3	(6) 0,3/4,1/8 (12) 0,1/4,7/8 (18) 1/2,1/4,5/8 (24) 1/2,3/4,3/8	(')	 (6) 1/4,3/4,3/4 (14) 3/4,1/4,1/4 		
					(')	 (7) 3/4,1/4,3/4 (15) 1/4,3/4,1/4 		
					(')	 (8) 3/4,3/4,1/4 (16) 1/4,1/4,3/4 		

Table 1. Notations, numbers and positions in the unit cell of the three magnetic sublattices. U, V and W are the local axes of the RE³⁺ in D2(222) symmetry

213

The adapted magnetic modes for the $RE^{3+}{24c}$ and $Fe^{3+}(24d)$ sublattices with j = 1-3 are

$$F_{j} = S_{j} + S_{j+3} + S_{j+6} + S_{j+9}; G_{j} = S_{j} - S_{j+3} + S_{j+6} - S_{j+9};$$

$$C_{j} = S_{j} + S_{j+3} - S_{j+6} - S_{j+9}; A_{j} = S_{j} - S_{j+3} - S_{j+6} + S_{j+9}$$
(12)

For the Fe³⁺[16a] sublattice, it is necessary to consider two distinct magnetic modes. The first chosen notation is the unprimed magnetic modes for the numbered spins $S_p = S_p + S_{p+8}$ (p = 1-4)

$$F = S_1 + S_2 + S_3 + S_4; G = S_1 - S_2 + S_3 - S_4; C = S_1 + S_2 - S_3 - S_4; A = S_1 - S_2 - S_3 + S_4$$
(13)

The second chosen notation of the magnetic modes is the primed notation (') for the numbered spins $S_{p'} = S_{p'} + S_{p'+8}$ (p' = 5-8)

$$F' = S_5 + S_6 + S_7 + S_8; G' = S_5 - S_6 + S_7 - S_8; C' = S_5 + S_6 - S_7 - S_8; A' = S_5 - S_6 - S_7 + S_8$$
(14)

For the RE³⁺ ions in the Wyckoff site {24c}, the local axes (U,V,W) in the D₂(222) symmetry are identified to the parameters g_{α} ($\alpha = x,y,z$) of the magnetic tensor \tilde{g} in the hypothesis of the effective spin Hamiltonian model (Wolf et al., 1962, Wolf, 1964). The particular superstructure lines (222)* and (622)* are not observed in the whole temperature range below $T_{\rm N}$. For example, the associated magnetic structure factors are all equal to zero

$$F_{\rm M}^{(a)}(222)^{*} = F - F' = 0; F_{\rm M}^{(d)}(222)^{*} = +i(A_{1} + A_{2} + A_{3}) = 0$$
(15)

$$F_{\rm M}^{\rm (c)}(222)^{*} = -i(A_{1} + A_{2} + A_{3}) = 0; F_{\rm M}^{\rm (c)}(622)^{*} = +i(A_{1} - A_{2} - A_{3}) = 0$$
(16)

In these conditions, the spins vectors S_j and S_{j+6} (j = 1-3) of Tb³⁺{24c} and Fe³⁺(24d) ions are coupled ferromagnetically and the symmetry operation $(\overline{1}|0,0,0)$ is an inversion center. These significant properties related to magnetic symmetry involve that all the magnetic modes A_i are absent. This absence will be accompanied by the elimination of the modes C_i . Consequently, the above crystallographic sites split into six magnetically inequivalent sublattices C_{i} , C'_{i} and D_{i} , D'_{i} with (j = 1-3) respectively as it is indicated on Table 1. In each sublattice we have four ions which are equivalent under the symmetry operations: $(1 \mid 0,0,0)$ and (1 | 1/2, 1/2, 1/2). The sublattices C₂ and C₃ are related to C₁ by a rotation of 120 and 240° around the 3-fold symmetry [111] axis (also for C'_i); the same remarks can be made for the sublattices D_{i} , D'_{i} (j = 1-3). Contrary to previous spin rotation observed at T_{comp} (260 ± 5 K) by mössbauer spectroscopy (Hong et al., 2004), no deviation from the colinearity along the easy axis [111] for the spins of the Fe³⁺[16a] ions is evidenced around $T_{\rm comp}$ (244 ± 10 K) (right of Fig. 9). In this *T*-region, this sublattice is described by one ferromagnetic configuration of the cubic magnetic modes F and F' with the equality of the left of the equation (15). The magnetic structure factors $F_{M^{(c)}}(hkl)^*$ calculated for all the observed superstructure lines imply to know if we need both magnetic modes $F_{j}^{(c)}$ and $G_{j}^{(c)}$ in the description of the non collinear magnetic structures at low temperature

$$F_{\rm M}^{\rm (c)} (100)^* = \sqrt{2} / 2(G_1 - G_2) + i[\sqrt{2} / 2(C_1 + C_2) + A_3]$$
(17)

$$F_{\rm M}^{\rm (c)}(200)^* = -F_2 + F_3 + iA_1 \tag{18}$$

$$F_{\rm M}^{\rm (c)}(310)^* = \sqrt{2} / 2(-G_1 + G_2) + i[\sqrt{2} / 2(C_1 - C_2) + A_3]$$
(19)

$$F_{\rm M}^{(\rm c)} (411)^* = \sqrt{2} / 2(G_2 - G_3) + i[\sqrt{2} / 2(C_2 + C_3) - A_1]$$
⁽²⁰⁾

$$F_{\rm M}^{\rm (c)}(330)^{*} = \sqrt{2} / 2(-G_{1} + G_{2}) + i[\sqrt{2} / 2(C_{1} + C_{2}) - A_{3}]$$
(21)
$$F_{\rm M}^{\rm (c)}(510)^{*} = \sqrt{2} / 2(-G_{4} - G_{2}) + i[\sqrt{2} / 2(-C_{4} + C_{2}) + A_{3}]$$
(21)

