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Chemical Nanomanipulation of Two-Dimensional Nanosheets and Its Applications

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1. Introduction

Two-dimensional (2D) nanosheets obtained *via* exfoliation of layered compounds have attracted intensive research in recent years, opening up new fields in the science and technology of 2D nanomaterials.¹⁻⁶ These 2D nanosheets, which possess atomic or molecular thickness and infinite planar dimensions, are emerging as important new materials due to their unique properties. Research in such exotic 2D systems recently intensified as a result of emerging progress in graphene (carbon nanosheet)^{1, 2} and novel functionalities in oxide nanosheets.³⁻⁵ In particular, oxide nanosheets are exceptionally rich in both structural diversity and electronic properties, with potential application in areas ranging from catalysis to electronics. Now, by using the exfoliation approach, it is possible to investigate dozens of different 2D oxide nanosheets in search of new phenomena and applications.

One of the important and attractive aspects of the exfoliated nanosheets is that various nanostructures can be fabricated using them as 2D building blocks.⁷⁻¹⁸ It is even possible to tailor superlattice-like assemblies, incorporating into the nanosheet galleries a wide range of materials such as organic molecules, polymers, and inorganic and metal nanoparticles. Sophisticated functionalities or nanodevices may be designed through the selection of nanosheets and combining materials, and precise control over their arrangement at the molecular scale.

In this chapter, we review the current research on oxide nanosheets. Our particular focus is placed on recent progress that has been made in the synthesis and properties of oxide nanosheets, highlighting emerging functionalities in electronic applications.

2. Synthesis of oxide nanosheets

Various nanosheets based on transition-metal oxides have been synthesized by delaminating the precursor crystals of layered oxide into their elemental layers (Table 1).

Chemical exfoliation is the most facile route for making isolation of single layers (oxide nanosheets) separately from thicker layered compounds (Fig. 1). These procedures have attracted much attention as an efficient method for preparing single layers with lateral sizes of up to several micrometers. Pioneering works in this line appeared in the 1990s by Sasaki

Ti oxide	$\text{Ti}_{0.91}\text{O}_2$, $\text{Ti}_{0.87}\text{O}_2$, $\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2$, $\text{Ti}_{0.6}\text{Fe}_{0.4}\text{O}_2$, $\text{Ti}_{(5.2-2x)}/6\text{Mn}_{x/2}\text{O}_2$ ($0 \leq x \leq 0.4$) $\text{Ti}_{0.8-x/4}\text{Fe}_{x/2}\text{Co}_{0.2-x/4}\text{O}_2$ ($0 \leq x \leq 0.8$)	Semiconducting, Photocatalytic, Dielectric, Ferromagnetic
Mn oxide	MnO_2	Redoxable
Nb/Ti and Ta oxide	Nb_6O_{17} , Nb_3O_8 , TiNbO_5 , Ti_2NbO_7 , $\text{Ti}_5\text{NbO}_{14}$, TaO_3	Photocatalytic, Dielectric
Perovskite	$\text{Gd}_{1.4}\text{Eu}_{0.6}\text{Ti}_3\text{O}_{10}$, $\text{Bi}_4\text{Ti}_3\text{O}_{12}$, LaNb_2O_7 , $\text{La}_{0.9}\text{Eu}_{0.05}\text{Nb}_2\text{O}_7$, SrTa_2O_7 , $\text{La}_{0.7}\text{Tb}_{0.3}\text{Ta}_2\text{O}_7$, $\text{Eu}_{0.56}\text{Ta}_2\text{O}_7$, $\text{Ca}_2\text{Nb}_3\text{O}_{10}$, $\text{Sr}_2\text{Nb}_3\text{O}_{10}$, $\text{Ca}_2\text{Ta}_3\text{O}_{10}$, $\text{Sr}_2\text{Ta}_3\text{O}_{10}$	Photocatalytic, Dielectric, Photoluminescence
Mo oxide	MoO_2	Conducting
Ru oxide	$\text{RuO}_{2.1}$, RuO_2	Redoxable, Conducting
W oxide	W_2O_7 , $\text{Cs}_4\text{W}_{11}\text{O}_{36}$	Redoxable Photochromic

Table 1. Library of oxide nanosheets

et al.,^{19, 20} reporting the successful delamination of layered titanates into single titanate nanosheets. Prompted by the findings related to functional oxide nanosheets, several strategies on functional oxides can be found in the literature.

