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# Carbon-Based Nanocomposite Materials for High-Performance Supercapacitors

*Prasanta Kumar Sahoo, Chi-Ang Tseng, Yi-June Huang  
and Chuan-Pei Lee*

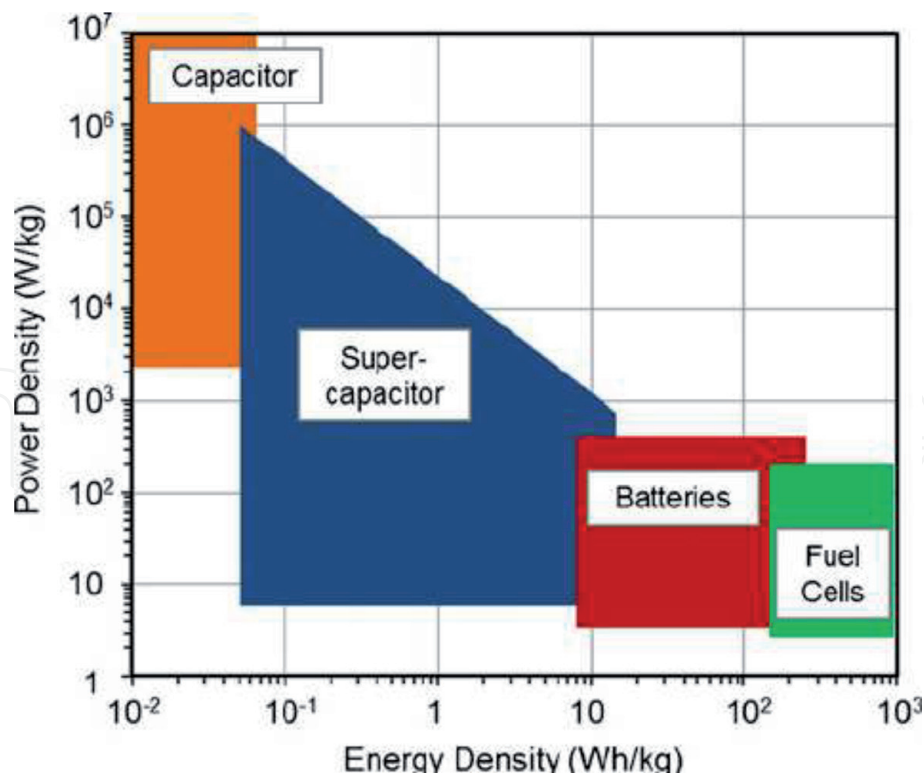
## Abstract

Lightweight, flexible, wearable, and portable electronic gadgets have drawn significant attention in modern electronics industry. To power these gadgets, great efforts have been made to develop highly efficient energy-storage equipment. Among various power sources, a supercapacitor, acting as a bridge between the conventional battery and electrolytic capacitor, has been considered a promising portable energy storage device because of its high power density, fast charge/discharge rate, adequate operational safety, and excellent working lifetime. Hybrid supercapacitors, which combine redox materials with carbon-based materials, exhibit tremendous potential to fulfill the requirement of practical applications. In this chapter, we will review recent reports focusing on composite materials (*i.e.* metal oxide, metal hydroxide, and metal dichalcogenide composited with carbon materials) for the application in supercapacitors. The conclusion and futuristic prospects and challenges of highly efficient supercapacitors are briefly discussed.

**Keywords:** energy storage, composites, metal oxides, metal hydroxides, transition metal dichalcogenides, supercapacitor

## 1. Introduction

There is sharply increasing demand for energy with the rapid growth of the global economy. The energy generation from sustainable sources, such as wind and solar, plays an important role in power supply. However, the intermittent nature and imbalanced regional distribution of the sustainable energy make them unable to stably supply the power [1]. The development of energy storage systems is an urgent requirement to meet the sufficient and stable power supply for industrial and residential usage. Although rechargeable lithium-ion batteries, dominant energy sources in each field, as high energy density providers have filled their position [2], lithium-ion batteries still have the limitations of poor cycle life and low power performance [3]. Supercapacitors (SCs), also known as ultracapacitor and electrochemical capacitors, are an emerging class of energy storage device, which possess high power density and tens of thousands of charge/discharge cycles [4, 5]. **Figure 1** shows the Ragone plot of different energy conversion and storage devices. SCs have a unique position to bridge the gap between conventional capacitors and batteries. Compared with conventional capacitor, SCs possess higher specific energy density



**Figure 1.**  
*Ragone plot for various energy storage and conversion devices [6].*

in several orders of magnitude. Moreover, SCs provide higher specific power density than batteries due to its unique charge storage mechanism.

## 2. Theoretical background for supercapacitors

### 2.1 Principle and mechanism of supercapacitors

Based on different charge storage mechanisms, SCs are mainly divided into two categories, electrical double layer capacitors (EDLCs) and pseudocapacitors, as shown in **Figure 2**. EDLCs store the electrical charge by electrostatic force at the electrode-electrolyte interface, which is a physical process without involving electrochemical reactions on the electrode surface. In order to increase the capacitance and energy density of SCs, some electrochemically active materials, such as transition metal oxide and conducting polymers, have been explored as electrode materials for pseudocapacitors. The energy storage in pseudocapacitors originates from reversible surface faradaic redox reactions at the interface of electrolyte and electroactive materials.

### 2.2 Factors affecting the performance of Supercapacitors

The capacitance of EDLCs is strongly dependent on effective surface area and the pore size distribution of the electrode [7, 8]. Typically, the carbon-based materials and their derivatives, including activated carbon, carbon nanotubes (CNTs) and graphene, with high conductivity, chemically-stability, and large surface area are widely utilized in EDLCs. Although the EDLCs possess high power density and excellent charge/discharge cycling stability, they suffer from low energy density owing to the relatively low capacitance of carbon-based materials. Pseudocapacitors

can achieve significantly higher energy density, as compared to EDLCs, because they have a variety of oxidation states for redox charge transfer reactions. However, relatively low electrical conductivity and poor rate capability and cycle stability of pseudocapacitive materials limit their widespread commercial applications [9]. Therefore, carbon-based materials with high conductivity and distinct structures can be combined with pseudocapacitive materials to exhibit synergistic effects for supercapacitive performance, known as hybrid SCs.

### 3. Carbon based composite electrode materials

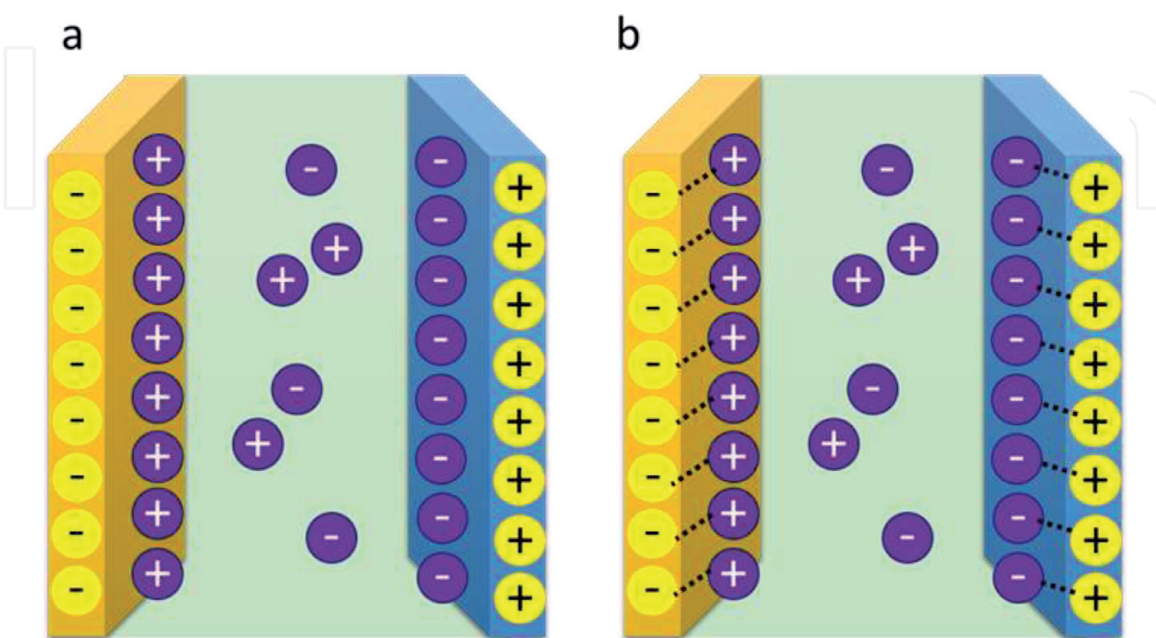
Carbon material is EDLCs type for supercapacitor. In section 2.1, EDLCs has introduced their property, which store the electrical charge by electrostatic force at the electrode-electrolyte interface, as shown in **Figure 2**. It is not involving electrochemical reactions on the electrode surface. There are different types of carbon nanostructured materials, which can be used as single electrode materials due to their unique structural, mechanical, and electrical properties.

#### 3.1 Zero-dimensional (0-D) carbon nanoparticles


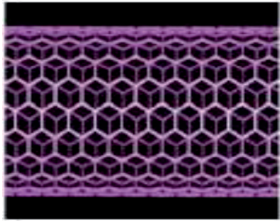
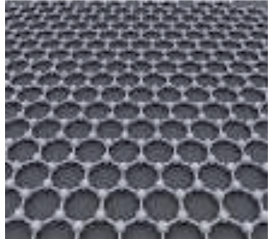

They are round-shaped particles such as ultrafine activated carbon (AC), mesoporous carbon, carbon nanosphere, and carbon quantum dot, with a high specific area (AC:  $\sim 3000 \text{ m}^2 \text{ g}^{-1}$ ) and an aspect ratio of nearly [10]. In addition, by tuning the pore size distribution and pore content, they can use as suitable supporting materials for composite electrodes.

#### 3.2 One-dimensional (1-D) carbon nanostructures

These are the high aspect ratio materials with fiber shaped and good electronic properties e.g. carbon nanotubes (CNT), carbon nanocoils, and carbon nanofibers (CNF), which facilitates the electrochemical reaction kinetics by 1-D charge transfer pathway.