$$F_{\rm M}^{\rm (c)} (433)^* = \sqrt{2} / 2(-G_2 + G_3) + i[\sqrt{2} / 2(C_2 + C_3) + A_1]$$
(23)

$$F_{\rm M}^{\rm (c)}(530)^* = \sqrt{2} / 2(-G_1 - G_2) + i[\sqrt{2} / 2(-C_1 - C_2) - A_3]$$
(24)

$$F_{\rm M}^{\rm (c)} (600)^* = -F_2 + F_3 - iA_1$$
⁽²⁵⁾

$$F_{\rm M}^{(\rm c)} (442)^* = -F_2 + F_1 + iA_3$$
⁽²⁶⁾

From these expressions, one can observe that both magnetic modes $F_{j}^{(c)}$ and $G_{j}^{(c)}$ are necessary in the description of the non collinear structures of the Tb³⁺ ions. The absence of the small superstructure lines (200)^{*} and (600, 442)^{*} above 13 K gives rise to the equality between the magnetic modes $F_{j}^{(c)}$ (j = 1-3). It must be noted that the superstructure line (510)^{*} appears at the same Bragg peak of the pure nuclear reflection (431) (θ = 30.7°). A magnetic contribution of the RE³⁺ ions exists for (431). We observe also a magnetic contribution for another reflection (541). It is very difficult to isolate only the magnetic contribution of these two reflections which are represented by their magnetic structure factors

$$F_{\rm M}^{\rm (c)}(431) = \sqrt{2} / 2(-G_2 + G_3) + i[\sqrt{2} / 2(C_2 - C_3) - A_1]$$
⁽²⁷⁾

$$F_{\rm M}^{\rm (c)}(541) = \sqrt{2} / 2(G_1 + G_3) + i[\sqrt{2} / 2(-C_1 + C_3) - A_2]$$
(28)

5.2 Representation analysis of magnetic structures

Bertaut (Bertaut, 1963, 1968, 1971, 1972) has created a group theory method called "*Representation Analysis*" which has been widely used in the last four decades by the Bertaut himself and by other researchers. The essential role is plaid by the "*Basis Vectors of Irreductible Representations*" of the paramagnetic space group Ia $\overline{3}$ d of TbIG and its highest subgroups.

5.2.1 Representation analysis of la $\overline{3}$ d

The representation analysis of the cubic space group Ia $\overline{3}$ d was applied in the past (Bertaut et al., 1970; Tchéou et al., 1970a) in order to determine the spin configurations of the Tb³⁺

ions in TbIG. In the point group O_h ten possible irreductible representations Γ_{ig} and Γ_{iu} (i = 1-5) are present and listed usually on quantum mechanics text books (Flury Jr, 1980; Kahan, 1972). The subscripts g and u refer to irreductible representations which are even (gerade) and odd (ungerade) respectively under the inversion $I = (1 \mid 0, 0, 0)$. In their original works, two representations ($\Gamma_{4g} = T_{1g}$) and ($\Gamma_{5g} = T_{2g}$) were used. However, the coupling between the RE^{3+} in sites {24} and the two iron ions Fe^{3+} in sites [16a] and (24d) has not been taken account in their study. The representation analysis of Ia3d has been developed later completely by the author (Lahoubi, 1986) in order to choose the common irreductible representation which could be able to represent both the Néel model at high temperature below T_N and the non collinear magnetic structures observed below 160 K. For both RE³⁺{24c} and Fe³⁺(24d) sublattices, we need the even Γ_{ig} (i = 1-5) which appear only for Fe³⁺[16a] sublattice. Using the precedent linear combinations of the spins of equations (12), (13) and (14), the sets of magnetic basis vectors of the three sublattices belonging to Γ_{ig} (i = 1-5) which are formed by the functions Ψ_{lm} (l = 1-d_{ig}, d_{ig} the dimension of Γ_{ig} and m fixed) are listed on Tables 2, 3 and 4. The letters A and E are assigned to one and two dimensional representations where $(\Psi_{11})^*$ belongs to the complex conjugate representation with ϵ = $\{\exp(2\pi i/3)\}$. Due to the equation (13), it can be show that $(\Gamma_{5g} = T_{2g})$ is excluded and only $(\Gamma_{4g} = T_{1g})$ may be used to describe formally in a first approximation all magnetic structures.

$\Gamma_{1g} = A_{1g}$			
$\Gamma_{2g} = A_{2g}$	$\Psi_{11} = G_{1x} + G_{2y}$	+ G _{3z}	
$\Gamma_{3g} = E_g$	$Ψ_{11} = G_{1x} + εG_{2y}$ $Ψ_{21} = -ε(Ψ_{11})^*$	$_{7} + \epsilon^2 G_{3z}$	
$\Gamma_{4g} = T_{1g}$	$\Psi_{11} = F_{1x}$ $\Psi_{21} = F_{2y}$ $\Psi_{31} = F_{3z}$	$\Psi_{12} = F_{2x} + F_{3x}$ $\Psi_{22} = F_{1y} + F_{3y}$ $\Psi_{32} = F_{1z} + F_{2z}$	$\Psi_{13} = G_{2z} + G_{3y}$ $\Psi_{23} = G_{3x} + G_{1z}$ $\Psi_{33} = G_{1y} + G_{2x}$
$\Gamma_{5g} = T_{2g}$	$\Psi_{11} = F_{3x} - F_{2x}$ $\Psi_{21} = F_{1y} - F_{3y}$ $\Psi_{31} = F_{2z} - F_{1z}$		$\Psi_{13} = G_{3y} - G_{2z}$ $\Psi_{23} = G_{1z} - G_{3x}$ $\Psi_{33} = G_{2x} - G_{1y}$