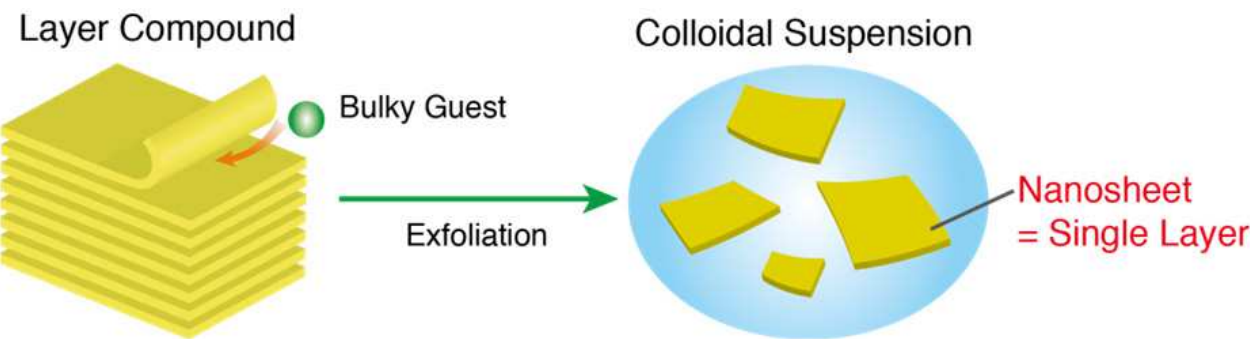


Fig. 1. Schematic illustration for the exfoliation of a layer compound into nanosheets.

In the case of metal oxides, protonation usually resulted in electrostatic repulsions that facilitated exfoliation. By this procedure, single layers of Ti oxides,¹⁹⁻²² Mn oxides,²³ Nb/Ta oxides,²⁴⁻²⁶ Mo oxides,²⁷ Ru oxides,²⁸ and W oxides,²⁹ as well as sheets of several perovskites^{7, 9, 30-35} have been separated from bulk samples (Fig. 2).

In these cases, a chemical intercalator that assists the separation of layers and hampers the reassembly of the bulk lamellar material is always required. Tetrabutylammonium (TBA) is the most commonly used intercalator, but also tetramethylammonium and ethylammonium have been used successfully for these purposes. Layered transition-metal oxides such as $\text{Cs}_{0.7}\text{Ti}_{1.825}\text{Y}_{0.175}\text{O}_4$ (Y: vacancy), $\text{K}_{0.45}\text{MnO}_2$, and $\text{KCa}_2\text{Nb}_3\text{O}_{10}$ can be used as the starting material for the nanosheet.^{19, 20, 23, 31} A common feature of these host compounds is cation-exchange properties involving interlayer alkali metal ions, which are a key to facilitating exfoliation. As the first step to delamination, these layered materials are acid-exchanged into protonated forms such as $\text{H}_{0.7}\text{Ti}_{1.825}\text{Y}_{0.175}\text{O}_4 \cdot \text{H}_2\text{O}$, $\text{H}_{0.13}\text{MnO}_2 \cdot 0.7\text{H}_2\text{O}$, and

$\text{HCa}_2\text{Nb}_3\text{O}_{10} \cdot 1.5\text{H}_2\text{O}$, in which the interlayer alkali metal ions can be completely removed under suitable conditions while maintaining the layered structure. The resulting protonic oxides are subsequently delaminated through reaction with a solution containing TBA ions, producing turbid colloidal suspensions of $\text{Ti}_{0.91}\text{O}_2$, MnO_2 , and $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ nanosheets. Such an exfoliation process is quite general: exfoliation of the other layered host compounds proceeds in a similar fashion.

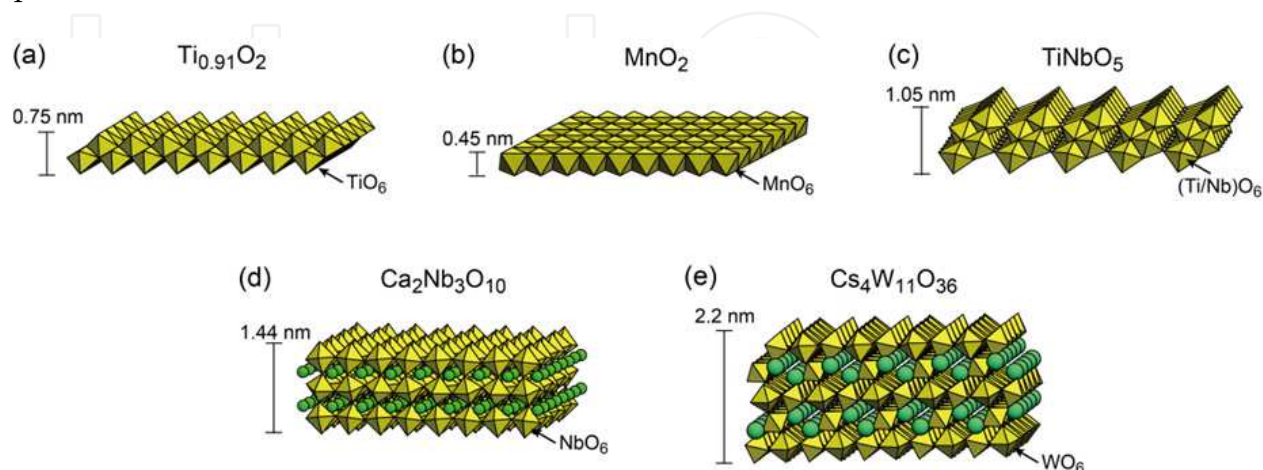


Fig. 2. Structures of selected oxide nanosheets. (a) $\text{Ti}_{0.91}\text{O}_2$, (b) MnO_2 , (c) TiNbO_5 , (d) $\text{Ca}_2\text{Nb}_3\text{O}_{10}$, (e) $\text{Cs}_4\text{W}_{11}\text{O}_{36}$.