**Figure 2.**  
 Schematic diagram of (a) an electrical double layer capacitor and (b) a pseudocapacitor.

Materials	Carbon onions	Carbon nanotubes	Graphene	Templated carbon
Dimensionality	0D	1D	2D	3D
Conductivity	High	High	High	Low
Volumetric Capacitance	Low	Low	Moderate	Low
Cost	High	High	Moderate	High
Structure				

**Table 1.**  
*Different carbon nanostructures used as electrode materials for EDLCs with onion-like carbon, carbon nanotubes, graphene, activated carbon, carbide-derived carbon, and templated carbon [16].*



### 3.3 Two-dimensional (2-D) nanosheets

They are sheet like structures with high aspect ratio such as graphene, graphene oxide (GO) or reduced graphene oxide (rGO). In addition, they have high specific surface area, good mechanical strength, and excellent electrotonic conductivity, which helps them as promising electrode materials for SCs. For an example, single layered graphene has theoretical surface area of  $2756 \text{ m}^2 \text{ g}^{-1}$  and charge mobility of  $200000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  [11].

### 3.4 Three-dimensional (3-D) porous nanostructures

These are the low dimensional building blocks such as carbon nanofoams or sponges with hierarchical porous channels, rich pore structures, higher electrical conductivity and better structural mechanical stability, which are extensively used in composite electrode materials for SCs. For an example, foam has high specific surface area with continuous electron transport path and large area of electrolyte-electrode interface.

**Table 1** shows some examples of different carbon nanostructured materials such as carbon onions, carbon nanotubes, graphene, and templated carbon, which are used as electrode materials for EDLCs. Each carbon nanostructured materials have its advantages and disadvantages. For example, carbon onions have high power performance due to excellent conductivity with high accessible ion adsorption capacity but low capacitance of  $\sim 30 \text{ F g}^{-1}$  [12]. On the other hand, CNTs have high energy density due to superior electrical properties and unique tubular structures for fast charge transportation but due to the high cost, their widespread applications are limited [13]. Recently, graphene has been attracted much attention as electrode materials for EDLC applications due to unique properties, like as ultrahigh specific surface area, unique conductivity, and exceptionally high mechanical strength [14]. However, the aggregation of sheets during electrode preparation limits the aspect of application. More recently, 3D porous carbon nanostructured materials are widely used for EDLCs because of rich pore structures and high surface areas but due to relative low conductivity and presence of micropores specific capacitance is insufficient at a high current density [15]. Therefore, it is necessary to construct composite materials by coupling the advantages of different types of carbon nanostructured materials and high energy electrode materials such as transition metal oxides, metal hydroxides and metal dichalcogenides (TMDs) to enhance the energy density without the compromise of power density and also meet the requirement for fabrication of high energy storage devices. In the composite electrode material, different types of carbon nanostructured materials not only contribute to high capacitance but also provide an easy conductive path for charge transportation due to conductive nature.

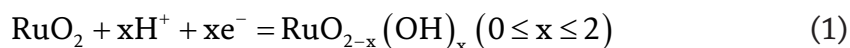
## 4. Carbon-metal oxide composite electrode materials

Many metal oxide such as  $\text{RuO}_2$ ,  $\text{MnO}_2$ ,  $\text{Fe}_3\text{O}_4$ ,  $\text{V}_2\text{O}_5$ ,  $\text{NiO}$ ,  $\text{Co}_3\text{O}_4$ , and  $\text{TiO}_2$ , has been received significant attention and extensive studied as SC electrode materials due to Pseudo capacitance nature, which depends on the fast reversible redox reaction of electroactive species directly as well as in the vicinity of electrode surface [17–20]. The redox behavior is due to the multivalent property of the above oxides which changes their oxidation states by interaction with protons or hydroxide ions reversibly. In spite of their excellent specific capacitance, they still suffer from low conductivity, low rate capability, poor stability and durability during the process of charge/discharge. In contrast carbon materials shows excellent performance in

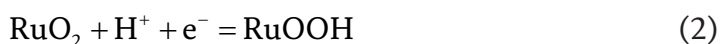
these regards but suffer from comparatively limited specific capacitance. Hence, the synergic integration of metal oxides with conducting carbon supports may form high potential carbon-metal oxide composite electrodes materials for SCs and hybrid devices because of their enhanced electrochemical performance through the combined effect of pseudocapacitive/faradaic charge storage and electrical double layer capacitance mechanisms [21–23].

#### 4.1 Carbon-ruthenium oxide (RuO<sub>2</sub>)-based composite electrode materials

Among the metal oxides, ruthenium oxide (RuO<sub>2</sub>) has been considered as very common electrode materials for SCs in acidic medium due to their excellent pseudocapacity which is arising from high conductivity, good thermal stability, highly reversible redox reactions, three different oxidation states within 1.2 V, and high specific capacitance natures. The pseudocapacitance mechanism of RuO<sub>2</sub> for SC electrodes can be described as equation [24]:

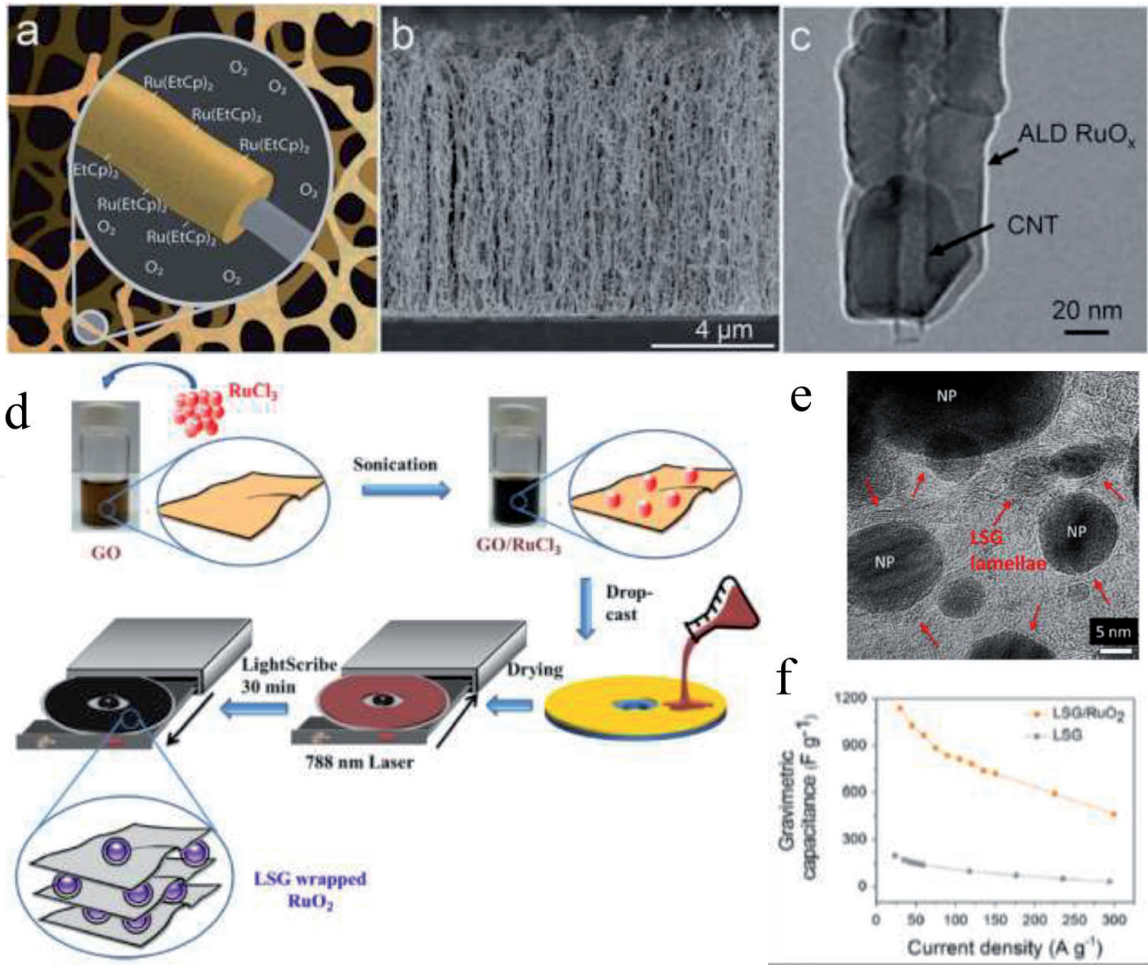


Or



However, its scarcity and high cost limits the fabrication of RuO<sub>2</sub> based electrodes for potential applications. But, smartly use of composite materials by synergic integration of pseudocapacitive RuO<sub>2</sub> materials with conductive carbonaceous substrates not only improves the capacitance but also reduces the cost of the electrode. Recent studies are more focus about the selecting the best carbonaceous substrate and the synthesis procedures to fabricate ruthenium oxide (RuO<sub>2</sub>)-coated on the porous carbonaceous substrates.