Table 2. Basis vectors of the RE³⁺ in site $\{24c\}$ of Ia $\overline{3}$ d

$\Gamma_{1g} = A_{1g}$	$\Psi_{11} = G_{1x} + G_{2y} + G_{2y}$	G _{3z}		
$\Gamma_{2g} = A_{2g}$				
$\Gamma_{3g} = E_g$	$\Psi_{11} = G_{1x} + \epsilon G_{2y} + \Psi_{21} = + \epsilon (\Psi_{11})^*$	$-\epsilon^2 G_{3z}$		
	$\Psi_{11} = F_{1x}$	$\Psi_{12} = F_{2x} + F_{3x}$	$\Psi_{13} = G_{3y} - G_{2z}$	
$\Gamma_{4g} = T_{1g}$	$\Psi_{21} = F_{2y}$	$\Psi_{22} = F_{1y} + F_{3y}$	$\Psi_{23} = G_{1z} - G_{3x}$	
	$\Psi_{31} = F_{3z}$	$\Psi_{32} = F_{1z} + F_{2z}$	$\Psi_{33} = G_{2x} - G_{1y}$	
	$\Psi_{11} = F_{3x} - F_{2x}$	$\Psi_{12} = G_{3y} + G_{2z}$		
$\Gamma_{5g} = T_{2g}$	$\Psi_{21} = F_{1y} - F_{3y}$	$\Psi_{22} = G_{1z} + G_{3x}$		
	$\Psi_{31} = F_{2z} - F_{1z}$	$\Psi_{32} = G_{2x} + G_{1y}$		

Table 3. Basis vectors of the Fe³⁺ in site [24d] of Ia $\overline{3}$ d

$\Gamma_{1g} = A_{1g}$	$\Psi_{11} = (A_x + C_y + G_z) - (G'_x + A'_y + C'_z)$	
$\Gamma_{2g} = A_{2g}$	$\Psi_{11} = (A_x + C_y + G_z) + (G'_x + A'_y + C'_z)$	
Г. – Е	$\Psi_{11} = A_x + \epsilon C_y + \epsilon^2 G_z \qquad \Psi_{12} = - (A'_y + \epsilon^2 G'_x + \epsilon C'_z)$	
$\Gamma_{3g} = E_g$	$\Psi_{21} = -(A'_y + \epsilon G'_x + \epsilon^2 C'_z) \Psi_{22} = A_x + \epsilon^2 C_y + \epsilon G_z$	
	$\Psi_{11} = F_x + F'_x$ $\Psi_{12} = C_z + C'_y$ $\Psi_{13} = G_y + A'_z$	
$\Gamma_{4g} = T_{1g}$	$\Psi_{21} = F_y + F'_y$ $\Psi_{22} = G_x + G'_z$ $\Psi_{23} = A_z + C'_x$	
	$\Psi_{31} = F_z + F'_z$ $\Psi_{32} = A_y + A'_x$ $\Psi_{33} = C_x + G'_y$	
) <i> (4</i>	$\Psi_{11} = F_x - F'_x$ $\Psi_{12} = C_z - C'_y$ $\Psi_{13} = G_y - A'_z$	$\left(\begin{array}{c} \\ \end{array} \right)$
$\Gamma_{5g} = T_{2g}$	$\Psi_{21} = F_y - F'_y$ $\Psi_{22} = G_x - G'_z$ $\Psi_{23} = A_z - C'_x$	
	$\Psi_{31} = F_z - F'_z$ $\Psi_{32} = A_y - A'_x$ $\Psi_{33} = C_x - G'_y$	

Table 4. Basis vectors of the Fe³⁺ in site (16a) of Ia $\overline{3}$ d

In the high temperature region, the observed spectra which are well interpreted within the ferrimagnetic model of Néel are easily identified to the magnetic modes F_{j} , F and F'. At $T_{comp} = 243.5$ K the mean exchange field acting on the Tb³⁺ ions by the iron sublattices is too strong (~ 174 kOe) by comparison with the Tb³⁺–Tb³⁺ exchange field (~ 8 kOe) (Lahoubi, 1986); this last coupling beetwen the Tb3+ ions will be not able to decouple at low temperature the two sublattices. This remarkable property excludes permanently the intervention of the three-dimensional irreductible representation ($\Gamma_{5g} = T_{2g}$) where the magnetic modes of the rare earth sublattice are along the $<0\overline{1}$ 1> directions. Using the basis vectors of ($\Gamma_{4g} = T_{1g}$), we present in Fig. 10 the four cubic models of "*double umbrella*" of type II (Lahoubi, 1986) in the irreductible representation ($\Gamma_{4g} = T_{1g}$) of Ia $\overline{3}$ d

"Double umbrella of type II": $S_1 = +S_7 = (f, F+G, F+G); S_4 = +S_{10} = (f, F-G, F-G); c.p. (29)$

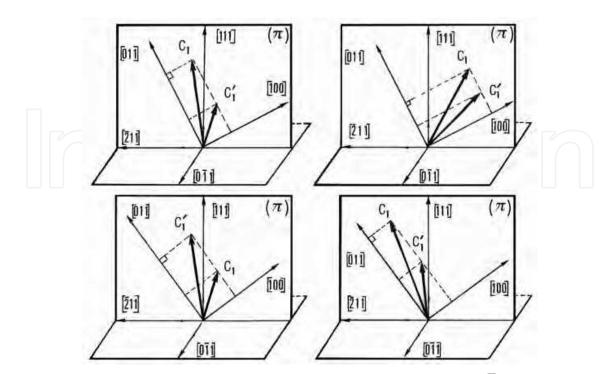


Fig. 10. Four cubic "double umbrella" models of type II for ($\Gamma_{4g} = T_{1g}$) of Ia $\overline{3}$ d.

