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# High Frequency Properties of Carbon Nanotubes and their Electromagnetic Wave Absorption Properties

Mangui Han and Longjiang Deng

*State Key Laboratory of Electronic Thin Films & Integrated Devices,  
University of Electronic Science and Technology of China,  
China*

## 1. Introduction

Since the discovery of carbon nanotubes (CNT) in 1991 by Iijima (Iijima, 1991), the intensive investigations on CNT have shown that CNT has a series of unique mechanical, electrical, magnetic, optical and thermal properties (Poncharal et al, 1999; Mintmire et al, 1992; Wang, et al, 2005; Xu et al, 2006; Zhao et al, 2006). For a completely pure, defect-free state CNT, it is diamagnetic with a large diamagnetic susceptibility spanning more than two orders of magnitude. Generally, there are two categories of CNTs available for various applications: single wall carbon nanotubes (SWCNT) and multiwalled carbon nanotubes (MWCNT). MWCNT consists of two or more concentric cylindrical shells of graphene sheets coaxially arranged around a central hollow core. SWCNT comprises a single graphene cylinder. Such a tube configuration has a so-called electronic conjugate  $\pi$ -structure which is responsible for its unique electronic transport behaviors. CNTs have been reported to find new applications in bolometers (Lu et al, 2010); gas pressure sensors (Tooski et al, 2011), nanometer resolution optical probes (Nakata et al, 2011), etc. Recently, there is an increasing interest in the developing materials for electromagnetic engineering applications, such as electromagnetic wave absorbers, antennae, circulators, etc. Especially, as telecommunication systems and electronic devices such as wireless local area networks and antenna system are ushered into the GHz arena, electromagnetic interference (EMI) problems that degrade the performance of electric circuits and communication systems are becoming more and more serious. One solution to these problems is to absorb the unwanted electromagnetic wave using microwave absorbing materials (Han et al, 2010a, 2010b, 2009a, 2009b; Xie et al, 2007; Chen et al, 2010). Since the application of conventional microwave absorbers such as spinel ferrites is restricted in the MHz region due to their Snoek's limit (Snoek, 1948), researches are urged to explore new types of microwave absorbers that can meet requirements like light weight and strong absorption in a wide frequency range, such as ferrites, metallic nanostructured magnetic materials (such as Fe, Co, Ni and alloys made from them), magnetic microwire, etc. The significant drawback of these materials is their large density, especially for ferrites, which are the most widely used materials for electromagnetic wave absorbers. For instance, the density of ferrites is usually around 4~5 g/cm<sup>3</sup>. A CNT/epoxy resin composite is considered a good candidate material for microwave applications due to its lower density

(for instance  $0.14 \text{ g/cm}^3$ ), large dielectric loss, such as antireflection, electromagnetic interference shielding or microwave absorbing. Wu et al (Wu et al, 2004) found the MWCNT/epoxy resin composites exhibited large relative permittivity over a broad bandwidth, and the permittivity increased with the increase of the MWCNT contents in the composites. Kim et al (Kim et al, 2004) had found that the MWCNT/PMMA (polymethyl methacrylate) composites exhibited excellent EMI shielding performances in GHz range. Recently, the molecular electronics research which aims at combining spintronics with molecular structures has been motivated to study the magnetic proximity effect in carbon nanotubes, where a measurable induced magnetic moment has been observed when CNTs are contacting with magnetic materials, especially with half metal magnetic materials, for instance  $\text{Fe}_3\text{O}_4$ . The induced magnetic moment was believed due to the spin polarized charge transfer between CNTs and magnetic material (Cespedes et al, 2004; Coey et al, 2002). Except for this induced magnetism, there are always some nano-sized residual magnetic metals (Fe, Co or Ni) left on CNT during the CNT manufacture process, and enable the CNT to exhibit a weak ferromagnetism. A magnetoresistance effect also has been found in carbon nanotube (Stamenov et al, 2005). None the less, there are many other unexploited application areas for the carbon nanotube-based system. For instance, whether there is an interaction between the ferrite and the MWCNT; If such an interaction exists between the ferrite and the MWCNT, whether the dynamic magnetic properties (such as resonance frequency) of ferrite can be tuned by doping with the MWCNT; As we know, electromagnetic wave absorbers work by dissipating electromagnetic energy into heat via magnetic loss and dielectric loss. Spinel ferrites are one of the frequently used electromagnetic wave absorbing materials, but they suffer from the lower spin rotation resonance compared with other ferrites, such as hexagonal ferrites. In this chapter, we will firstly discuss the static magnetic properties of MWCNTs and SWCNTs and their origins, then their electromagnetic wave properties. Finally, we will study the doping effect of MWCNTs on the electromagnetic properties of NiCoZn spinel ferrites.