RuO<sub>2</sub>-CNT composite has been prepared by uniformly coating of RuO<sub>2</sub> on the vertically aligned porous carbon nanotubes porous through atomic layer deposition (ALD) technique and further activation by voltammetry potential coulometry (**Figure 3(a-c)**) [25]. This ALD technique has many advantages such as deposition on large surface area, accurate thickness and exceptional uniformity for electrode designing in energy storage devises. The as-prepared RuO<sub>2</sub>-CNT composite shows excellent electrochemical performance as an electrode material for SC in respect of capacitance, power density and stability. Several publications have been reported the specific capacitance and power density of RuO<sub>2</sub>-CNT composite, which are around 650 F g<sup>-1</sup> and 17 kW kg<sup>-1</sup>, respectively. Kaner *et.al* recently demonstrated the synthesis and processing of 3D porous RuO<sub>2</sub>/laser-scribed graphene (LSG) composite electrode for miniaturized and interdigitated SC that exhibit ultrahigh energy and power density (**Figure 3(d)**) [26]. The high-resolution TEM (HRTEM) image of 3D porous RuO<sub>2</sub>/LSG composite in **Figure 3(e)** shows that multiple layers of the graphene sheets wrap around each RuO<sub>2</sub> nanoparticle. 3D porous RuO<sub>2</sub>/LSG composite electrode showed an ultrahigh specific capacitance of 1139 Fg<sup>-1</sup> with outstanding rate capability and the asymmetric supercapacitor (ASC) made of 3D porous RuO<sub>2</sub>/LSG composite electrode as positive electrode exhibited an extremely high energy density of 55 W h kg<sup>-1</sup> at a power density of 12 kW kg<sup>-1</sup> (**Figure 3(f)**). Other interesting composite of RuO<sub>2</sub> made of RuO<sub>2</sub> decorated nitrogen-doped reduced graphene oxide aerogel (NGA) are used as high-performance transparent



**Figure 3**  
(a) Schematic presentation of RuO<sub>x</sub> deposited on the vertically aligned porous carbon nanotubes porous through ALD by sequential pulsing of Ru(EtCp)<sub>2</sub> and oxygen. (b) and (c) SEM and TEM images of vertically aligned CNTs coated with ALD RuO<sub>x</sub> [25]. (d) Microfabrication process of 3D porous RuO<sub>2</sub>/LSG interdigitated micro-supercapacitors through direct laser writing on a DVD disc using a LightScribe DVD burner. (e) A high-magnification TEM image of 3D porous RuO<sub>2</sub>/LSG composite showing complete wrapping of the RuO<sub>2</sub> nanoparticles (NP) by multiple layers of the graphene sheets. (f) The gravimetric capacitance retention of laser scribed graphene (LSG) and RuO<sub>2</sub>/LSG electrodes as a function of the applied current density [26].

solid-state supercapacitors. RuO<sub>2</sub>/NGA composite with finely tuned mass loading of 16.3 μg cm<sup>-2</sup> and transmittance of 34.1% (λ = 550 nm) demonstrated maximum areal energy of 0.074 μW h cm<sup>-2</sup> and power of 64 μW cm<sup>-2</sup> with cyclic stability of 100% over 2000 cycles [27]. This RuO<sub>2</sub>/NGA based high transparent SC can be practically used in many advanced transparent electrical devices.

#### 4.2 Carbon-manganese oxides (MnO<sub>2</sub>)-based composite electrode materials

MnO<sub>2</sub> has been considered as a promising pseudocapacitive electrode materials for energy storage applications due to low price, abundant reserve, high specific capacitance, and environmental environment benign nature and low toxicity in comparison to other transition-metal oxides. In general, the charge storage mechanism of MnO<sub>2</sub> involves change in manganese oxidation state from +3 to +4 and the contribution of protons or alkali cations, which can be shown in the following equation [28].

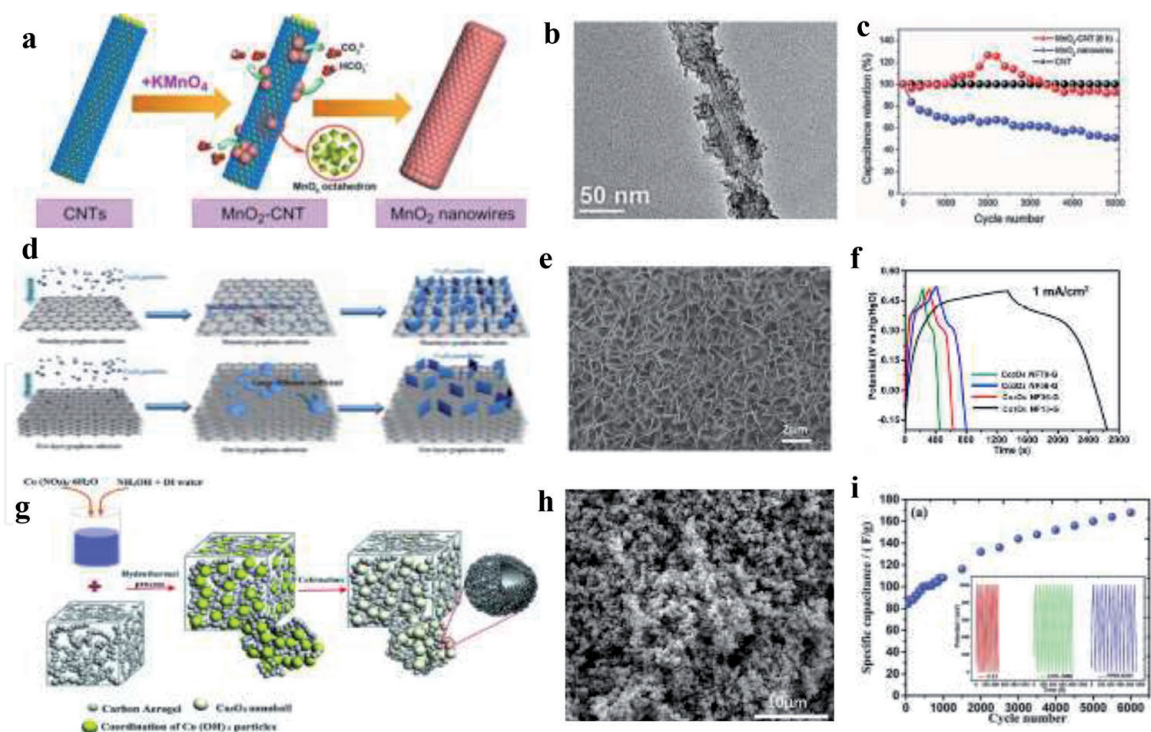




Where  $C^+$  represents protons or alkali cations ( $Li^+$ ,  $Na^+$ ,  $K^+$ ).

However,  $MnO_2$  based electrodes limits the capacity and power density due to their low surface area and poor electronic/ionic conductivity. Therefore, the composite of  $MnO_2$  with high-surface area and conducting carbonaceous materials may improve the electrochemical performance in terms of specific capacity, energy and power densities by providing the larger interfacial area between the  $MnO_2$  particles and the electrolyte solution [29].

Gao *et al.* fabricated a  $MnO_2$ /activated carbon (AC) based hybrid SC, where AC not only acted as a conducting support but also increase the capacitance as well as energy and power densities [30]. In addition, engineering the morphology of  $MnO_2$  into different nanostructures is considered to be a practical approach to increase its electrochemical performance. It is reported that the pore sizes of the mesoporous- $MnO_2$ /AC are greatly affected the specific capacitance and the rate capability of the SCs. Huang *et al.* demonstrated the influence of CNT on the electrochemical properties of  $MnO_2$ -CNT composite electrode by controlling the growth of  $MnO_2$  nanostructures on CNTs through a facile redox approach (Figure 4(a-c)) [31]. The as-prepared  $MnO_2$ -CNT composite electrode showed a maximum specific capacitance of  $247.9 \text{ F g}^{-1}$  with outstanding cyclic stability of 92.8% after 5000 cycles. In addition, it has been noticed that the aligned CNTs are more favoured as SC electrodes over nonaligned CNTs due to their large specific surface area, low contact resistance, and fast electron-transfer kinetics. Graphene is being used as a supporting material for  $MnO_2$  nanostructures due to its large surface area, high conductivity, and high stability nature. For example, microwave



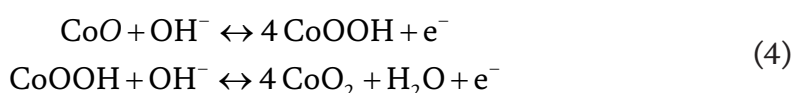
**Figure 4.** (a) Controlled growth of  $MnO_2$  nanostructured on CNT surface through facile redox method. (b) TEM images displaying coverage of  $MnO_2$  on the surface of CNT. (c) The cyclic curve of a  $MnO_2$ -CNT nanowire composite at current density of  $2 \text{ A g}^{-1}$  [31]. (d) Schematic illustration of fabrication of  $Co_3O_4$  nanoflake/graphene@Ni hybrid electrode materials by in situ synthesis method. (e) Top-view SEM images of the  $Co_3O_4$  nanoflake/graphene/Ni hybrid electrode. (f) GCD curves of  $Co_3O_4$  nanoflake/graphene/Ni hybrid electrode at current density of  $1 \text{ mA cm}^{-2}$  [39]. (g) A schematic of the synthesis of the porous  $Co_3O_4$  nanoball/CA hybrid. (h) FE-SEM images of the porous  $Co_3O_4$  nanoball/CA hybrid. (i) The specific capacitance test of the porous  $Co_3O_4$  nanoball/CA hybrid electrode at a current density of  $1 \text{ A g}^{-1}$  as a function of cycle number (inset: 11 cycles continuous GCD curves obtained for porous  $Co_3O_4$  nanoball/CA hybrid electrode for the different cycle numbers) [41].

irradiation synthesised MnO<sub>2</sub>-graphene composites exhibited the maximum capacitance of 310 F g<sup>-1</sup>, which is much higher than the bare graphene and MnO<sub>2</sub> (110 F g<sup>-1</sup>) [32]. Beside their high capacitance, MnO<sub>2</sub>-graphene composites have better cyclic stability of 95% over 15000 cycles. The excellent electrochemical performance of MnO<sub>2</sub>-graphene composites is due to large surface area and high conductivity of graphene network. Recently, Zhang *et al.* reported highly flexible ASCs based on graphene hydrogel (GH)/copper wire (CW) as the negative electrode and hierarchical MnO<sub>2</sub>/graphene/carbon fiber (CF) as the positive electrode, which exhibited excellent areal energy density of 18.1 μW h cm<sup>-2</sup> and operated reversibly at potential window of 0-1.6 V [33]. 3D porous carbon nanostructures can also be used as MnO<sub>2</sub> support for supercapacitor (SC) electrodes as they provided large surface area, well-defined pathways to electrolyte access, and better mechanical stability. Fang *et al.* demonstrated a novel solid-state symmetric supercapacitor (SSC) based on 3D rGO@MnO<sub>2</sub> foam electrode and Polyacrylic Acid (PAA)-Portland cement-KOH electrolyte, which showed a very high areal capacity of 1.84 F cm<sup>-2</sup> at current density of 0.5 mA cm<sup>-2</sup> and excellent capacitance retention of 61% at a current density of 40 mA cm<sup>-2</sup> [34].