In these four models, the angles and the modulus of the moments m_1 and m'_1 are different. They must to respect some requirements: the components of the moments m_1 , m'_1 along the cubic axis [100] are necessary equals; also for the components of the moments m_2 , m'_2 and m_3 , m'_3 respectively along [010] and [001]. The components of the moment m_1 along the cubic axes [010] and [001] must be equals; also for m_2 and m'_2 along [100] and [001]; also for m_3 and m'_3 along [100] and [010]. The first refinements at 4.2 K (with $m_a = m_d = 5 \mu_B$) lead to values of the moment m'_1 (~ 10 μ_B), i.e., above the free Tb³⁺ ion value (9 μ_B). The reason of this discrepancy is associated to the deviation of the cubic description which imply a rhombohedral distortion observed on powder sample by X-ray diffractions at 5 K (Hock et al., 1970, Sayetat, 1974, 1986 and refs. herein) and neutron diffractions at 5 K (Hock et al., 1990) where two subgroups of Ia $\overline{3}$ d have been proposed respectively: R $\overline{3}$ c and R $\overline{3}$.

5.2.2 Representation analysis of R $\overline{3}$ c

A detailed description of the representation analysis of the subgroup R $\overline{3}$ c is presented here. According to the earlier precise X-ray diffraction measurements of the rhombohedral distortions carried out on single crystals of the terbium-yttrium iron garnet system Tb_xY_{3-x}Fe₅O₁₂ with $0 \le x \le 3$ (hereafter Tb_xY_{3-x}IG) (Levitin et al., 1983), the choice of the subgroup R $\overline{3}$ c seems more appropriate. This choice will be confirmed later by our high field magnetization measurements and the use of the method of "*the symmetry lowering device*" (Bertaut, 1981) which is connected with representation analysis of Bertaut (Bertaut, 1968, 1971, 1972). From the International Tables (Hahn, 1983), the Bravais lattice of the crystallographic space group R $\overline{3}$ c-(D⁶_{3d}) No. 167 is defined in the system of the rhombohedral directions [$\overline{111}$], [$1\overline{11}$] with the three fundamental vectors: A_1 , A_2 , A_3 with the same parameter A_{Rh} and an angle $\alpha_{Rh} \# 90^\circ$ and < 120° respectively along the unit vectors g_j , j = 1-3). There are related to the cubic axes by unit vectors {*i*, *j*, *k*}

$$g_1 = \frac{1}{\sqrt{3}} (-i+j+k); g_2 = \frac{1}{\sqrt{3}} (+i-j+k); g_3 = \frac{1}{\sqrt{3}} (+i+j-k)$$
(30)

$$X = y + z; Y = x + z; Z = x + y$$
 (31)

The correspondence between the ion positions in the two space groups Ia $\overline{3}$ d and R $\overline{3}$ c is reported on Table 5. For TbIG, the values of the parameters found at 6.75 K (Sayetat, 1974, 1986) are: $A_{\rm Rh} = 10.7430$ Å; $\alpha_{\rm Rh} = 109^{\circ}24'40''$. The rhombohedral unit cell contains only the half atoms of the cubic unit cell. In order to use the representation analysis to determine the magnetic basis vectors, we choose the following generators of the space group R $\overline{3}$ c: the identity E = (1|0,0,0), the inversion I = ($\overline{1} | 0,0,0$), a ternary axis 3 = (3 | 0,0,0) and a diagonal binary axis 2_d = ($2_{X\overline{Y}} | 1/2,1/2,1/2$). This axis is perpendicular to the glide plane c = I. 2_d of the symbol R $\overline{3}$ c. The wave vector being k = 0, the six irreductible representations of the space group R $\overline{3}$ c are those of the point group D_{3d}, $\Gamma_{\rm jg}$ and $\Gamma_{\rm ju}$ with j = 1-3. The previous three-dimensional irreductible representation ($\Gamma_{4g} = T_{1g}$) of O_h is reduced to: $\Gamma_{2g} = A_{2g} + E_g$. Only the one-dimensional irreductible representation E_g being complex, she is not considered). It appears that the previous linear combinations of the spin vectors F, G, C and A are not in reality

		Ia	3 d	R3c			
Ions	Sites Symmetry		Positions	Sites Symmetry		Positions	
DES	24		±(1/8,0,1/4); c.p.	6e	2	$\pm (X,1/2-X,1/4)$ X $\approx 3/8$; c.p.	
RE ³⁺	24c	222	±(3/8,0,3/4); c.p.	6e'	2	$\pm (X,1/2-X,1/4)$ X $\approx 7/8$; c.p.	
Fe ³⁺	24d	Ī	±(3/8,0,1/4); c.p. ±(1/8,0,3/4); c.p.	12f		±(X,Y,Z); c.p. ±(Y+1/2,X+1/2,Z+1/2); c.p. X≈5/8,Y≈3/8,Z≈1/4	
Fe ³⁺	16a	3	0,0,0; 1/4,1/4,1/4 0,1/2,1/2; c.p. 1/4,3/4,3/4; c.p.	2b 6d	3 ī	0,0,0; 1/2,1/2,1/2 0,1/2,1/2; c.p. 1/2,0,0; c.p.	
O ²⁻	96h	1		12f	1	O _I O _{II} O _{III} O _{IV}	

Table 5. Correspondence between the positions of the ions in Ia $\overline{3}$ d and R $\overline{3}$ c

adapted to describe the magnetic structures of the three magnetic ions. For the RE³⁺ ions, the preceding magnetically inequivalent sublattices C_j and C'_j (j = 1-3) become crystallographic inequivalent sites 6e and 6e'; they are described in the rhombohedral axis X, Y, Z by a new linear combination for the ferromagnetic mode $f_j = S_j + S_{j+6}$ (j = 1-3). Concerning the iron ions sublattices D_j and D'_j of the 24d site, they will be associated to a new 12f site where the basis vectors are described with the ferromagnetic mode $V_j = S_j + S_{j+6}$ (j = 1-3). The associated basis vectors of ($\Gamma_{2g} = A_{2g}$) are presented here on Table 6. A new combination of these basis vectors is proposed and four modified double umbrella models are presented for the six sublattices C_j and C'_j on Table 7. In this description, the first part (fn) of the moment m_j represents the collinear ferromagnetic mode of the component along the [111] direction.