These materials have prompted many efforts to elucidate their structural properties. The formation of unilamellar nanosheets was confirmed by direct observation with atomic force microscopy (AFM), x-ray diffraction (XRD), and transmission electron microscopy (TEM).³⁶⁻⁴⁰ Fig. 3 depicts an AFM image for $\text{Ti}_{0.87}\text{O}_2$ nanosheets. The AFM data clearly reveals a sheet-like morphology, which is inherent to the host layer in the parent compounds. The average thickness was 0.93 ± 0.1 nm. This value is nearly comparable to the crystallographic thickness of the host layer in the corresponding parent compound, supporting the formation of unilamellar nanosheets. On the other hand, the lateral size depends on the choice of starting materials. For nanosheets derived from polycrystalline powder samples, the lateral size ranges from submicrometers to several tens of micrometers. After tuning the exfoliation conditions by using flux-grown single crystals, the technique provides high-quality nanosheet crystallites up to ~ 100 μm in size, which is suitable for electronic applications.²¹

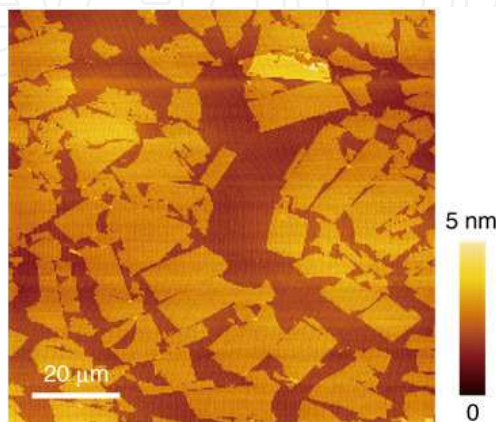


Fig. 3. AFM image of $\text{Ti}_{0.87}\text{O}_2$ nanosheets dispersed on a Si substrate.

3. Chemical nanomanipulation of oxide nanosheets

Oxide nanosheets are an important and promising component for creating new materials. Oxide nanosheets have a high 2D anisotropy of the crystallites: thickness is ~ 1 nm whereas lateral size ranges from submicrometers to ~ 100 μm . In addition, these nanosheets are obtained as negatively charged crystallites that are dispersed in a colloidal suspension. These aspects make the nanosheets a suitable building block for designing nanostructured films. In practice, colloidal nanosheets can be organized into various nanostructures or combined with a range of foreign materials at the nanometer scale by applying wet-process synthetic techniques involving flocculation and layer-by-layer (LbL) self-assembly. Through these processes, oxide nanosheets can be combined with a wide range of polyions such as organic polyelectrolytes, metal complexes, clusters and even oppositely charged nanosheets, which is a major advantage of this approach. Furthermore, control of particulate shape as thin flakes and hollow spheres has been achieved through freeze- or spray-drying techniques.

One of the highlights is the fabrication of nanocomposite films of organic polymer/nanosheet materials that exhibit useful properties. Several groups have demonstrated that the electrostatic LbL self-assembly *via* sequential adsorption and Langmuir-Blodgett (LB) procedure are effective for this purpose (Fig. 4).