## 2. Experimental details

The MWCNTs and SWCNTs were purchased from a vendor (TimesNano, Inc.<sup>®</sup>, Chengdu, China) and the MWCNTs were prepared by a chemical vapor deposition method using Ni as the catalyst. While SWCNTs were prepared using the same technique but Co was used as the catalyst. The average length of MWCNT is about  $50 \mu\text{m}$ . The outer diameter of MWCNTs is about  $10\sim 30 \text{ nm}$ . The purities of MWCNTs and SWCNTs have been checked by the energy dispersive x-ray (EDX) measurements. The content of Ni impurity in MWCNTs is about  $0.5 \text{ wt. \%}$ . The Co impurity in SWCNTs is about  $0.2 \text{ wt. \%}$ . The images of MWCNTs were taken by a transmission electronic microscope (TEM). Field emission scanning electron microscopy (FE-SEM) images also have been taken for both SWCNTs and MWCNTs. In order to study the high frequency properties of MWCNTs and SWCNTs, toroidal samples with the inner, outer diameter and thickness as  $3, 7$  and  $3.7 \text{ mm}$  respectively were prepared by mixing the MWCNTs (SWCNTs) and wax for microwave measurements. The weight ratio of CNTs over wax is  $0.2$ . The dependences of permittivity on frequency were theoretically studied based on the Cole-Cole law. With the obtained permeability and permittivity values, their electromagnetic wave absorption properties were evaluated and compared. To show the effect of CNTs on tuning the electromagnetic properties of ferrites, we have synthesized spinel ferrites ( $\text{Ni}_{0.4}\text{Co}_{0.24}\text{Zn}_{0.36}\text{Fe}_2\text{O}_4$ ) with average particle size of  $2\sim 3$

$\mu\text{m}$  by sintering the oxide mixtures of  $\text{Fe}_2\text{O}_3$ ,  $\text{NiO}$ ,  $\text{ZnO}$ , and  $\text{Co}_2\text{O}_3$  at  $1250^\circ\text{C}$  for 2 h. The crystal structure of prepared ferrite checked by x-ray diffraction shows a typical spinel structure. We have prepared three samples to study the doping effect of MWCNTs on the electromagnetic properties of ferrites. Sample 0 has not been doped with MWCNTs. Sample 1 contains 5.2 wt. % MWCNTs. Sample 2 contains 10.4 wt. % MWCNTs. For these three ferrite samples, the weight percentage of wax is about 16%. The ferrite, MWCNT and wax are carefully mixed to ensure that the samples are homogenous and isotropic. The high frequency properties were also measured. All the high frequency measurements were carried out on an Agilent Vector Network Analyzer 8720 based on the transmission/reflection method. During a high frequency measurement, a sample under study was inserted into a segment of coaxial transmission line. All four scattering parameters can be obtained from the measurements.

$$S_{11} = R_1^2 \frac{\Gamma(1 - T^2)}{1 - \Gamma^2 T^2} \quad (1)$$

$$S_{22} = R_2^2 \frac{\Gamma(1 - T^2)}{1 - \Gamma^2 T^2} \quad (2)$$

$$S_{21} = S_{12} = R_1 R_2 \frac{\Gamma(1 - T^2)}{1 - \Gamma^2 T^2} \quad (3)$$

, where  $R_1$  and  $R_2$  are the reference plane transformations at two ports:  $R_i = \exp(-\gamma L_i)$  ( $i = 1, 2$ ),  $T$  is the transmission coefficient:  $T = \exp(-\gamma L)$ , where  $L$  is the thickness of sample under study. The reflection coefficient  $\Gamma$  can be given as:

$$\Gamma = \frac{\sqrt{\mu_r / \epsilon_r} - 1}{\sqrt{\mu_r / \epsilon_r} + 1} \quad (4)$$