	R3c			
Sites	Basis vectors			
6e 6e'	$\Psi_{11}(I) = f_{1X} + f_{2Y} + f_{3Z}$ $\Psi_{11}(II) = f_{1Y} + f_{1Z} + f_{2Z} + f_{2X} + f_{3X} + f_{3Y}$ $\Psi_{11}(IV) = f_{4X} + f_{5Y} + f_{6Z}$ $\Psi_{11}(V) = f_{4Y} + f_{4Z} + f_{5Z} + f_{5X} + f_{6X} + f_{6Y}$			
12f	$\Psi_{11}(\mathbf{I}) = V_{1X} + V_{2Y} + V_{3Z} + V_{4X} + V_{5Y} + V_{6Z}$ $\Psi_{11}(\mathbf{II}) = V_{2X} + V_{3Y} + V_{1Z} + V_{6X} + V_{4Y} + V_{5Z}$ $\Psi_{11}(\mathbf{III}) = V_{3X} + V_{1Y} + V_{2Z} + V_{5X} + V_{6Y} + V_{4Z}$			
2b	$\Psi_{11}(\mathbf{I}) = (S_1 + S_5)_X + (S_1 + S_5)_Y + (S_1 + S_5)_Z$			
6d	$\Psi_{11}(II) = (S_2 + S_6)_X + (S_3 + S_7)_Y + (S_4 + S_8)_Z$ $\Psi_{11}(III) = (S_3 + S_8)_X + (S_4 + S_6)_Y + (S_2 + S_7)_Z$ $\Psi_{11}(IV) = (S_4 + S_7)_X + (S_2 + S_8)_Y + (S_3 + S_6)_Z$			
	6e 6e' 12f 2b			

Table 6. Basis vectors of ions in the irreductible representation A_{2g} of R $\overline{3}$ c

Four A_{2g} models in $R\overline{3}$ c with $f \neq f'$ and $a \neq a'$	New model for TbIG with $f \neq f'$ and $a' \approx a - \varepsilon$ ($\varepsilon \rightarrow 0$ at $T > 5$ K)
(6e): $C_j : S_j = S_{j+6} = fn \pm ag_j$	C ₁ : $m_1 = S_1 = S_7 = (f - a/3)n - (\frac{2\sqrt{2}}{3})ap_1$
(6e'): C'_j: S'_j = S'_{j+6} = f'n \pm a'g_j	C' ₁ : $m'_1 = S_4 = S_{10} = (f' + a'/3)n + (\frac{2\sqrt{2}}{3})a'p_1$

Table 7. Four rhombohedral models in A_{2g} of R $\overline{3}$ c and new model for TbIG (Lahoubi, 2012)

The second part $(\pm ag_i)$ represents the non collinear antiferromagnetic modes of the components of the moment m_i along the three rhombohedral axes {[111], [111], [111]}. Some requirements from the previous "cubic description" are now ignored in the "rhombohedral description": the axis [100] and equivalent directions cease to be principal axes. In the four models of non collinear arrangements for the RE3+ ions around the ternary axis [111], the sublattices C_i , C'_i are situated in the three glide planes c, c.3 and c.3² of the subgroup R $\overline{3}$ c: each plane containing [111] (unit vector *n*) and one of the rhombohedral directions {[111], [111], [111]} represented by the unit vectors g_{j} , j = 1-3. In this condition, we have: $c = (n, g_3)$; $c.3 = (n, g_3)$ g_1) and $c_{32} = (n, g_2)$ (Fig. 11a). Furthermore, the projection on the plane (111) of the rhombohedral direction [111] (g₁) for example, is equivalent to the low symmetry axis [211](with the unit vector p_1); the local axis [011] (*W*) is also chosen. If one takes into account the smallness superstructure lines (200)* and (600,442)* at 5 K (D1B), new parameters are found for C_1 (m_1 , θ_1 , ϕ_1) and C'_1 (m'_1 , θ'_1 , ϕ'_1) sites (Lahoubi, 2012) (Fig. 11b). The moments of C_1 and C'_1 are drawed in the (π) plane which corresponds to glide plane c.3: this plane remains a principal plane of the magnetic tensor \tilde{g} of the earlier "cubic description". A good reliability factor R of the order of 6.7 % is found for a refinement based on all the reflections and the method of validation of the magnetic structures (Wills, 2007 and refs. herein):

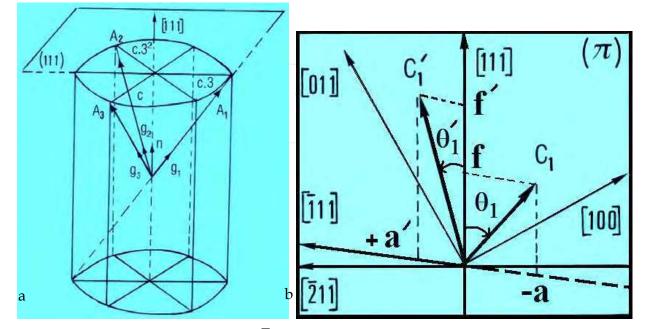


Fig. 11. a. The three glide planes of R $\overline{3}$ c, b. New model for C₁ and C'₁ at 5 K (D1B)

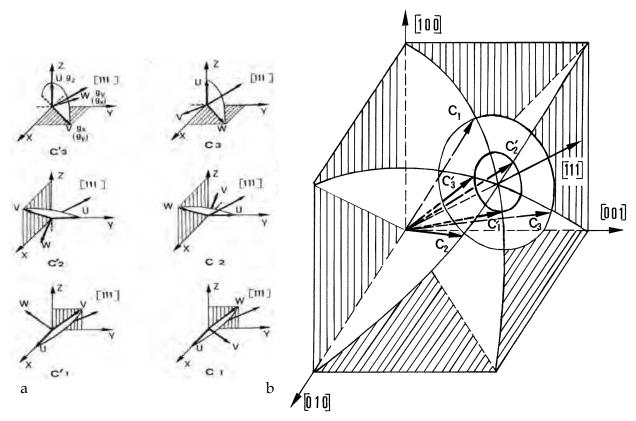


Fig. 12. a. Local axes of the D2(222) symmetry, b. Novel "double umbrella" at 5 K

$$C_1: m_1 = 8.07 \ \mu_B; \ \theta_1 = 32^\circ; \ \phi_1 = 180^\circ$$
 (32)