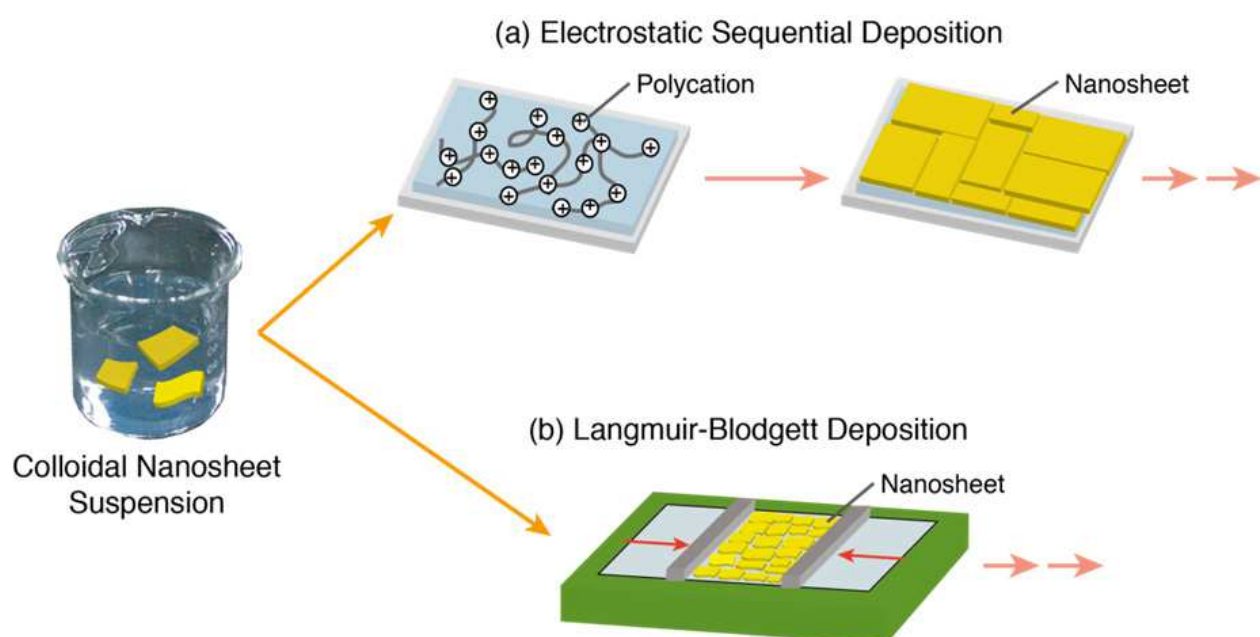


Fig. 4. Schematic illustration for chemical nanomanipulation of oxide nanosheets. (a) Electrostatic sequential deposition and (b) Langmuir-Blodgett deposition.

Electrostatic sequential deposition is one of the most powerful methods of fabricating nanostructured multilayer films with precisely controlled composition, thickness and architecture on a nanometer scale [Fig. 4(a)]. This technique, often called “molecular beaker epitaxy”, has been first developed by Decher⁴¹ and applied to various charged materials. In this LbL process, a multilayer assembly can be built up by alternately dipping the substrate in a colloidal suspension of nanosheets and an aqueous solution of suitable polyelectrolytes. Polycations such as poly (diallyldimethylammonium chloride) (PDDA) and poly (ethylenimine) are usually used as a counterpart of the oxide nanosheets. Fig. 5(a) depicts an

example of the multilayer film of $(\text{PDDA}/\text{Ti}_{0.91}\text{O}_2)_{10}$ on a quartz glass substrate, showing UV-visible absorption spectra in the fabrication process. The absorption peak at 265 nm, attributable to the $\text{Ti}_{0.91}\text{O}_2$ nanosheets, was progressively enhanced as the number of deposition cycles increased, clearly indicating the repeated adsorption of nearly equal amounts of nanosheets. XRD data provides further evidence for the formation of multilayer films by the evolution of Bragg peaks and their progressive enhancement. Other characterizations by ellipsometry, FT-IR and AFM all support the growth of multilayer nanocomposite films. Multilayer films of other nanosheets were fabricated by similar procedures.

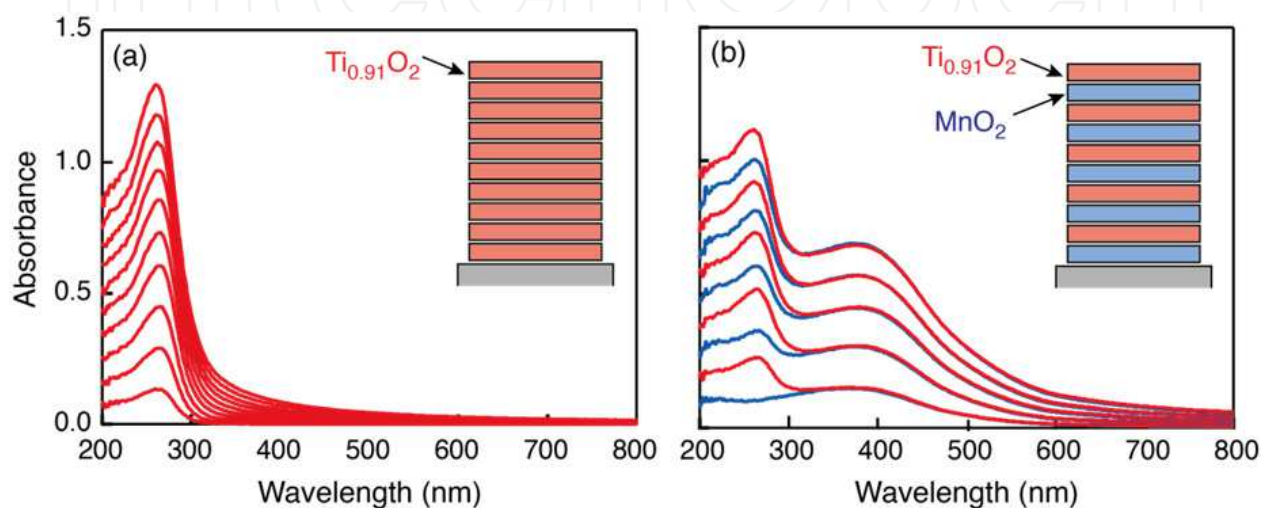


Fig. 5. UV-visible absorption spectra in the multilayer buildup processes for $(\text{Ti}_{0.91}\text{O}_2/\text{PDDA})_{10}$ (a) and $(\text{PDDA}/\text{Ti}_{0.91}\text{O}_2/\text{PDDA}/\text{MnO}_2)_{10}$ (b). The insets indicate the designed stacked structures of the nanosheets.

Such LbL assembly of various nanosheets also allows us to tailor superlattices or heterostructures by tuning the number of nanosheets and their stacking sequences. Fig. 5(b) shows UV-visible absorption spectra for the superlattice assembly composed of MnO_2 and $\text{Ti}_{0.91}\text{O}_2$ nanosheets. The observed spectral changes clearly indicate that the films grew as designed. The superlattice approach makes it possible to design complex functions that cannot be achieved using a single material.