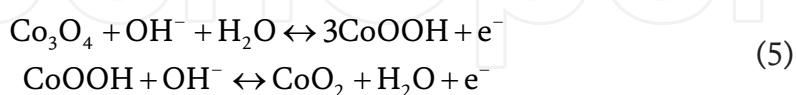
#### 4.3 Carbon-cobalt oxides (CoO/Co<sub>3</sub>O<sub>4</sub>)-based composite electrode materials

Cobalt oxides has been received considerable attention as highly promising SC electrode materials due to their non-toxic, low cost, easy synthesis, environmentally friendly, and more importantly high theoretical capacitance (CoO: 4292 F g<sup>-1</sup>, Co<sub>3</sub>O<sub>4</sub>: 3560 F g<sup>-1</sup>) [35]. In addition, cobalt oxides exhibits outstanding electrochemical behaviour in alkaline as well as organic electrolyte, which is possible due to their ability to interact with the ions at the electrolyte surface as well as through the bulk of the material. The pseudocapacitance of cobalt oxides (CoO/Co<sub>3</sub>O<sub>4</sub>) are originates from the following redox reaction: [36]

CoO:



Co<sub>3</sub>O<sub>4</sub>:



However, the low electrical/ionic conductivity of cobalt oxides hinders their practical performance as SC electrodes. Most efficient way to improve their electrochemical performance is to form composites of cobalt oxides by incorporation into a carbon-based conducting supports. A Co<sub>3</sub>O<sub>4</sub>/AC composite SC electrode was reported by Iqbal *et al.* [37]. The electrode exhibited maximum achievable specific capacitance 567 F g<sup>-1</sup> and maximum energy density of 63 W h kg<sup>-1</sup> at 0.7 A g<sup>-1</sup>. In addition to the high specific capacitance, Co<sub>3</sub>O<sub>4</sub>/AC composite of capacitive retentivity is 82% after 6000 charge/discharge cycles and safe to handle due to no leakage. The specific capacitance of the cobalt oxide strongly depends on the microstructure and morphology of the materials, which facilitate the electrolyte ion transport through the material more effectively. Sun *et al.* demonstrated a

simple and effective approach to grow well-aligned 3D cobalt oxide nanowire arrays (Co<sub>3</sub>O<sub>4</sub> NWAs) directly on carbon nanotube fibers (CNTFs) through CVD process [38]. The Co<sub>3</sub>O<sub>4</sub> NWAs/CNFs showed a specific capacitance of 734.25 F cm<sup>-3</sup> (2210 mF cm<sup>-2</sup>) at 1.0 A cm<sup>-3</sup> and a high energy density of 13.2 mW h cm<sup>-3</sup> at a current density of 1.0 A cm<sup>-3</sup>. Graphene along with cobalt oxides can be used as a composite material for SCs because of its high conductivity, high surface area, high carrier mobility, and excellent mechanical strength. For example, an in situ synthesised Co<sub>3</sub>O<sub>4</sub>/graphene@NF hybrid composite electrode with a thickness of 13 nm exhibited a high specific capacitance of 1.75 F cm<sup>-2</sup> at 1 mA cm<sup>-2</sup> and a capacitance increase of 12.2% after 5000 cycles at 10 mA cm<sup>-2</sup> (**Figure 4(d-f)**) [39]. Tseng *et al.* demonstrate a binder-free and flexible SC based on CoO/graphene hollow nanoballs (GHBs) composite electrode [40]. The as fabricated CoO/GHBs composite electrode exhibits high specific capacitance of 2238 F g<sup>-1</sup> at a current density of 1 A g<sup>-1</sup> and good rate capability of 1170 F g<sup>-1</sup> at a current density of 15 A g<sup>-1</sup>. The excellent capacitive performance and high rate capability were accomplished by the synergistic combination of conductive GHBs with large surface areas and highly pseudocapacitive CoO. In addition, as fabricated SSC demonstrated a very high power density (6000 W kg<sup>-1</sup> at 8.2 W h kg<sup>-1</sup>), high energy density (16 W h kg<sup>-1</sup> at 800 W kg<sup>-1</sup>), good cycling stability (~100% capacitance retention after 5000 cycles), and excellent mechanical flexibility at various bending positions. Recently, 3D-carbon aerogels (3D-CA) with appropriate electrical conductivity, high specific surface area and rich dielectric electrochemical stability when combined with the porous cobalt oxides can enabled the fabrication of an composite electrode with outstanding electrochemical performance. Co<sub>3</sub>O<sub>4</sub>/CA composite electrode which was synthesized through in situ growth method showed a specific capacitance of 350 F g<sup>-1</sup> at 1 A g<sup>-1</sup> and Energy density of 23.82 kW kg<sup>-1</sup> at a power density of 95.96 W kg<sup>-1</sup> (**Figure 4(g-i)**) [41]. The as-prepared ASC device could be cycled reversibly in a potential range of 0.0 to 1 V at 1 A g<sup>-1</sup> and showed a capacity retention of 210% over 6000 cycles. Zhu *et al.* adopted a facile hydrothermal method to synthesize self-assembled cobalt oxide (CoO) nanorod cluster on 3D-graphene foam (CoO-3DGF) which exhibits a very high performance compared with CoO nanorod clusters grown on Ni foam (680 F g<sup>-1</sup>) in terms of specific capacitance 980 F g<sup>-1</sup> at 1 A g<sup>-1</sup> and cycling stability of 103% over 10,000 cycles [42].

#### 4.4 Carbon-binary metal oxide based composite electrode materials

Recently, binary metal oxides such as NiCo<sub>2</sub>O<sub>4</sub>, NiFe<sub>2</sub>O<sub>4</sub>, CoFe<sub>2</sub>O<sub>4</sub>, ZnMnO<sub>4</sub>, and ZnCo<sub>2</sub>O<sub>4</sub> have attracted much attention due to higher electrical conductivity than individual metal oxide and provide higher capacitance due to more affluent redox reaction than individual components [43]. Even though binary metal oxides possess better electrochemical performance than individual metal oxide extremely, they still suffer from inferior rate performance, low utilization rate and poor cycle stability. However, by incorporating carbon based materials improve their conductivity as well as power density due to high surface area, high conductivity and stable chemical properties of carbon based materials [44]. Kumar *et al.* fabricated Carbon black (CB) decorated Ni/Co oxide composite electrode through by using the successive ionic layer adsorption and reaction (SILAR) method [45]. Carbon black (CB) decorated Ni/Co oxide composite electrode with 7% weight percentage of CB exhibited a high specific capacitance of 1811 F g<sup>-1</sup> at 0.5 mA cm<sup>-2</sup> with excellent cyclic retention of 92% over 8000 cycles and delivered an impressive high energy density of 91 W h Kg<sup>-1</sup> at a power density of 151 W Kg<sup>-1</sup>, which is significantly higher than pure Ni/Co oxide composite electrode as well as other carbon embedded composites. Veerasubramani *et al.* have adopted a novel approach to fabricate



CNT-deposited CoMoO<sub>4</sub>/Ni foam through a hydrothermal method followed by dry reforming reaction (DRR) of propane and CO<sub>2</sub> [46, 47]. The as fabricated CNT-deposited CoMoO<sub>4</sub>/Ni foam electrode achieved a maximum areal capacity of 160  $\mu\text{Ah cm}^{-2}$  at 1  $\text{mA cm}^{-2}$  with excellent cyclic stability of  $\sim 105\%$  over 3000 cycles and showed 22-fold higher performance than the heat-treated CoMoO<sub>4</sub>/Ni foam. The high electrochemical performance is due to the presence of CNTs on the surface of CoMoO<sub>4</sub>/Ni foam electrode, which increases the conductivity of the electrode and enhances the ion transport kinetics. Further as fabricated ASC device, consists of CNT-deposited CoMoO<sub>4</sub>/Ni foam as the positive electrode and reduced graphene oxide (rGO)-coated carbon cloth (CC) as the negative electrode stored a maximum areal energy density of 122  $\mu\text{Wh cm}^{-2}$  (29.04  $\text{Wh kg}^{-1}$ ) at 2  $\text{mA cm}^{-2}$  and delivered a high power density of 7,727  $\mu\text{Wcm}^{-2}$  (1835  $\text{W kg}^{-1}$ ) 10  $\text{mA cm}^{-2}$  with excellent capacitance retention of more than 95% of its initial capacitance over 1500 cycles. Soam *et al.* synthesized porous type of NiFe<sub>2</sub>O<sub>4</sub>/graphene nanocomposite electrode by a solution based process for supercapacitor application [48]. The as-prepared NiFe<sub>2</sub>O<sub>4</sub>/graphene nanocomposite electrode exhibited a maximum specific capacitance of 207  $\text{Fg}^{-1}$  at a scan rate of 5  $\text{mV/sec}$ , which is almost 4 times larger than pure NiFe<sub>2</sub>O<sub>4</sub> (60  $\text{Fg}^{-1}$ ) and showed the capacitance retention of 95% over 1000 cycles. The significantly enhanced specific capacitance of the NiFe<sub>2</sub>O<sub>4</sub>/graphene nanocomposite electrode material is due to the synergic effect of high porous graphene sheets and NiFe<sub>2</sub>O<sub>4</sub> particles, which are strongly interconnected together leading to a good electric/ionic conduction on the electrode and better contact of ions with the electrode materials. Zhou *et al.* reported a novel and green Cu<sub>2</sub>O template-assisted route based on “coordinating etching and precipitating” process for the synthesis of 3D porous reduced graphene (rGN)/NiCo<sub>2</sub>O<sub>4</sub> film [49]. The as-synthesized 3D rGN/NiCo<sub>2</sub>O<sub>4</sub> film exhibited high specific capacitance of 708.36  $\text{F g}^{-1}$  at a current density of 1  $\text{A g}^{-1}$  with a rate retention of 82.2% as current density ranges from 1 to 16  $\text{A g}^{-1}$ , and remarkable capacitance retention of 94.3% after 6000 cycles at a high current density of 10  $\text{A g}^{-1}$ .