C'₁:
$$m'_1 = 8.90 \ \mu_B$$
; $\theta'_1 = 27^\circ$; $\phi'_1 = 0^\circ$ (33)

where, ϕ_1 and ϕ'_1 are the angles from p_1 in the (111) plane and $\varepsilon = 0.20 \ \mu_B$; with $\delta(M_S) < 111 > = M_S^{cal} - M_S^{mes} = 0.03 \ \mu_B/mol$. For a better presentation one can show the novel "*double umbrella*" (Fig. 12b) in the local axes of the D2(222) symmetry (Fig. 12a). These results are in good agreement with our recent high magnetic field magnetizations performed at 4.2 K (Lahoubi, 2012) where a third low critical field H_{c0} (Fig. 13) is observed along the <100> direction and added to the previous H_{c1} and H_{c2} (Lahoubi et al., 1984). They confirm unambiguously the presence of the three magnetic glide planes c' of the symbol R $\overline{3}$ c'. The earlier results at 5 K described in the subgroup R $\overline{3}$ (Hock et al., 1990) lead to values of the components m_{1x} and m'_{1z} above the value (9 μ_B) of the free Tb³⁺ ion.

6. Temperature evolution of the double umbrella structure

The parameters for the sites $C_1(m_1, \theta_1, \phi_1 = 180^\circ)$ and $C'_1(m'_1, \theta'_1, \phi'_1 = 0^\circ)$ are refined with the same model found at 5 K with the condition $\varepsilon \to 0$ $(m_1^{\perp} = -m'_1^{\perp})$ due to the absence of the superstructure lines (200)^{*} and (600,442)^{*} above 13 K (D1B) (Fig. 4). The magnetic moments m_a and m_d of the iron sublattices were not concerned by the refinements for all reflections of the patterns and the observed values found by N.M.R. experiments (Gonano et al., 1967) are used. Good agreement is obtained with the reliability factors *R* varying between 6 and 10 % in the 4.2–283 K temperature range.

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221

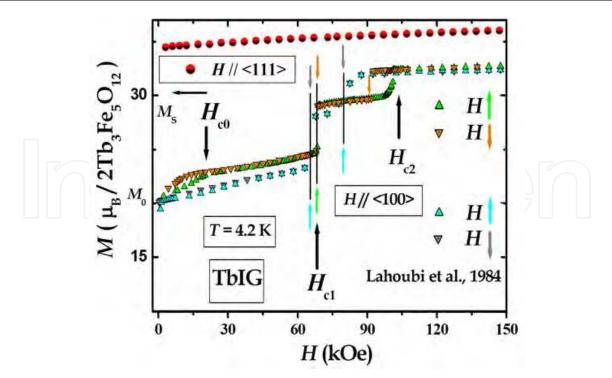


Fig. 13. $M_T(H)$ versus H at 4.2 K along the <111> and <100> directions (Lahoubi, 2012)

The equation (5) in the section 3, permit us to deduce the magnetic intensities $I_{\rm M}$ by assuming negligible the T-variations of $I_{\rm N}$ with a rather good precision for the reflections (hkl) which have a great magnetic contribution such as (211), (321), (521) and (532, 611) where the ratio $(I_M/I_N)_{cal}$ is equal respectively to: 200, 20, 4 and 2. Consequently, the high magnetically reflections with a small nuclear contribution are only (211) and (321). The thermal variations of these two magnetic reflections which are responsible of the collinear ferrimagnetic ordering are reported in Fig. 14. They appear at T_N and present a first increase below 160 K with a second rapid increase at 68 K. Different and complex temperature dependences are observed for the reflections (220) and (440). At first, the values of $(I_M)_{cal}$ are higher than those observed for $(I_M)_{obs}$. These two reflections increase simultaneously between 4.2 and 160 K and present an inflexion point near 68 K. Above 160 K, they tend to a plateau until room temperature after which they decrease progressively and reach zero at $T_{\rm N}$. The possible explanation of this characteristic behavior seems to be related to the magnetic contribution of the irons in the octahedral site [16a] to the total intensity *I*, which is not the case for the reflections (211) and (321). Another good agreement between $(I_M)_{obs}$ and $(I_M)_{cal}$ is found in the thermal variations of the superstructure lines (310)*, (110)*, (411, 330)* and (530, 433)* plotted in Fig. 15 where a rapid variation is observed at 68 K. Two distinct magnetic behaviors separated by the specific temperature 160 K are clearly evidenced.

The refined values of the parameters (m_j, m'_j) and (θ_j, θ'_j) listed on Table 8 in the 4.2–283 K temperature range lead to a good agreement between $M_{\rm S}^{\rm cal}$ and $M_{\rm S}^{\rm mes}$. During the refinement at 109 ± 2 K, two different results related to the set {a, b} are found and lead to identical values of $M_{\rm S}^{\rm cal}$ and reliability factor *R*. The same feature is observed for the set {c, d} at 127 ± 5 K. We observe in the thermal variations of the parameters (m_j, m'_j) and (θ_j, θ'_j) plotted in Fig. 16 a broad variation between 54 K and 80 K which disappears beyond 160 K.