Employing the Nicolson-Ross-Weir(NRW) algorithm, the complex relative permittivity and permeability can be derived from above equations, which are given as following:

$$\mu_r = \frac{1 + \Gamma}{(1 - \Gamma)\Lambda \sqrt{(1/\lambda_0^2) - (1/\lambda_c^2)}} \quad (5)$$

$$\epsilon_r = \frac{\lambda_0^2}{\mu_r [(1/\lambda_c^2) - (1/\Lambda^2)]} \quad (6)$$

, where  $\lambda_0$  is the free-space wavelength,  $\lambda_c$  is the cutoff wavelength of the transmission line section. For a coaxial line,  $\lambda_c = \infty$ .

$$\frac{1}{\Lambda^2} = -\left[\frac{1}{2\pi D} \ln\left(\frac{1}{T}\right)\right]^2 \quad (7)$$

More detailed descriptions on this measurement principle can be found in Ref. (Chen et al, 2004).

### 3. Morphologies of MWCNTs and SWCNTs

TEM images have been taken for both MWCNTs and SWCNTs, as shown in Fig. 1. Tabular structures can be seen in these images. FE-SEM images have also been taken for MWCNTs and SWCNTs, which are shown as Fig. 2(a) and (b). The diameter of SWCNTs is found to be around 18.4 nm. The diameter of MWCNTs is about 23.9 nm.