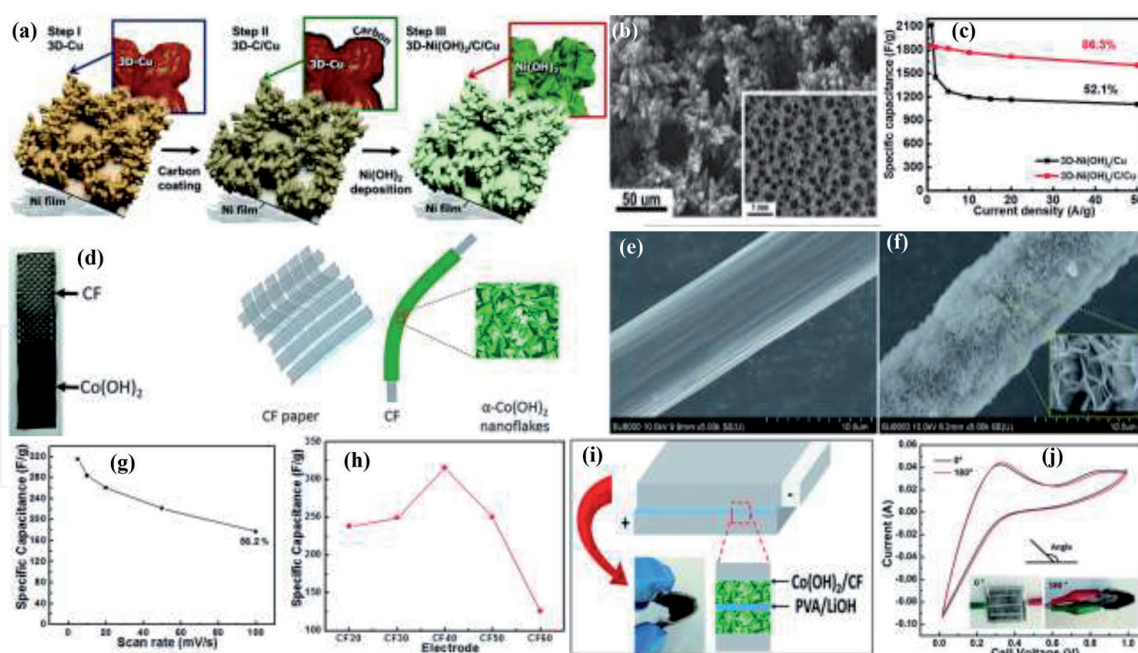
## 5. Carbon-metal hydroxide composites electrode materials

Among the active materials, metal hydroxides have also been considered promising electrode materials for electrochemical SCs because of extremely high specific capacitance. Metal hydroxide in several forms such as Ni(OH)<sub>2</sub>, Co(OH)<sub>2</sub>, NiCo(OH)<sub>2</sub>, Cu(OH)<sub>2</sub>, FeOOH have been investigated as electrodes for SC [50–52]. These materials have large internal spaces for fast insertion and desertion of electrolyte ions. Moreover, these metal hydroxides can be synthesized using simple synthetic approaches. Metal hydroxide consists of stacked layers intercalated having interlayer space to occupy more ions hence larger capacitance.

### 5.1 Carbon-nickel hydroxide (Ni(OH)<sub>2</sub>) composite electrode materials

Ni(OH)<sub>2</sub> is being considered as an attractive candidate as electrode in SCs because of its high theoretical capacitance (2358  $\text{F g}^{-1}$ ). It can be prepared by a simple and low cost process. It has demonstrated good stability in alkaline electrolytes. Its low electrical conductivity is a barrier to achieve higher capacitance. Therefore, a thin region near the surface of nickel hydroxide contributes to the charge storage process due to diffusion-limited redox reactions. To obtain larger capacitance, it has to be utilized completely in the charge storage process. In this regard, researchers have generally adopted conductive additives to effectively improve utilization of active materials and result in larger capacitance. Kang *et al.* have used the same concept





**Figure 5.**

(a) A schematic of the growth process of 3D-Ni(OH)<sub>2</sub>/C/Cu, (b) Morphology of the as-synthesized 3D-Ni(OH)<sub>2</sub>/C/Cu electrode (inset: large-area uniform porous morphology of the 3D-Ni(OH)<sub>2</sub>/C/Cu), (c) Specific capacitance of 3D-Ni(OH)<sub>2</sub>/C and 3D-Ni(OH)<sub>2</sub>/C/Cu as a function of the current density based on the galvanostatic charge/discharge measurement [53], (d) Photograph of CF paper coated with cobalt hydroxide nanoflakes and schematic diagram illustrating the loading procedure of cobalt hydroxide on CF, (e) SEM image of bare CF, (f) SEM image of cobalt hydroxide nanoflakes coated on CF (Inset: magnified SEM image of the nanoflakes), (g) variation of specific capacitance with mass loading of each electrode, (h) specific capacitances of CF electrode at scan rates of 5, 10, 20, 50 and 100 mV/s, (i) schematic of flexible SC fabrication, (j) CV curves at bending conditions of 0° and 180° at scan rate of 20 mV/s [56].

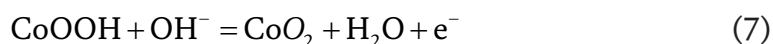
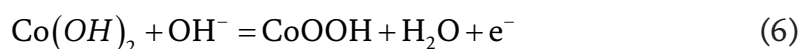
and deposited an ultrathin nickel hydroxide film on carbon-coated 3D porous copper structure in order to prepare binder-free conductive electrode (**Figure 5(a-b)**) [53]. This electrode has short electron path distances and large electrochemical active sites, which improved structural stability for high performance SCs. A carbon coating was used to improve the electron transport behavior and to prevent the oxidation of Cu. Nickel hydroxide supported on mesoporous hollow dendritic three-dimensional-nickel exhibited a specific capacitance of 1860 F g<sup>-1</sup> at a current density of 1 A g<sup>-1</sup> (**Figure 5(c)**). It could retain 86.5% capacitance over 10,000 cycles. Tang *et al.* have prepared an additive-free, nano-architected nickel hydroxide/carbon nanotube (Ni(OH)<sub>2</sub>/CNT) electrode for high performance SCs [54]. This Ni(OH)<sub>2</sub>/CNT electrode was fabricated by depositing Ni(OH)<sub>2</sub> nano-flakes on CNT bundles which were directly grown on Ni foams. The above electrode exhibited the specific capacitance of 3300 F g<sup>-1</sup> and an aerial capacitance of 16 F cm<sup>-2</sup>. Ma *et al.* have synthesized electrode of Ni(OH)<sub>2</sub> nanosheet/3D GF framework using two methods, CVD and hydrothermal [55]. They have compared the capacitive properties of Ni(OH)<sub>2</sub> electrode/graphene fiber with Ni(OH)<sub>2</sub>/Ni foam and Ni(OH)<sub>2</sub> nanosheet/carbon fiber cloth electrodes. Ni(OH)<sub>2</sub> electrode with graphene fiber exhibited better performance in terms of specific capacitance and rate capability. The Ni(OH)<sub>2</sub> nanosheet/graphene fiber electrode exhibited electrochemical capacitance as high as 2860 F g<sup>-1</sup> at a current density of 2 A g<sup>-1</sup>, and maintains 1791 F g<sup>-1</sup> at 30 A g<sup>-1</sup>.

## 5.2 Carbon-cobalt hydroxide (Co(OH)<sub>2</sub>) composite electrode materials

Co(OH)<sub>2</sub> has recently received increasing attention as electrode for SC application because of its low cost and high capacitance. Jagadale *et al.* have used cobalt hydroxide nanoflakes which were uniformly loaded on flexible carbon fiber (CF)

paper as electrode for SC (**Figure 5(d)**) [56]. The carbon fiber was basically used to provide unique porous nanostructure offering low ion diffusion and charge transfer resistance to the electrode (**Figure 5(e, f)**). The electrode exhibited maximum specific capacitance of  $386.5 \text{ F g}^{-1}$  at a current density of  $1 \text{ mA cm}^{-2}$  with a mass loading of  $2.5 \text{ mg cm}^{-2}$  (**Figure 5(g, h)**). An energy density of  $133.5 \text{ W h kg}^{-1}$  has been obtained with power density of  $1769 \text{ W kg}^{-1}$ . The carbon fiber has improved the cyclic stability of 92% over 2000 cycles. To check applicability of electrodes, these electrodes further employed to fabricate flexible solid state supercapacitor. CV curves of SC at bending conditions of  $0^\circ$  and  $180^\circ$  at scan rate of  $20 \text{ mV/s}$ . It is clearly seen that the area under curve doesn't change significantly after bending which proves that SC is highly flexible and does not lose its structural integrity under bending conditions (**Figure 5(i, j)**).