LB deposition has been proved much simple and effective as another approach for organizing 2D nanosheets [Fig. 4(b)].⁴²⁻⁴⁵ LB film deposition, the formation of a floating monolayer on water surface in a Langmuir trough followed by an appropriate level of compression, is preferable for achieving dense packing or neat tiling. Through vertical-dipping/lifting, the monolayer is deposited onto a flat substrate in LbL fashion. Pioneering work⁴² has demonstrated that exfoliated nanosheets could float by adhering to amphiphilic ammonium cations at the air/water interface through electrostatic interaction, and thus the ordinary LB procedure is applicable for fabricating nanosheet films. Although LB technique has been used for decades, its application for nanoparticles and nanorods is often frustrated by defects ranging from pinholes to larger reorganization of the layers. In the case of nanosheets, the LB technique provides nearly perfect mono- and multilayer films with atomically flat surfaces. The LB-based LbL approach with the use of an atomically flat substrate is effective for fabricating atomically uniform and highly dense nanofilms of oxide nanosheets. Fig. 6 shows a cross-sectional high-resolution TEM image of a 5-layer (7.5 nm

thick) $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ film on a SrRuO_3 substrate.⁴⁶ The image clearly reveals a stacking structure corresponding to the LbL assembly of nanosheets. Such LB-deposited nanofilms are very suitable for a number of applications in electronic devices.

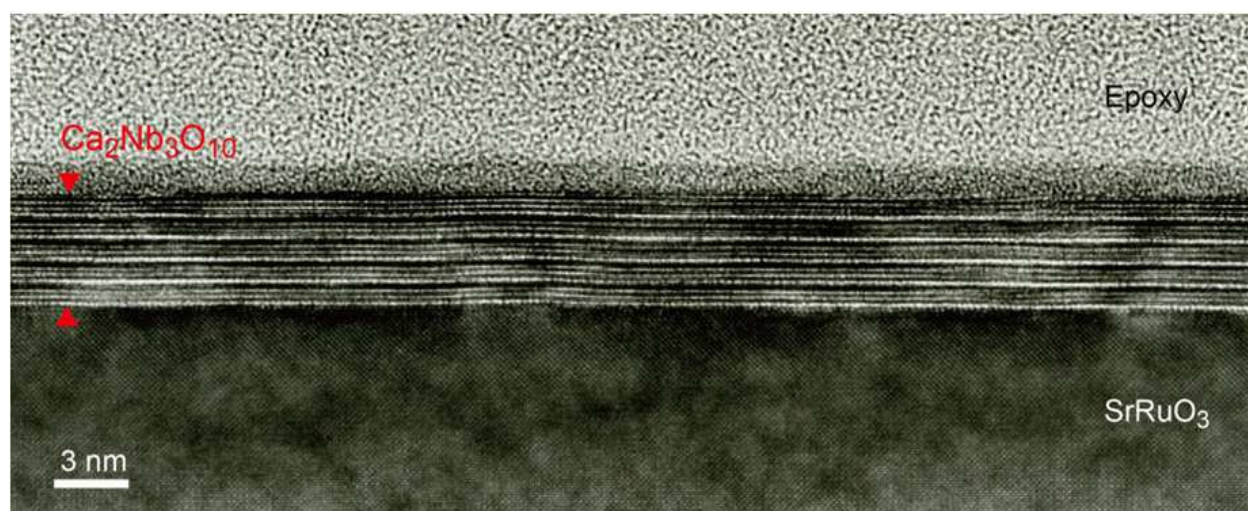


Fig. 6. Cross-sectional high-resolution TEM image of a 5-layer (7.5 nm thick) $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ film on a SrRuO_3 substrate. Note that the film/substrate interface is atomically flat without an interfacial layer between $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ and SrRuO_3 substrate. The nanofilms of this quality show an excellent dielectric property as will be discussed in section 4.1.

A clear benefit of these LbL approaches is the engineering of the clean interface, which appears to be a key step in the design of film properties. Currently, physical depositions such as vapor deposition and laser ablation are the main methods of fabricating oxide films. These techniques, however, usually require a complex and difficult deposition process involving high-temperature postannealing ($> 600^\circ\text{C}$), which can cause degradation in the film-substrate interface arising from both nonstoichiometry and thermal stress. In that sense, the solution-based bottom-up fabrication using oxide nanosheets provides new opportunities for room-temperature fabrication of oxide nanoelectronics, while eliminating integration problems encountered in current film-growth techniques.

4. Electronic applications

The development of a wide range of nanosheets with various properties is very important in the design of nanodevices with sophisticated functionality. Currently, extensive effort is being made to develop oxide nanosheets with new physical and chemical properties. The range of applications of nanoassemblies could therefore be widened significantly. Here, we describe the current status of researches on oxide nanosheets, highlighting emerging functionalities in electronic applications.