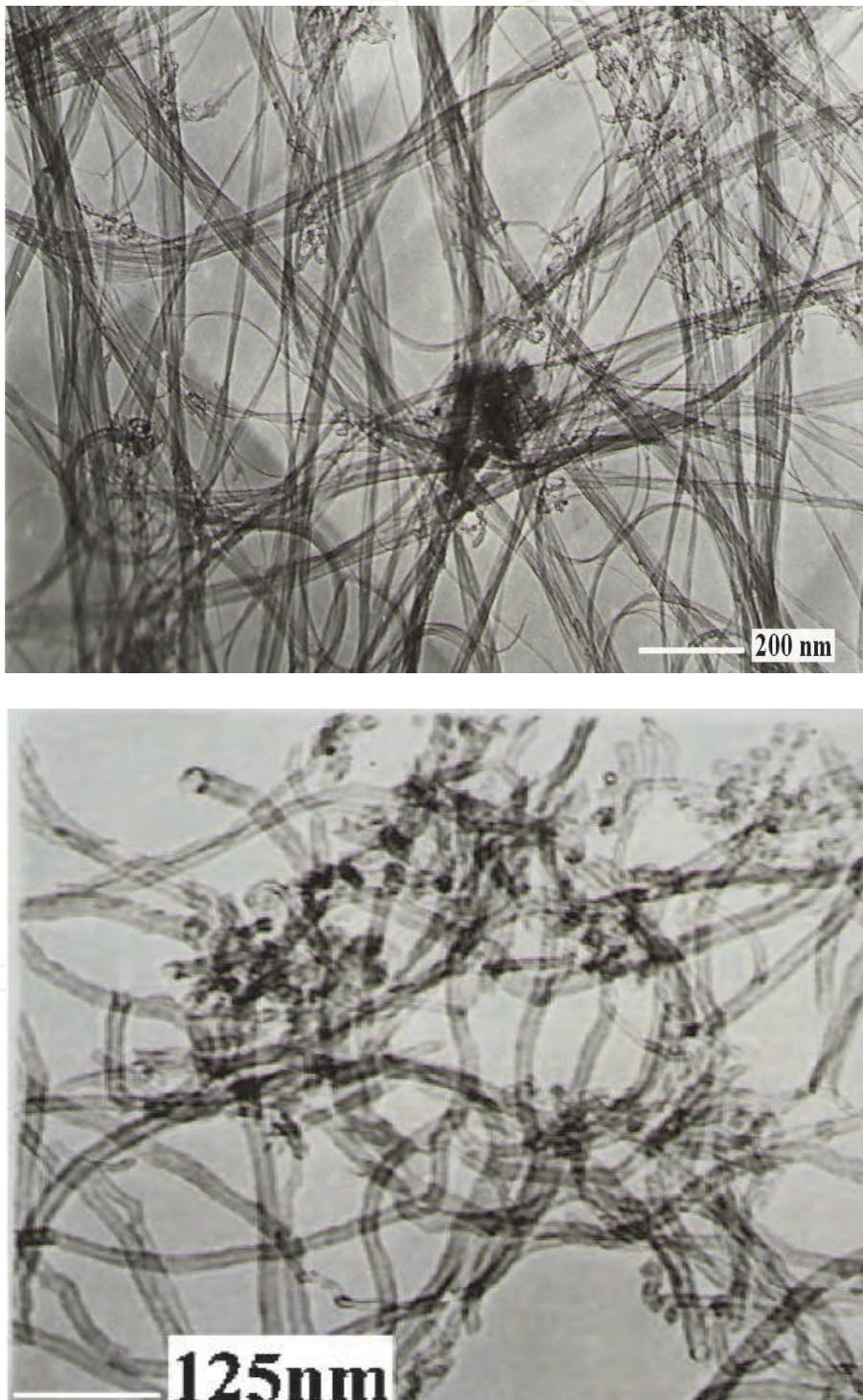


Fig. 1. TEM images of SWCNTs (a) and MWCNTs (b)