Two possible reactions are suggested for the electrochemical reactions of  $\text{Co(OH)}_2$  in KOH electrolyte [57]:

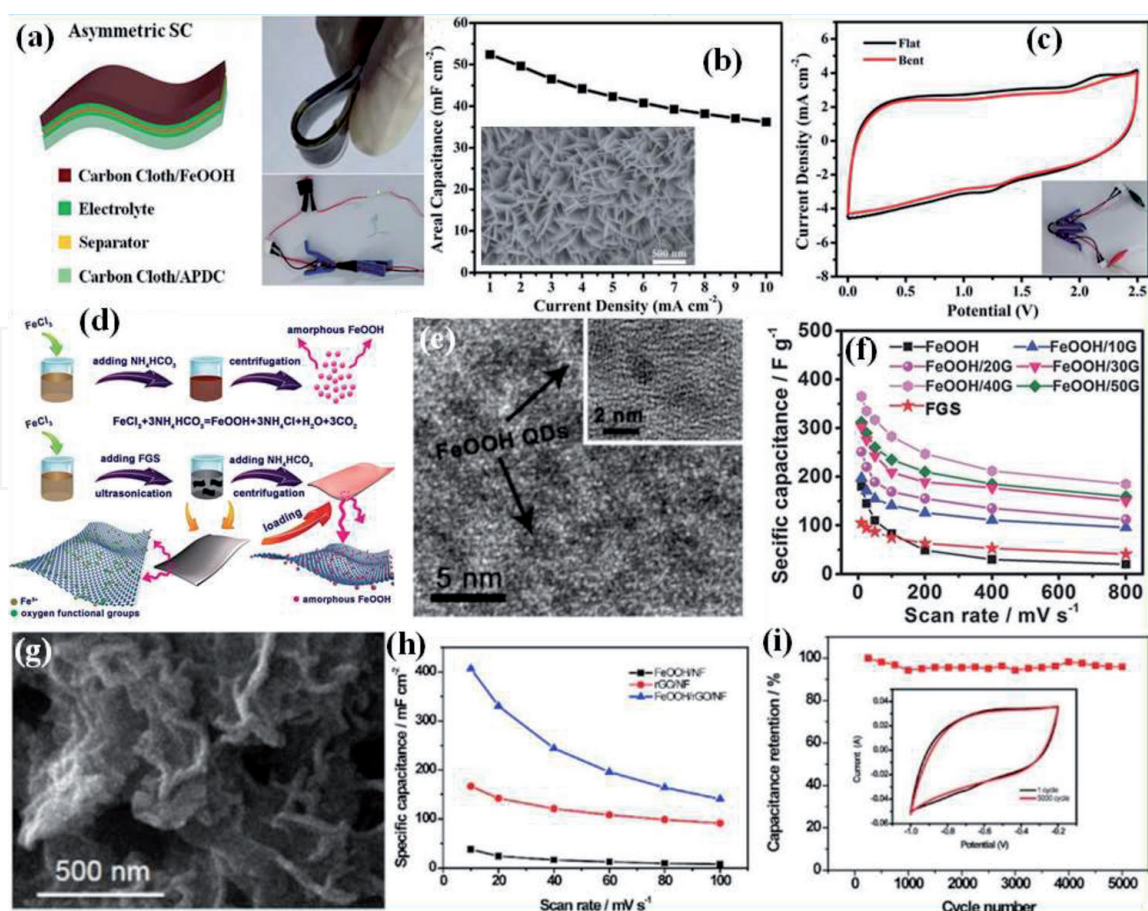


$\text{Co(OH)}_2$  nano-sheet-decorated graphene-CNT composite structure has been designed for SC application [58]. Suspensions method was used to prepare graphene-CNT composite by sonication and vacuum filtration. The graphene-CNT composite may offer high porosity with high conductivity, chemical stability and a three-dimensional structure. The vertically aligned  $\text{Co(OH)}_2$  nano-sheets were then deposited on 3D graphene-CNT composite by solution based process. The ASC of  $\text{Co(OH)}_2$  with graphene-CNT has shown a specific capacitance of  $310 \text{ F g}^{-1}$ . The electrode exhibited an energy density of  $172 \text{ W h kg}^{-1}$  and maximum power density of  $198 \text{ kW kg}^{-1}$  in ionic liquid electrolyte 1-ethyl-3-methylimidazolium bis (trifluoromethanesulfone)imide (EMI-TFSI). Zhang *et al.* have deposited  $\text{Co(OH)}_2$  on multi-walled CNT which were grown on the carbon paper substrate [59]. The composite electrode showed the specific capacitance of  $1083 \text{ F g}^{-1}$  determined at a current density of  $0.83 \text{ A g}^{-1}$  in aqueous electrolyte. CNTs were added to  $\text{Co(OH)}_2$  in order to improve the electrical conductivity of the electrode. The interconnected nanosheets of the  $\text{Co(OH)}_2$  would help to facilitate the contact of the electrolyte with active materials, exhibiting good cycling stability and lifetime.

### 5.3 Carbon-iron oxy hydroxide ( $\text{FeOOH}$ ) composite electrode materials

$\text{FeOOH}$  has been recognized is an attractive electrode material for SC due to low cost, high theoretical specific capacitance, and broad potential window. In addition, the unique tunnel structure of  $\text{FeOOH}$  with open permeable channels are beneficial for ion transportation and shorten the diffusion path for electrolyte ion diffusion [60]. However, the poor electrical conductivity and low specific surface area limited the use of  $\text{FeOOH}$  as a potential electrode for SC, which limited specific capacitance and rate capability [61]. Alternatively, composite system by assembling  $\text{FeOOH}$  on the carbon based supporting materials (AC, carbon black, graphene, etc.) can be enhance the capacitive performance. Shen *et al.* synthesized radiating  $\gamma\text{-FeOOH}$  Nanosheets on CC substrate ( $\gamma\text{-FeOOH NSs/CC}$ ) by a simple one-step electrodeposition method and investigated its pseudocapacitive behaviour in a typical ionic liquid [1-ethyl-3-methylimidazolium bis imide (EMIM-NTF2)] through electrochemical quartz crystal microbalance (EQCM). The charge storage is mainly due





**Figure 6.**

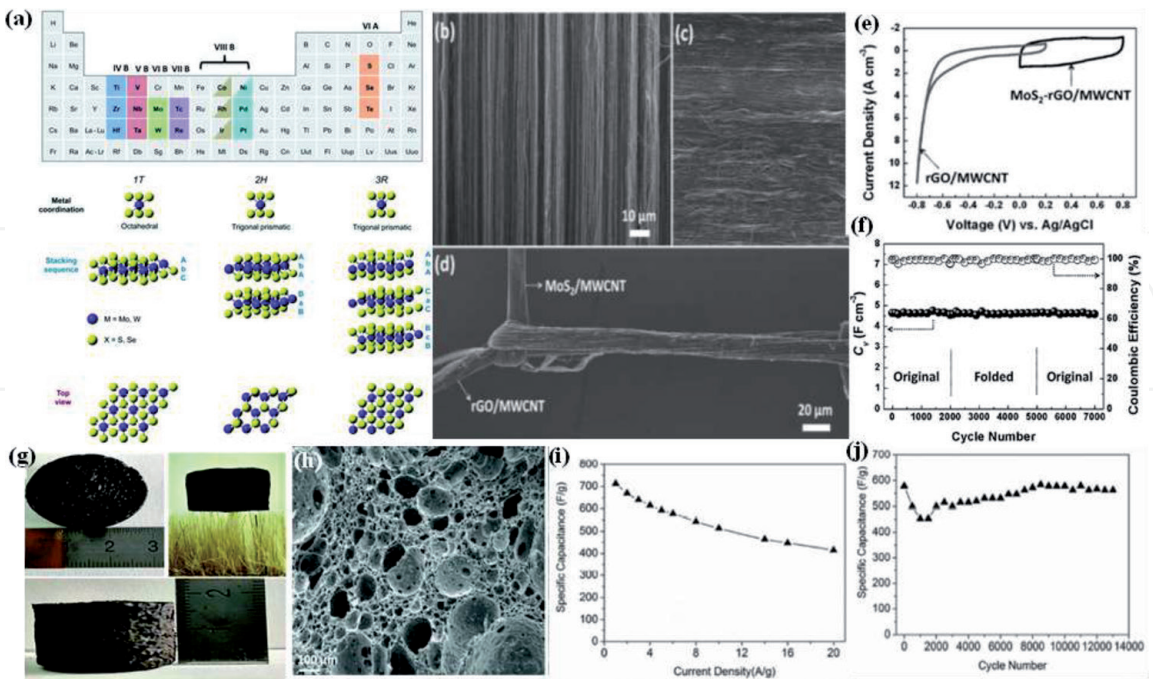
(a) Schematic illustrations of the fabrication procedure for the FeOOH//APDC f-SSC electrodes and flexibility and operating status as supercapacitor device, (b) The areal capacitance as a function of the discharge current density (Inset: SEM images of as-prepared  $\gamma$ -FeOOH nanosheets on a carbon cloth substrate), (c) CV curves of the FeOOH//APDC f-SSC at bent and flat statuses [62]. (d) Schematic illustration of the synthesis of amorphous FeOOH QDs and amorphous FeOOH/FGS hybrid nanosheets, (e) HRTEM images of the FeOOH QDs (Inset: enlarged HRTEM for FeOOH QDs), (f) The specific capacitances of the FeOOH, functionalized graphene sheet (FGS), and FeOOH/FGS composite electrodes as a function of the scan rate [64]. (g) High-magnification SEM images of as-prepared 3D FeOOH/rGO/NF, (h) Areal capacitance of FeOOH/NF, rGO/NF and FeOOH/rGO/NF electrodes calculated from CV curves as a function of scan rate, and (i) Cycling performance of MnO<sub>2</sub>//FeOOH-ASC collected at a scan rate of 100 mV s<sup>-1</sup> for 5000 cycles [66].

to the insertion and extraction of [EMIM]<sup>+</sup> cations through the transport pathways offered by the crystalline network of  $\gamma$ -FeOOH during charging-discharging process.  $\gamma$ -FeOOH NSs/CC exhibited a good areal capacitance of 210 mF cm<sup>-2</sup> at a current density of 1 mA cm<sup>-2</sup> and the ASC device made of  $\gamma$ -FeOOH//APDC (activated polyaniline-derived carbon nanorods) solid-state flexible SCs acquired a high energy density of 1.44 mW h cm<sup>-3</sup> at a current density of 3 A g<sup>-1</sup> with a cycling stability of 80.5% retention over 2000 cycles (**Figure 6(a-c)**) [62]. An amorphous FeOOH nanoflowers@multi-walled CNT (FeOOH NFs@MWCNTs) composite was prepared by Sun *et al.* [63]. The as-prepared composite electrode displays a high specific capacitance of 345 F g<sup>-1</sup> at 1 A g<sup>-1</sup> current density and outstanding rate performance (167 F g<sup>-1</sup> at 11.4 A g<sup>-1</sup>) with good cycling stability of 76.4% over 5000 cycles. The outstanding electrochemical performance of the composite electrode is due to the mesoporous structure and high surface area of the electrode materials as well as fast ion/electronic transport and easy accessibility of the active materials to electrolytes. Liu *et al.* demonstrated FeOOH quantum dots (QDs)/graphene hybrid nanosheets, which exhibited a high specific capacitance of 365 F g<sup>-1</sup> at a current density of 1 A g<sup>-1</sup> with excellent capacitance retention of 89.7% of initial capacitance over 20000 cycles as well as a great rate capability (189 F g<sup>-1</sup> at a high current density of 128 A g<sup>-1</sup>) (**Figure 6(d-f)**) [64]. In addition, specific capacitance