4.1 Electronic devices

In nanosheets, 2D structures created by lateral confinement can potentially lead to not only the modification of electronic structures but also the modulation of electron-transport phenomena that arise from the quantum confinement effect. Research in such exotic 2D systems recently intensified as a result of emerging progress in graphene and its novel

functionalities.^{1, 2, 47} In graphene, a number of unique conducting phenomena have already been found, such as anomalous quantum Hall effect, bipolar supercurrent, etc.

Despite the similar 2D structural nature, oxide nanosheets are quite different electronically (Table 1). Most oxide nanosheets synthesized to date are d^0 transition metal oxides (with Ti^{4+} , Nb^{5+} , Ta^{5+} , W^{6+}), where the empty d orbitals of metal mix with the filled p orbitals of the ligands.⁴⁸ Such d^0 materials are not electronically interesting, but semiconducting or insulating materials. In current research on oxide nanosheets, experimental efforts have thus focused on their use as a semiconducting host or dielectric.

$\text{Ti}_{0.91}\text{O}_2$ nanosheets possess semiconducting properties similar to those of bulk TiO_2 , such as rutile and anatase except for some modifications due to size quantization.⁴⁹ $\text{Ti}_{0.91}\text{O}_2$ nanosheets generate anodic photocurrent by band gap excitation under light irradiation with wavelengths shorter than 320 nm, corresponding to wider band gap energy of 3.8 eV.⁵⁰ In contrast, MnO_2 nanosheets have a broad absorption peak centered at 372 nm, which results from $d-d$ transitions in the MnO_2 nanosheets.¹³ Various interesting and useful properties have also been developed by organizing or assembling these oxide nanosheets into composite materials or multilayer films. $\text{Ti}_{0.91}\text{O}_2$ nanosheets flocculated with lanthanide cations emitted intense photoluminescence at room temperature through effective energy transfer from the semiconducting nanosheet host.^{15, 51} Highly stable photoinduced charge separation was attained in a composite film of restacked $\text{Ti}_{0.91}\text{O}_2$ nanosheets and mesoporous silica or clay minerals, in which electron donors and acceptors are spatially separated at a distance of micrometers.

Another enticing possibility is the use of oxide nanosheets in high- κ dielectrics, a key material for future semiconducting technology. For example, $\text{Ti}_{0.87}\text{O}_2$ and $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ nanosheets act as high- κ nanoblocks, and their multilayer assemblies exhibit low leakage current density ($<10^{-7} \text{ A cm}^{-2}$) with a high dielectric constant of >100 even for thicknesses as low as 10 nm.^{43, 46, 52-55} Fig. 7 summarizes the ϵ_r values for oxide nanosheets and various high- κ oxides. In the ultrathin region ($< 20 \text{ nm}$), the ϵ_r values of $\text{Ti}_{0.87}\text{O}_2$ and $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ nanosheets are larger than the values reported for any other high- κ materials. It should be noted that the high ϵ_r values of $\text{Ti}_{0.87}\text{O}_2$ and $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ nanosheets persist even in the $< 10 \text{ nm}$ region, which is in sharp contrast to a size-induced dielectric collapse in $(\text{Ba}_{1-x}\text{Sr}_x)\text{TiO}_3$.^{56, 57} These results suggest that $\text{Ti}_{0.87}\text{O}_2$ and $\text{Ca}_2\text{Nb}_3\text{O}_{10}$ are a very promising candidate for high- κ applications such as high-density capacitors and gate dielectrics.

Oxide nanosheets are reported to be an excellent material for electric batteries. In particular, owing to their unique 2D morphology, it is expected that laterally confined 2D nanosheet crystals can significantly enhance the host capabilities of active electrode materials. $\text{RuO}_{2.1}$ nanosheets showed high performance as electrochemical supercapacitors.²⁸ Reassembled $\text{Ti}_{0.91}\text{O}_2$ or MnO_2 nanosheets, either with or without carbon, are reported to have as large a capacity as Li-ion batteries.^{14, 58-60} Multilayer films of MnO_2 nanosheets prepared on ITO substrate also exhibited electrochromic behavior associated with the electrochemical redox process between Mn^{3+} and Mn^{4+} .⁶¹ This electrochromic efficiency is estimated to be $64.2 \text{ cm}^2 \text{ C}^{-1}$ at 385 nm, which is a relatively high value among manganese oxides.

An alternative route to nanosheet-based electronics is to consider oxide nanosheet not as an active component for nanodevices but as a seed layer, in which 1-nm-thick monolayer films can be used for design and orientation control of crystal films.⁶²⁻⁶⁴ The idea is to exploit the advantage of oxide nanosheets having high thermal stability even in ultrathin form. Such a technique is expected to have great potential for advances in thin-film technology.