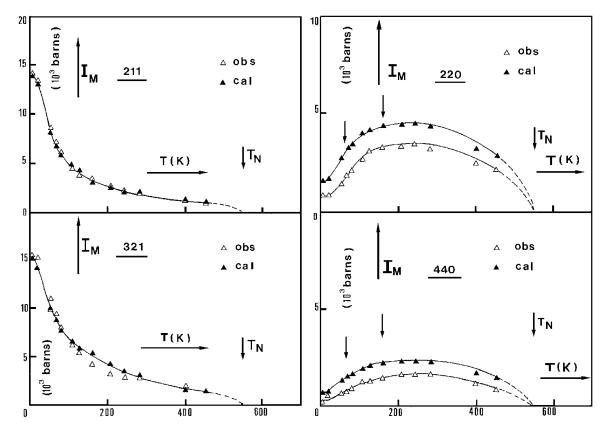


Fig. 14. Thermal variations of $(I_M)_{obs}$ and $(I_M)_{cal}$ for the reflections (211), (321), (220) and (440)

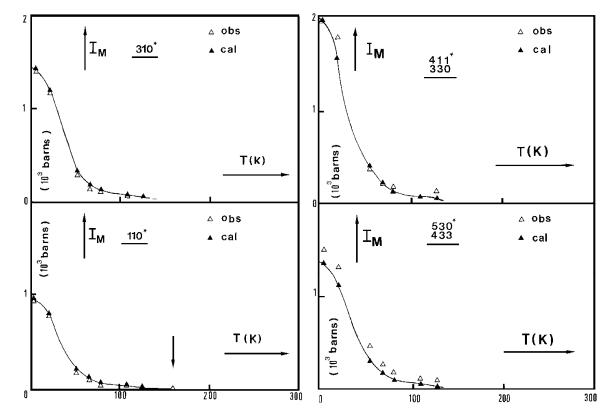


Fig. 15. Thermal variations of $(I_M)_{obs}$ and $(I_M)_{cal}$ for the superstructure lines (310)^{*}, (110)^{*}, (411, 330)^{*} and (530, 433)^{*}

	Sites C_1 ; C'_1		$M_{ m S}^{ m cal}$		M _S ^{mes} (µ _B /mol			₃ /mol)
<i>T</i> (K)	$m_1(\mu_{ m B}) \ m'_1(\mu_{ m B})$	$ heta_1(^\circ) \\ heta'_1(^\circ) heta'$	[111]	[110]	[100]	[111]	[110]	[100]
4.2	8.18 8.90	30.79 28.07	34.64	28.28	20.00	34.53	28.35	20.17
5	8.07 8.99	32.00 27.00	34.56	28.22	19.95	"	"	"
20	7.49 8.77	30.11 25.39	33.21	27.12	19.18	33.34	27.21	19.56
54	4.79 6.27	22.57 17.03	21.32	17.41	12.31	21.24	17.24	12.80
68	4.23 5.21	19.57 15.76	17.10	13.97	9.87	17.01	13.44	10.05
80	3.57 4.72	16.72 12.56	14.25	11.63	8.23	13.97	11.70	8.38
109 ± 2	2.76 a 3.79 2.78 b 3.78	15.04 10.86 15.23 11.20	9.48	7.74	5.47	8.78	7.20	5.26
127 ± 5	2.59 c 3.12 2.71 d 3.00	14.94 12.30 14.52 13.08	7.13	5.82	4.12	6.52	5.40	3.88
160	2.30 2.31	0.11 0.00	4.65	3.80	2.68	4.61	3.76	2.66
208 ± 2	1.74(3) 1.75(7)	0.05 0.00	1.92	1.56	1.11	1.97	1.36	1.12
244 ± 10	1.34(4) 1.35(8)	0.08 0.00	0	0	0	0	0	0
283	1.15 1.15	0 0	0.66	0.54	0.38	0.79	0.59	0.63

Table 8. Values of the parameters m_j , m'_j , θ_j and θ'_j in the 4.2–283 K temperature range with a comparison between the calculated $M_{\rm S}^{\rm cal}$ and measured $M_{\rm S}^{\rm mes}$ magnetizations.

The thermal variations of the parallel $(m_{!}/, m'_{!}/)$ and perpendicular $(m_{j} \perp, m'_{j} \perp)$ components are also reported in Fig. 17. The double umbrella magnetic structure appears to close slowly around the <111> direction in the three magnetic glide planes c' near 160 K with an abrupt increase between 54 and 68 K. Previous temperature dependences of the calculated non collinear magnetic structure in TbIG (Druzhinina & Shkarubskii, 1988) and the recent neutron scattering on TbIG single crystal, (Louca et al., 2009) are not consistent with these thermal variations. In another recent study of the magnetic and magneto-optical properties of the Tb³⁺ ions in TbIG and in the mixed system of terbium-yttrium ferrites garnets Tb_xY₃₋ _xIG (x < 1) (Zhang et al., 2009) the differences between the two non collinear magnetic structures which exist at low temperature were not taken account in their calculations.

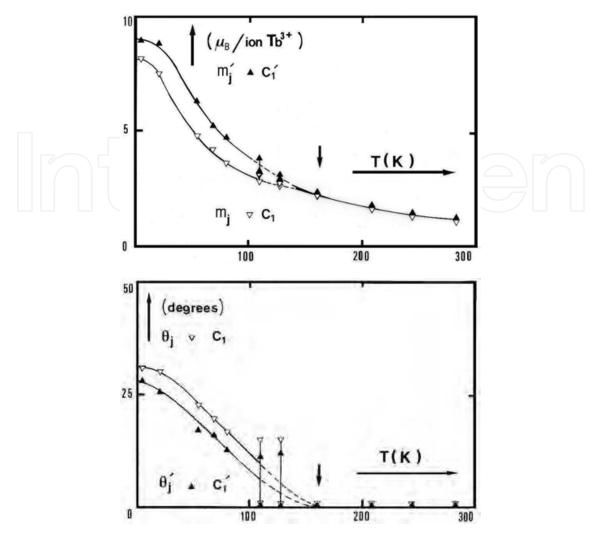


Fig. 16. Thermal variations of the parameters (m_j, m'_j) and (θ_j, θ'_j)

The <111> direction ceases to be the easy axis of magnetization and changes below 140 K to the <100> direction up to 4.2 K in $Tb_{0.37}Y_{2.63}IG$ for example with the appearance of the low symmetry phases <uuw> (Lahoubi et al, 2000) in the spontaneous spin reorientation phase transitions.