of the SC increased to  $1243 \text{ F g}^{-1}$  at  $5 \text{ mV s}^{-1}$  while the voltage window was extended from  $-0.8$  to  $0 \text{ V}$  to  $-1.25$  to  $0 \text{ V}$  but the cycling performance declined sharply. Wei *et al.* synthesized ultrathin  $\alpha$ -FeOOH nanorods/graphene oxide (GO) composite by hydrothermal method, which exhibited high specific capacitance of  $127 \text{ F g}^{-1}$  at a current density of  $10 \text{ A g}^{-1}$ , good cyclic performance of 85% capacitance retention over 2000 cycles, and excellent rate capability ( $100 \text{ F g}^{-1}$  at  $20 \text{ A g}^{-1}$ ) as compared to than bare  $\alpha$ -FeOOH nanorods [65]. The outstanding electrochemical performance of  $\alpha$ -FeOOH nanorods/GO composite is due to its unique structure, which provides fast electron/ions transport and high charging/discharging rate. 3D FeOOH/reduced graphene oxide/Ni foam (FeOOH/rGO/NF) based hybrid electrodes fabricated by the electrodeposition of FeOOH nanosheets on the rGO/Ni foam surface exhibited an exception high areal capacitance of  $406.5 \text{ mF cm}^{-2}$  at a scan rate of  $10 \text{ mV s}^{-1}$ , which is 10-fold higher than the bare FeOOH/NF electrode (**Figure 6(g-i)**) [66]. This high areal capacitance of FeOOH/rGO/NF is due to the improved conductivity and increased surface area, which not only provide a superior pathway for electron transfer, but also offer more active sites for energy storage. In addition, an ASC device made of 3D FeOOH/rGO/NF electrode as anode and  $\text{MnO}_2/\text{TiN}$  electrode as cathode attained a remarkable maximum power density of  $0.19 \text{ W cm}^{-3}$  with maximum energy density of  $0.48 \text{ mW h cm}^{-3}$ .

## 6. Carbon-transition metal dichalcogenides (TMDs) composite electrode materials

TMDs are layered inorganic materials with a chemical configuration of  $\text{MX}_2$ , in which M is a transition metal element (M: Ti, Mo, V, W, Re, Ta), and X can be any chalcogenide element (X: S, Se, Te) (**Figure 7(a)**). Each  $\text{MX}_2$  unit cell is stacked



**Figure 7.** (a) Different metal coordination and stacking sequence in TMD unit cells [67]. (b-d) SEM images of aligned MWCNT sheets, MWCNT/MoS<sub>2</sub> hybrids, and tightly knotted MoS<sub>2</sub>/MWCNT and rGO/MWCNT fibers, respectively. (e) CV curves of rGO/MWCNT (cathode) and MoS<sub>2</sub>-rGO/MWCNT (anode) at different potential windows. (f) Cycle stability test of the fiber-based asymmetric device at  $0.55 \text{ A cm}^{-2}$  current density [74]. (g) Optical photographs and (h) SEM images of the MoS<sub>2</sub>/C composite aerogel. (i) specific capacitances at different current densities and (j) long-term cycle stability at a current density of  $6 \text{ A g}^{-1}$  of the MoS<sub>2</sub>/C composite aerogel electrode material [75].



together through Vander Waals force in such a way that transition metal layer is present in between the two chalcogen sheets [67]. On the basis of crystal structure, there are two types of phases of TMDs, which are metallic 1T phase with an octahedral structure and semiconducting 2H phase with a trigonal structure. Recently, TMDs have been attracted great attention as SC electrode materials due to their large surface area, low cost, variable oxidation states, high mechanical properties, high chemical stability and easy synthesis [68]. The variable oxidation states, large surface area, and active edges of TMDs allow electrical double layer and fast/reversible redox charge storage mechanisms and offer high energy storage capability in SCs. However, due to the inherently low conductivity, poor cycle life, large volume change during cycling and restacking limits their electrochemical performance as SC electrodes [69]. For example, Soon *et al.* has synthesized sheet-like morphology of MoS<sub>2</sub> by chemical vapor deposition method, which has a very large surface area favorable for double layer storage. But due to its poor electrical conductivity, it showed low specific capacitance of  $\sim 100 \text{ F g}^{-1}$  at a scan rate of  $1 \text{ mV s}^{-1}$  [70]. Therefore, in order to improve the electrochemical performance of TMDs, they have been compositing with highly conducting/electroactive carbon based supporting materials by various top-down/bottom-up and both synthetic approaches. The synergic effect of carbon-TMDs based composite materials such as carbon offers conductive channels and increasing the interfacial contact, whereas TMDs provide a short ion diffusion path and followed by short electron transport path enhances the overall electrochemical performance of the SC.

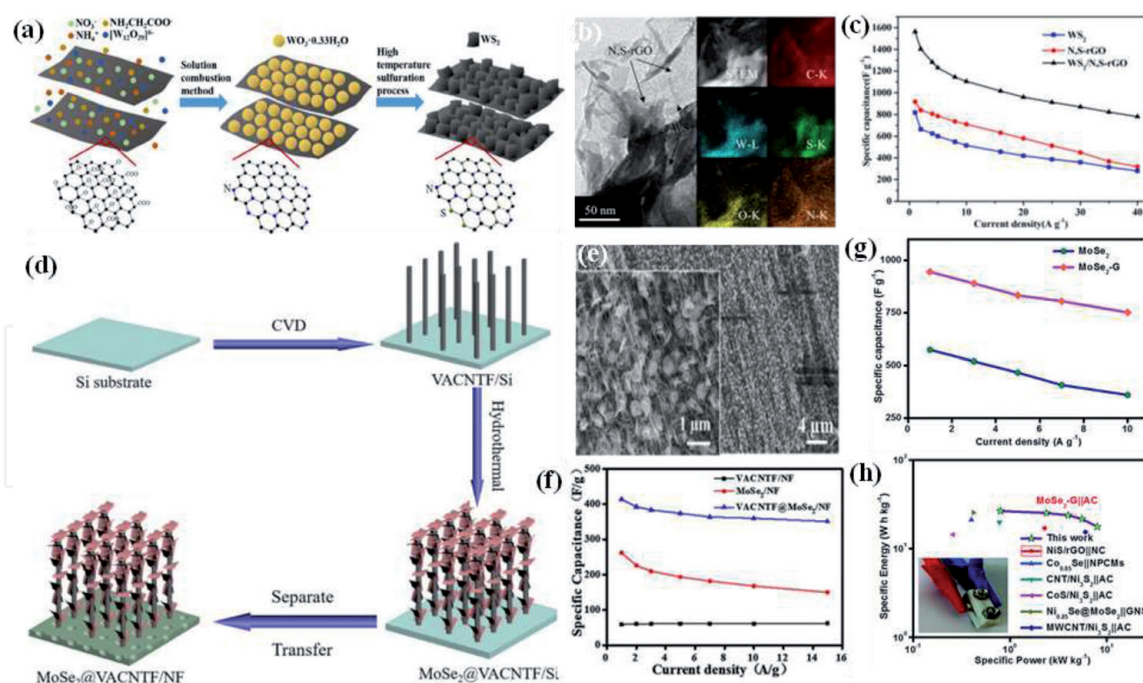
### 6.1 Carbon-MoS<sub>2</sub> composite electrode materials

MoS<sub>2</sub>/MWCNT nanocomposite synthesized by a hydrothermal method exhibited a large surface area and fast ionic transport properties and showed a high specific capacitance of  $452.7 \text{ F g}^{-1}$  with good cycling stability (95.8% retention after 1000 cycles), which is almost three times larger than the bare MoS<sub>2</sub> ( $149.6$  to  $452.7 \text{ F g}^{-1}$ ) [71]. Ali *et al.* fabricated MoS<sub>2</sub>/graphene composite from bulk MoS<sub>2</sub> and graphite rod through facile electrochemical exfoliation method and exhibited high specific capacitance of  $227 \text{ F g}^{-1}$  as compared with the exfoliated MoS<sub>2</sub> ( $70 \text{ F g}^{-1}$ ) and exfoliated graphene ( $85 \text{ F g}^{-1}$ ) at a current density of  $0.1 \text{ A g}^{-1}$  [72]. The high specific capacitance of MoS<sub>2</sub>/graphene composite is due to the synergistic effect between MoS<sub>2</sub> and graphene. Ali *et al.* demonstrated the electrochemical performance of MoS<sub>2</sub>/CNT/GNF composite and compared the performance with MoS<sub>2</sub>/CNTs, MoS<sub>2</sub>/graphene nanoflakes [73]. It has been noticed that the electrochemical charge storage performance has been improved by incorporation of the carbon materials into the composite and the composite showed a maximum specific capacitance of  $104 \text{ F g}^{-1}$  at a current density of  $0.5 \text{ A g}^{-1}$  with capacitance retention of 75% after the 1000 cycle at a scan rate of  $10 \text{ mV/s}$ . Another interesting MoS<sub>2</sub>-rGO/MWCNT fiber electrode was fabricated by incorporating rGO nanosheets and MoS<sub>2</sub> into aligned MWCNT, which operated at a stable potential window of  $1.4 \text{ V}$  and exhibited high coulombic efficiency of 100% over 7000 cycles in the bending state (**Figure 7(b-f)**) [74]. Zhang *et al.* reported an agarose induced technique to synthesize MoS<sub>2</sub>/carbon composite aerogel, which showed a high specific capacitance of  $712.6 \text{ F g}^{-1}$  at a current density of  $1 \text{ A g}^{-1}$  with cyclic stability of 97.3% over 13000 charge-discharge cycles (**Figure 7(g-j)**) [75]. The high specific capacitance of MoS<sub>2</sub>/carbon composite aerogel is because of 3D intercalated network with hierarchal porous and interlayer MoS<sub>2</sub> expanded structures, which were beneficial for easy ion transportation. 3D graphene/MoS<sub>2</sub> composite electrode material has been synthesized by Sun *et al.* and co-workers through a simple and facile one-step hydrothermal process [76]. The as-synthesized composite electrode exhibited

gravimetric capacitance of  $410 \text{ F g}^{-1}$  at a current density of  $1 \text{ A g}^{-1}$  and an excellent cycling stability of 80.3% over 10,000 continuous charge-discharge cycles at  $2 \text{ A g}^{-1}$  current density. The outstanding electrochemical performance of 3D graphene/MoS<sub>2</sub> composite electrode is due to the 3D architecture of conducting network graphene and flower-like structure of MoS<sub>2</sub>, which enhances the electrolyte ions diffusion process.