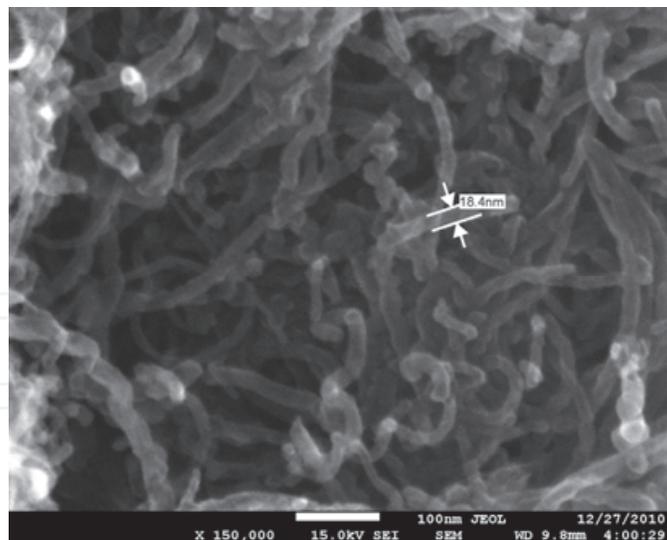


Fig. 2 (a). FE-SEM of SWCNTs

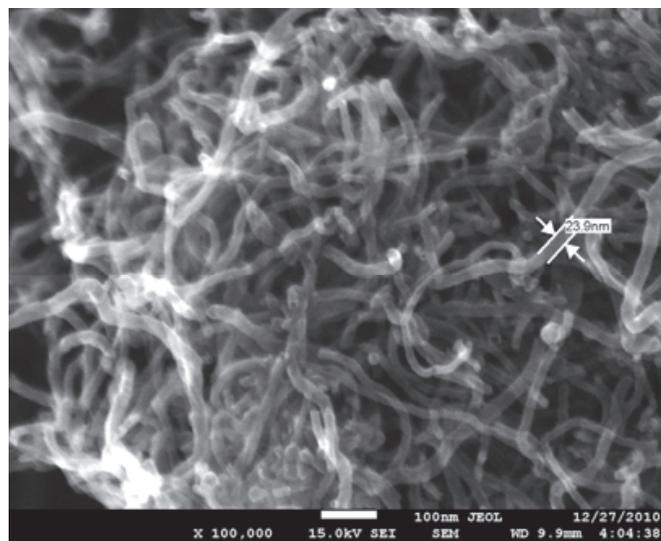


Fig. 2 (b). FE-SEM of MWCNTs

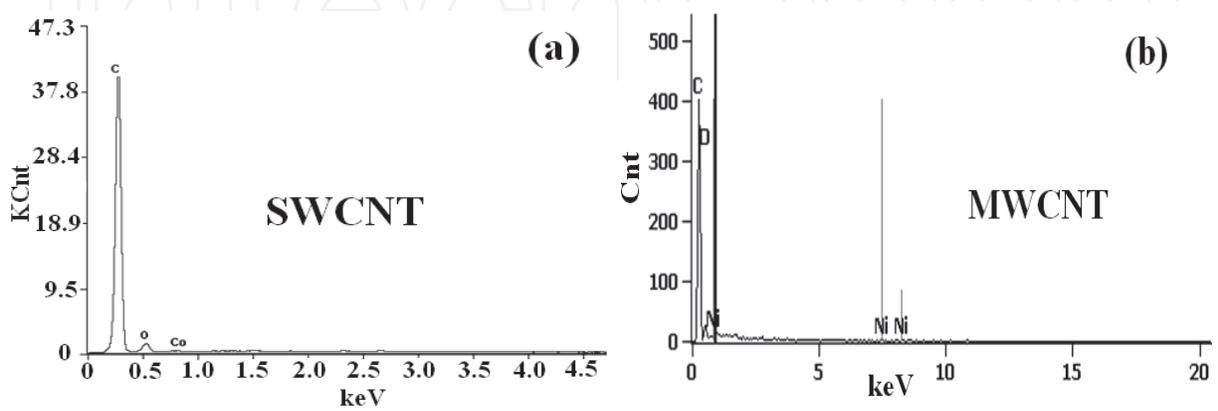


Fig. 3. EDX of SWCNTs in (a) and MWCNTs in (b)

#### 4. The static magnetic properties of MWCNTs and SWCNTs

Usually, pure SWCNTs or MWCNTs should not exhibit ferromagnetic properties. However, many researchers have reported that CNTs exhibit weak ferromagnetic properties (Tang et al, 2008; Zhang et al, 2000; Kudo, et al, 2007).

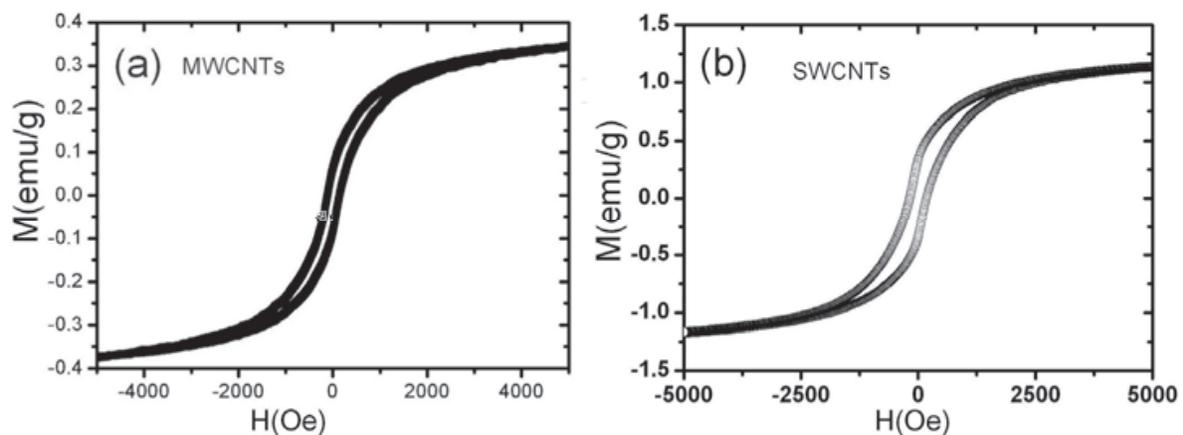


Fig. 4. Magnetic hysteresis loops of MWCNTs (a) and SWCNTs (b)

In this study, we have carried out the measurements of static magnetic properties. The measurement results are shown in Fig. 4(a) and (b). The results show that both MWCNTs and SWCNTs exhibit ferromagnetic properties. Especially, SWCNTs have stronger magnetic response in term of saturation magnetization ( $M_s$ ). For SWCNTs,  $M_s$  value is about 1.17 emu/g. For MWCNTs,  $M_s$  is about 0.35. The coercivity ( $H_c$ ) of SWCNTs is about 179.7 Oe. For MWCNTs,  $H_c$  value is about 127.5 Oe. Our EDS measurements show that there are a tiny amount of ferromagnetic substances can be found in MWCNTs or SWCNTs. For MWCNTs, the ferromagnetic impurity is nickel. For SWCNTs, the ferromagnetic impurity is cobalt. Therefore, we hypothesis that the observed weak ferromagnetic properties arise from the existence of ferromagnetic impurities in CNTs. The different static magnetic properties arise from the differences between Ni and Co.