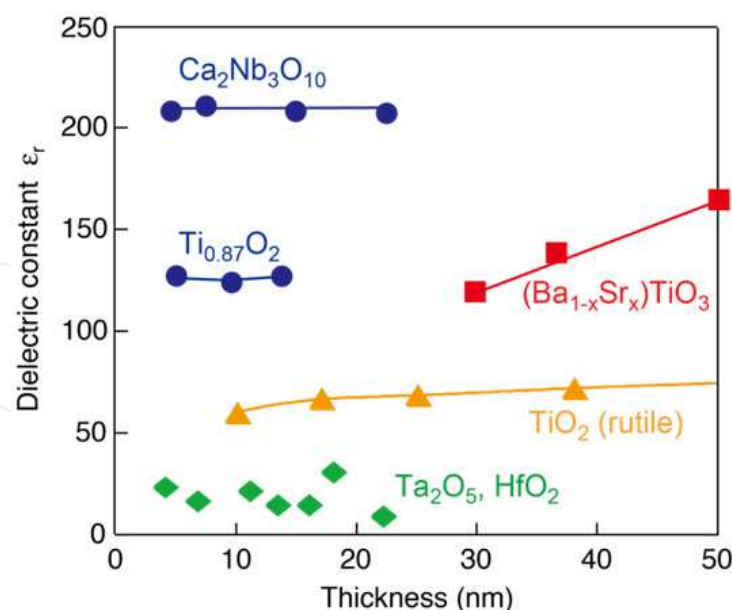


Fig. 7. The ϵ_r values for oxide nanosheets and various high- κ oxides.

4.2 Spin-electronic devices

Recent interest in room-temperature (RT) ferromagnetic semiconductors and low-dimensional magnetic nanostructures (motivated by possible application in spin-electronic devices) has stimulated research in the synthesis and characterization of TiO_2 nanosheet based materials. Titania nanosheets substituted with magnetic elements (Co, Fe, Mn) are ferromagnetic at room temperature.⁶⁵ The magnetization of $\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2$ nanosheets is anisotropic due to the 2D nature, and a maximum magnetic moment of $1.4 \mu_B/\text{Co}$ for $H //$ film is obtained, which is greater than the spin moment of $1\mu_B/\text{Co}$ theoretically expected for low-spin Co^{2+} as well as that in Co-doped anatase with semiconducting ($0.3 \mu_B/\text{Co}$) and insulating ($1.1 \mu_B/\text{Co}$) grounds.^{65, 66} Similar ferromagnetic properties have also been reported in a series of substituted and co-substituted titania nanosheets, including $\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2$, $\text{Ti}_{0.6}\text{Fe}_{0.4}\text{O}_2$,⁶⁷ $\text{Ti}_{0.8-x/4}\text{Fe}_{x/2}\text{Co}_{0.2-x/4}\text{O}_2$ ($0 \leq x \leq 0.8$)⁶⁸, and $\text{Ti}_{(5.2-2x)/6}\text{Mn}_{x/2}\text{O}_2$ ($0 \leq x \leq 0.4$)⁶⁹, and $\text{Co}_{1/3}\text{Al}_{2/3}(\text{OH})_2$. Spin-glass behavior was recently observed in the dried aggregate of tetramethylammonium (TMA)/ MnO_2 nanosheets, in which the geometrical frustration was caused by the triangular arrangement of the mixed-valence $\text{Mn}^{4+}/\text{Mn}^{3+}$ ions in the MnO_2 layer.⁷⁰

Concerning applications, ferromagnetic nanosheets have become a pivotal architectural element in magneto-optical (MO) and magneto-electronic devices, because low-dimensional nanostructures make use of the advantage offered by spin-polarized electrons and realize the integration of ferromagnetic materials into nanoelectronics. Indeed, the 2D nature of the electronic state of ferromagnetic nanosheets leads to a gigantic MO response, superior to that of bulk systems (Fig. 8). Multilayer films of $\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2$ and $\text{Ti}_{0.6}\text{Fe}_{0.4}\text{O}_2$ nanosheets exhibited a robust MO effect ($\sim 10^4 \text{ deg cm}^{-1}$) near the absorption edge at 280 nm, the shortest operating wavelength attained so far.⁶⁷ More interestingly, alternating stacking $(\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2/\text{Ti}_{0.6}\text{Fe}_{0.4}\text{O}_2)_5$ caused a strong enhancement in MO response ($\sim 3 \times 10^5 \text{ deg cm}^{-1}$) at 400–550 nm, which stems from the interlayer $d-d$ transitions ($\text{Co}^{2+}-\text{Fe}^{3+}$) between adjacent nanosheets.⁶⁷ A similar MO response ($\sim 2 \times 10^5 \text{ deg cm}^{-1}$) at 400–750 nm was also observed in (Co/Fe)-cosubstituted titania nanosheets, $\text{Ti}_{0.8-x/4}\text{Fe}_{x/2}\text{Co}_{0.2-x/4}\text{O}_2$ ($x = 0.2, 0.4, 0.6$).⁶⁸ These

MO materials are also important from a practical viewpoint as a key component for optical isolators in optical communication and data storage devices. In particular, such a large MO response including the blue light region offers potential for short-wavelength MO applications.