These results are in good agreement with the previous observed rhombohedral distortion below 190 K (Rodić & Guillot, 1990) and 200 K (Sayetat, 1974, 1986). They are also in good agreement with the anomalous behaviors observed previously below 200 K on TbIG single crystals without applied magnetic fields, in the acoustic properties (Kvashnina et al., 1984; Smokotin et al., 1985) and in the elastic constant measurements (Alberts et al., 1988) along the [100], [110] and [111] crystallographic directions, precisely in the temperature ranges, 60–140 K and 50–165 K respectively.

It seems that the behavior around 160 K has a relation with the previous predicted momentum angular compensation point T_J (Nelson & Mayer, 1971) localized at 150 and 190 K by assuming the free and quenched ion value respectively.

The large magnetodielectric (MD) effects which have been recently revealed on TbIG single crystal at low temperature as well when a very small external magnetic field ($H_{ex} < 0.2$ T) is

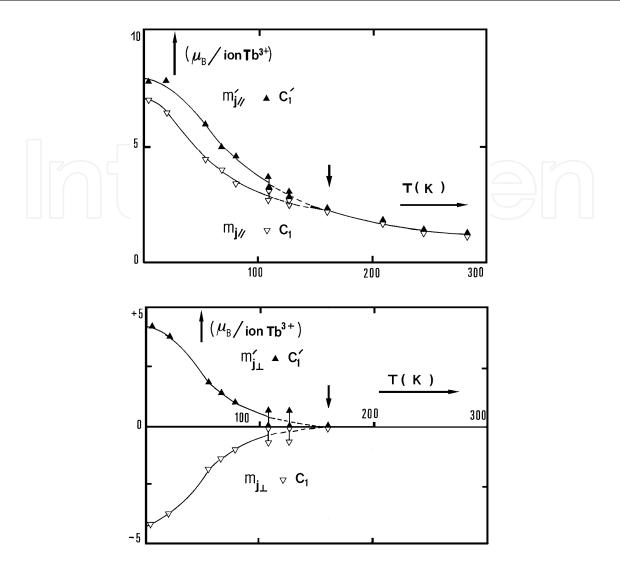


Fig. 17. Thermal variations of the parameters $(m_i/, m_i^{\perp})$ and $(m'_i/, m'_i^{\perp})$

applied (Hur et al., 2005) could be combined with the previous huge spontaneous magnetostriction measurements (Sayetat, 1974, 1986; Guillot et al., 1980) where a peak near 70 K has been observed in the thermal variations of $\lambda^{\epsilon, 2}$ (Tb³⁺)_{exp}/ $\lambda^{\epsilon, 2}$ (Tb³⁺)_{cal}, the ratio of the experimental values of the magnetostriction constant to the theoretical values derived from the one ion model. It correspond to the abrupt change in the long-range magnetic order in the Tb³⁺ sublattice near the previous predicted low-temperature point $T_B = 58$ K (Belov, 1996 and refs. herein) which is situated between 54 and 68 K in this study. More recently, some magnetoelectric (ME) and MD effects in weak and high external magnetic fields (H_{ex} up to 2 T and H_{ex} up to 10 T respectively) have been reported (Kang et al., 2010). A possible coupling between the magnetic exchange and the ligand-field excitations which occurs at a specific temperature situated between 60 and 80 K has been discovered without external magnetic field ($H_{ex} = 0$) with two distinct behaviors above and below another characteristic temperature (~ 150 K). All results confirm that Landau's theory of second order phase transitions does not apply to TbIG in the 5 K- T_N temperature range without applied external magnetic field and the magnetic space group is R $\overline{3}$ c' (Bertaut, 1997; Lahoubi et al., 1997).

7. Conclusion

In this chapter, the temperature evolution of the magnetic structure in TbIG is studied by neutron diffraction experiments below T_N (568 K). The "double umbrella" structure observed at 5 K appears below a specific temperature (~ 160 K) which could be related to the previous predicted T_J -point situated in the 150–190 K temperature range. The rapid variation of the Tb³⁺ moments is observed between 54 and 68 K where the predicted value (58 K) of the T_B -point is located. The magnetic symmetry doesn't change with the temperature and the magnetic space group is R $\overline{3}$ c'. The author is convinced that the symmetry considerations of the Representation Analysis of Bertaut presented in this chapter have demonstrated their usefulness in the determination of the thermal variation of the double umbrella magnetic structure in TbIG below T_N . It is hoped that these results which are in good agreement with the magnetization measurements will facilitate a better understanding of the possible correlations between the magnetic properties via the double umbrella magnetic structure and the recent ME and MD effects found in this ferrite garnet.

8. Acknowledgements

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9. References

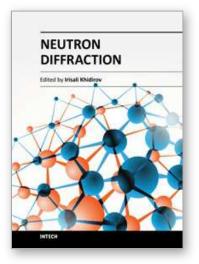
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Now neutron diffraction is widely applied for the research of crystal, magnetic structure and internal stress of crystalline materials of various classes, including nanocrystalls. In the present book, we make practically short excursion to modern state of neutron diffraction researches of crystal materials of various classes. The book contains a helpful information on a modern state of neutron diffraction researches of crystal materials for the broad specialists interested in studying crystals and purposeful regulation of their service characteristics, since the crystal structure, basically, defines their physical and mechanical properties. Some chapters of the book have methodical character that can be useful to scientists, interested in possibilities of neutron diffraction. We hope, that results of last years presented in the book, can be a push to new ideas in studying of crystalline, magnetic structure and a macrostructure of usual crystal materials and nanocrystals. In turn, it can promote working out of new materials with new improved service characteristics and to origin of innovative ideas.

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