#### 5. Electromagnetic properties of MWCNTs and SWCNTS

According to Ref. (Watts et al, 2003), there are many factors making contributions to the dielectric properties: dielectric relaxation, resonance, the motion of conduction electrons, defects in the nanotubes, length, diameters, chirality, etc. Therefore, it will be extremely difficult to describe the permittivity dispersion behaviors of CNTs when taking into account of all these factors. Here, we assume that there are only two mechanisms making major contributions to the permittivity dispersion of MWCNTs. One is the general permittivity behaviors of dielectric materials which can be described by the Cole-Cole model (Cole et al 1941; Li, 1990), such as electrons polarization, charges polarization, etc. The other one is due to the motion of conducting electrons (Leon et al, 2001; Wu et al, 2004; Schultz et al, 2003). For instance, the contribution to the permittivity dispersion due to the conductivities has also been considered for carbon nanotubes composites in Ref. (Leon et al, 2001) and Ref. (Wu et al, 2004). In this chapter, we will adopt the following equations to fit the permittivity dispersion spectra. Eqs. 8 and 9 represent the contributions from the polarization of electrons, charges polarization.

$$\epsilon' = \epsilon_{r\infty} + (\epsilon_{rs} - \epsilon_{r\infty}) \frac{1 + (\omega\tau)^{1-\alpha} \sin(\pi\alpha / 2)}{1 + 2(\omega\tau)^{1-\alpha} \sin(\pi\alpha / 2) + (\omega\tau)^{2(1-\alpha)}} \quad (8)$$

$$\epsilon'' = (\epsilon_{rs} - \epsilon_{r\infty}) \frac{(\omega\tau)^{1-\alpha} \cos(\pi\alpha / 2)}{1 + 2(\omega\tau)^{1-\alpha} \sin(\pi\alpha / 2) + (\omega\tau)^{2(1-\alpha)}} \quad (9)$$

where  $\epsilon_0$  is the permittivity in free space,  $\omega$  is the angular frequency,  $\epsilon_{r\infty}$  is the permittivity when frequency is infinity,  $\epsilon_{rs}$  is the static permittivity,  $\tau$  is the relaxation time,  $\alpha$  is the distribution of relaxation time ( $0 \leq \alpha \leq 1$ ), and a smaller  $\alpha$  value means a narrower distribution of relaxation time. The permittivity dispersion behaviors have been studied based on the Cole-Cole law, and are shown by the dotted line in Fig. 5(a) and Fig. 6(a). The fitting results are given in Table 1.

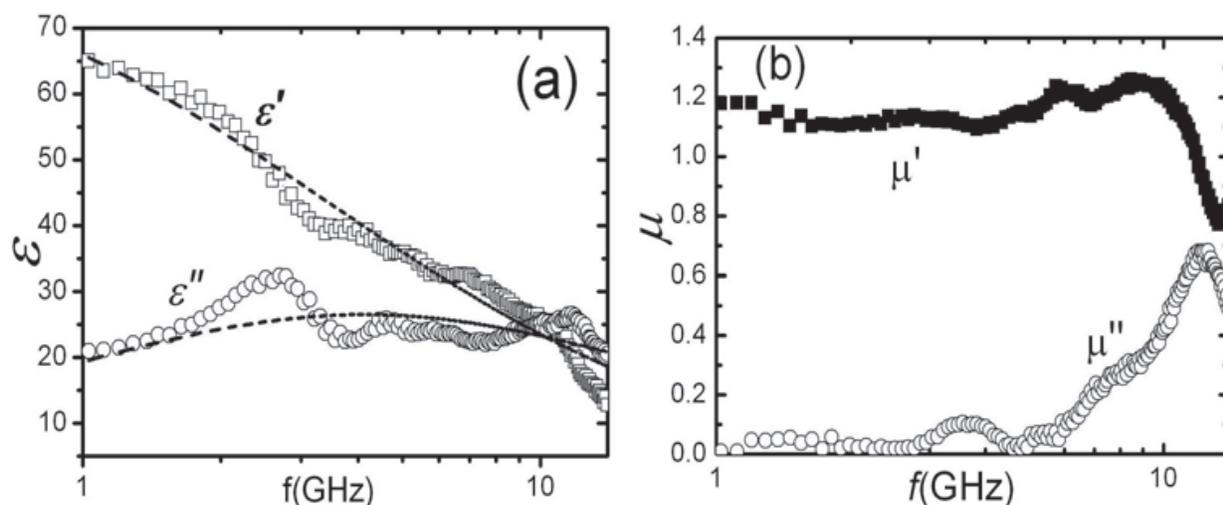


Fig. 5. High frequency permittivity and permeability dispersion spectra of SWCNTs-wax composites (SWCNTs: wax = 1: 5)