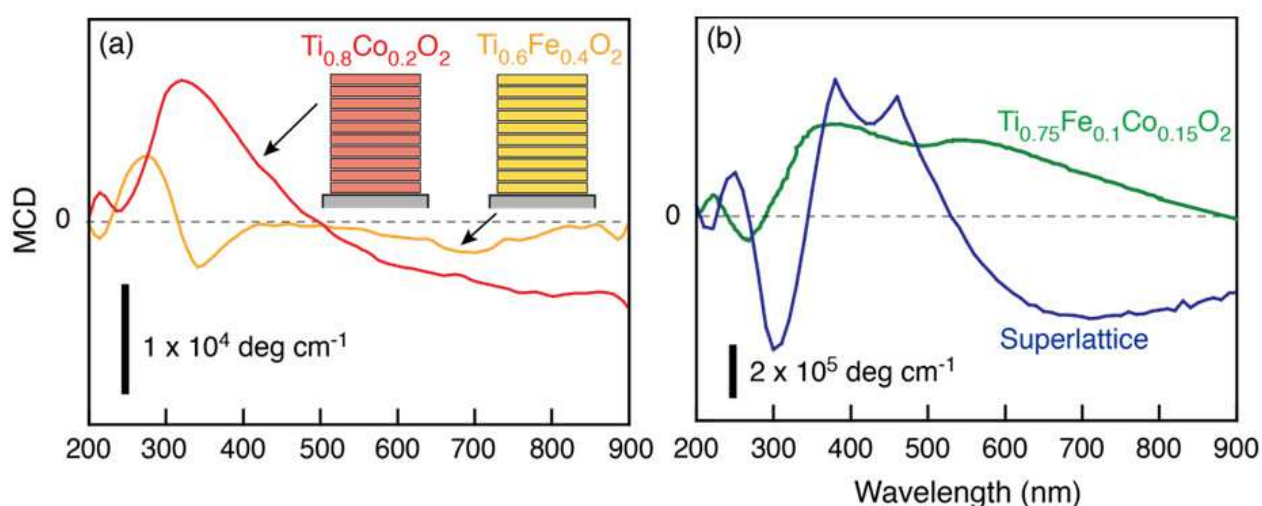


Fig. 8. (a) Magneto-optical spectra for multilayer assemblies of $(\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2)_{10}$ and $(\text{Ti}_{0.6}\text{Fe}_{0.4}\text{O}_2)_{10}$. (b) Magneto-optical spectra for $(\text{Ti}_{0.8}\text{Co}_{0.2}\text{O}_2/\text{Ti}_{0.6}\text{Fe}_{0.4}\text{O}_2)_5$ superlattice and $(\text{Ti}_{0.75}\text{Fe}_{0.1}\text{Co}_{0.15}\text{O}_2)_{10}$. We used magnetic circular dichroism (MCD) spectroscopy for the characterization of nanosheets. The MCD spectra were measured at RT on the basis of the difference in the absorption of right and left circularly polarized light. 1° of MCD corresponds to a 7% difference of optical absorption.

These ferromagnetic nanosheets are also a model experimental system for future spintronics studies, and their assembly has great potential for the rational design and construction of complex nanodevices, even combined with transparent electronics and molecular devices. Although we focus here only on MO devices, the assembled structure is naturally viewed as a tunnel junction, which could obviously be used in novel devices such as spin-tunneling switches, spin valves and optical interconnectors.

5. Conclusion

The current status of research on oxide nanosheets was reviewed. A variety of physical and chemical properties of oxide nanosheets have been developed to functionalize nanosheets for electronic and spin-electronic applications, and further studies will yield new information on their physics.

2D nanosheets also teach us how to handle and process 2D nanomaterials and develop nanotechnology in general. Although we have focused here only on high- κ properties in oxide nanosheets, 2D nanosheets exist in a whole class of functional materials, including metals, semiconductors, ferromagnetic, redox-active, photoluminescence, photochromic etc. 2D nanosheets with having regulated 2D would create the unconventional interactions of electrons as well as the confinements of electrons and ions inside the 2D nanospace or quantum well. Through new chemical design of 2D nanosheets, we can expect new or unprecedented functionalities in the 2D confined system. Furthermore, we can utilize

nanosheet-based LbL technology as a new tool to develop advanced fusion functions by promoting the cooperative interaction between organized components, which are difficult to attain with the current synthetic techniques and thin-film technologies.

Oxide nanosheets provide an ideal model to study phenomena in 2D systems. Previously restricted to theoretical study, 2D nanomaterials with their exotic properties are now open to experimentation using the individual 2D system. Graphene has already been found to exhibit a number of unique phenomena such as anomalous quantum Hall effect, bipolar supercurrent, half-metallic, etc. Although current experimental and theoretical efforts mainly focus on graphene, similar properties may be available with oxide nanosheets. We hope that all aspects described here demonstrate the great potential of oxide nanosheets, introducing more exciting physics and wide-ranging applications.

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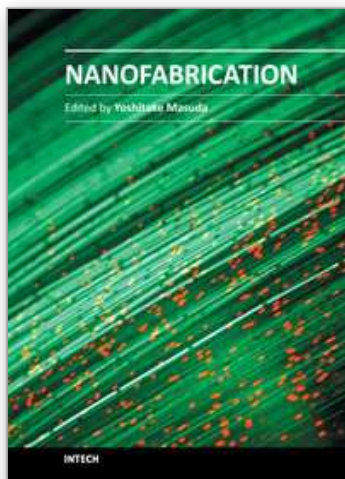
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