## 6.2 Carbon-WS<sub>2</sub> composite electrode materials

WS<sub>2</sub> nanoplates supported on carbon fiber cloth (WS<sub>2</sub>/CFC) have been synthesized by a facile solvothermal process and used as electrode material for SC [77]. The 3D network of CFC not only prevent the agglomeration of WS<sub>2</sub> nanoplates but also enhances the ion transport efficiency due to low charge transfer resistance ( $R_{ct}$ ) of  $0.1 \Omega$ . The as fabricated WS<sub>2</sub>/CFC electrode exhibited a high specific capacitance of  $399 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$  current density with cyclic retention of 99% over charge-discharge 500 cycles, which is higher than compared with bare WS<sub>2</sub>. In addition, developing such composite of WS<sub>2</sub> with the carbon fibre helps for fabricating wearable SCs which are in demand for wearable electronics. Yang *et al.* fabricated WS<sub>2</sub>@CNT hybrid film electrode by incorporating conducting CNTs into WS<sub>2</sub>. The WS<sub>2</sub>@CNT hybrid film with a unique skeleton structure showed a maximum specific area capacitance of  $752.53 \text{ mF cm}^{-2}$  at a scan rate  $20 \text{ mV s}^{-1}$  with very good cyclic stability by only loss of 1.28% capacitance after 10,000 cycles. In addition, a quasi-solid-state flexible SC made by WS<sub>2</sub>@CNT hybrid film exhibited excellent bendability under bending to 135 10, 000 times with the loss of 23.12% at scan rate of  $100 \text{ mV s}^{-1}$  [53]. Tu *et al.* have been synthesized WS<sub>2</sub>/RGO hybrid material by using a simple molten salt process, which showed a high specific capacitance of  $2508.07 \text{ F g}^{-1}$  at  $1 \text{ mV s}^{-1}$  scan rate with excellent capacitance retention of 98.6% over 5000 cycles, due to synergic effect of highly conducting RGO and large charge-accumulating sites of WS<sub>2</sub> networks. Likewise, Xu *et al.* demonstrated 3D composite of WS<sub>2</sub> nanoflakes and quantum dots on N and S co-doped reduced graphene oxide (WS<sub>2</sub>/N,S-rGO) crumpled nanosheets through a rapid solution combustion synthesis of the precursor and subsequent gas-solid phase sulfurization process, which presented a significant specific capacitance of  $1562.5 \text{ F g}^{-1}$  at  $1 \text{ A g}^{-1}$  current density, and a rate capability of  $780 \text{ F g}^{-1}$  at  $40 \text{ A g}^{-1}$  (Figure 8(a-c)) [78]. The high specific capacitance of WS<sub>2</sub>/N,S-rGO hybrids is because of synergistic effect between WS<sub>2</sub> and N,S-rGO, where N,S-rGO provides larger contact surface area, excellent charge transport, and shorter ion diffusion path. Hierarchical MoSe<sub>2</sub>/C hybrid was successfully fabricated by facile one-step hydrothermal strategy, which composed of few-layered MoSe<sub>2</sub> nanosheets and amorphous carbon obtained from the decomposition of the triethylene glycol. As fabricated hierarchical MoSe<sub>2</sub>/C electrode exhibited high specific capacitance of  $878.6 \text{ F g}^{-1}$  in comparison with the bare MoSe<sub>2</sub> at current density of  $1 \text{ A g}^{-1}$  and maintained 98% of initial capacitance over 2000 cycles without obvious decrease. The superior electrochemical performances of MoSe<sub>2</sub>/C hybrid can be ascribed to hierarchical structure of MoSe<sub>2</sub> and conducting nature of carbon, which help for providing large surface area for electrochemical reactions and enhancing charge carriers transfer at the electrolyte/electrode interface [79]. Liu *et al.* fabricated VACNTF@MoSe<sub>2</sub>/NF composite electrode through a combined chemical vapor deposition method and solvothermal methods by growing MoSe<sub>2</sub> nanoflakes on the vertically aligned carbon nanotube array film (VACNTF) with binder-free nickel foam as current collector [80]. The as fabricated VACNTF@MoSe<sub>2</sub>/NF composite electrode exhibited high specific capacitance of  $435 \text{ F g}^{-1}$  at a current density of  $1 \text{ A g}^{-1}$  with outstanding cycling stability



**Figure 8.**

(a) Schematic illustration of synthetic processes of WS<sub>2</sub>/N,S-rGO hybrid, (b) HRTEM, STEM and EDS elemental mapping images of WS<sub>2</sub>/N,S-rGO hybrid, and (c) The specific capacitances of the WS<sub>2</sub>, N,S-rGO and WS<sub>2</sub>/N,S-rGO hybrid at different current densities [78]. (d) Schematic illustration of the synthesis process of the VACNTF@MoSe<sub>2</sub>/NF composite electrode, (e) SEM images of the VACNTF@MoSe<sub>2</sub>/NF composite (inset: high magnification), and (f) The specific capacitance comparison of the MoSe<sub>2</sub>/NF, VACNTF/NF and VACNTF@MoSe<sub>2</sub>/NF electrodes at various current densities [80]. (g) Specific capacitance of the MoSe<sub>2</sub>, NS and MoSe<sub>2</sub>/G nanohybrid based electrodes as a function of current density, and (h) Ragone plot for the MoSe<sub>2</sub>/G||AC ASC device (inset: photograph of ASC device) [81].

of 92% after 5000 cycles (**Figure 8(d-f)**). In addition, the VACNTF@MoSe<sub>2</sub>/NF composite based ASC displays a high energy density with 22 W h kg<sup>-1</sup> for a power density of 330 W kg<sup>-1</sup>. Kirubasankar *et al.* MoSe<sub>2</sub>/graphene nanohybrid based electrode prepared by a simple and facile sonochemical route, which showed higher specific capacitance (945 F g<sup>-1</sup>) as compared to MoSe<sub>2</sub> nanosheets (576 F g<sup>-1</sup>) at 1 A g<sup>-1</sup> current density. Further, as fabricated ASC device based on MoSe<sub>2</sub>/graphene nanohybrid retains 88% of its capacitance over 3000 cycles and delivers an energy density of 26.6 W h kg<sup>-1</sup> at a power density of 0.8 kW kg<sup>-1</sup> (**Figure 8(g, h)**) [81]. The high specific capacitance with better rate capability is due to the effective penetration and migration of electrolyte, reduction of the contact resistance and shortness of the diffusion path of ions between the electrode-electrolyte interface, which enhances the redox kinetics and provide maximum utilization of the electroactive area, so providing a high structural stability during charge-discharge processes. Similarly, Huang *et al.* demonstrated MoSe<sub>2</sub>/graphene on flexible Ni electrode, which could deliver a specific capacitance of 1422 F g<sup>-1</sup> and fully retention of initial capacitance over 1500 cycles [82]. Wei *et al.* first time fabricated free-standing SC anode based on 3D MoSe<sub>2</sub> nanoflowers (MoSe<sub>2</sub> NFs) and hierarchically porous anisotropic carbonized delignified wood (CDW), which exhibited ultrahigh capacitance of 1043 mF cm<sup>-2</sup> at a current density of 1 mA cm<sup>-2</sup> and excellent cycling stability less than 5% capacitance loss over 5000 cycles. The ASC device was made by integration of 3D MoSe<sub>2</sub> NFs@CDW anode and a common MnO<sub>2</sub>-based cathode, which exhibited a high capacitance of 415 mF cm<sup>-2</sup> at a current density of 2.5 mA cm<sup>-2</sup> with high energy density of 147 mW h cm<sup>-2</sup> at power density of 2 mW cm<sup>-2</sup>. These results confirm that 3D MoSe<sub>2</sub> NFs@CDW based anode can be used as a potential anode for the development of high-performance SCs [83].



## 7. Conclusion

In the past few decades, SCs have been extensively studied as energy storage devices and more focusing area in the multidisciplinary science over the world. The selection of high performance SC electrode materials based on high specific capacitance, low internal resistance and good stability. In this article, we have reviewed the carbon-based composite materials (*i.e.*, metal oxide, metal hydroxide, TMDs composited with carbon materials) as promising SC electrode materials due to the synergic effect of the composite materials such as high surface area, interconnected porous structure, high electrical conductivity, excellent wettability towards the electrolyte, and presence of electrochemically active surface functionalities of the carbon supports which improves the EDL capacitance while metal oxide or metal hydroxide or TMDs enhances electrochemical performance through pseudocapacitive/faradaic charge-storage process. The carbon-based composite materials demonstrated herein usually possesses high specific capacity, impressive energy density and maintain long term stability with better mechanical flexibility. We also observe the microstructural changes in the carbon-based composite materials would be more favorable for fabrication of high performance supercapacitor. We also explained how the composite materials overcome the traditional obstacles while formulating the standard electrode designs as compare to individual components.

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## Author details

Prasanta Kumar Sahoo<sup>1</sup>, Chi-Ang Tseng<sup>2</sup>, Yi-June Huang<sup>3</sup> and Chuan-Pei Lee<sup>4\*</sup>

<sup>1</sup> Department of Mechanical Engineering, Siksha 'O' Anusandhan, Deemed to be University, Bhubaneswar, Odisha, India

<sup>2</sup> Department of Chemistry, National Taiwan University, Taipei, Taiwan

<sup>3</sup> Graduate Institute of Nanomedicine and Medical Engineering, College of Biomedical Engineering, Taipei Medical University, Taipei, Taiwan

<sup>4</sup> Department of Applied Physics and Chemistry, University of Taipei, Taiwan

\*Address all correspondence to: [CPLee@utapei.edu.tw](mailto:CPLee@utapei.edu.tw)

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