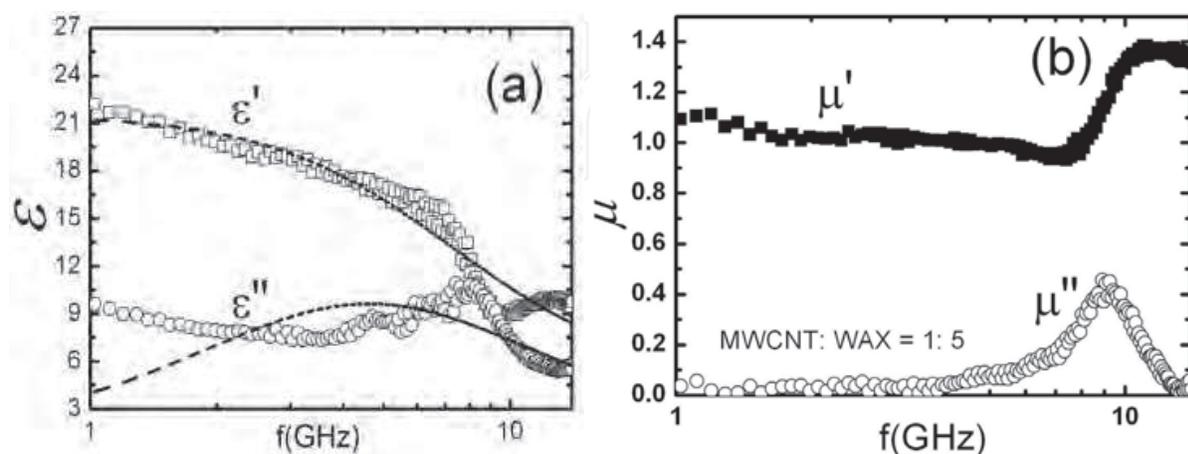


Fig. 6. High frequency permittivity and permeability dispersion spectra of MWCNTs-wax composites

Samples	$\tau$	$\alpha$	$\epsilon_{rs}$	$\epsilon_{r\infty}$
SWCNTs	$4.57 \times 10^{-11}$	0.31	86.56	$1.28 \times 10^{-11}$
MWCNTs	$2.20 \times 10^{-11}$	$3.24 \times 10^{-15}$	21.52	4.92

Table 1. Fitting results of permittivity spectra of MWCNTs and SWCNTs

As indicated in Table 1, in comparison to SWCNTs, MWCNTs have a smaller static permittivity value and smaller  $\alpha$  value which means MWCNTs have a narrower distribution of polarization relaxation time. The dependences of permeability on frequency have been shown in Fig. 5(b) and Fig. 6(b) for SWCNTs and MWCNTs respectively. A major dispersion peak has been seen in each graph, which might be due to the existence of ferromagnetic impurity. For SWCNTs, the major resonance peak is around 12.4 GHz, while the major resonance peak of MWCNTs is found to be about 8.9 GHz. As we mentioned, SWCNTs contain a small quantity of Co. MWCNTs contain a small quantity of Ni, see Fig. 3(a) and (b). According to the Snoek' law (Snoek J. 1948):  $(\mu_s - 1)f_r = (2/3)\gamma^2 4\pi M_0$ , where  $f_r$  is proportional to the magnetocrystalline field. Therefore, the observed differences in permeability dispersion behaviors arise from their difference static properties. Such as the magnetocrystalline anisotropy of Co is larger than that of Ni, which will result in SWCNTs will have a larger resonance frequency than MWCNTs. With the obtained values of permittivity and permeability of the material under study, usually we can then evaluate the performance of electromagnetic absorber made from this material. The electromagnetic wave absorbing property of an absorber are dependent on six parameters, such as the thickness ( $t$ ) of the absorber,  $\epsilon'$ ,  $\epsilon''$ ,  $\mu'$ ,  $\mu''$  and frequency ( $f$ ). The criterion of absorption performance usually is expressed by the reflection loss (RL) in dB unit. Based on the measured complex permittivity and permeability, and we assume that a single layer of carbon nanotubes/paraffin composite is attached on a metal plate, then the electromagnetic wave absorbing properties usually can be evaluated by the following equation(Naito et al, 1971; Maeda et al, 2004):

$$RL = 20 \log |(Z_{in} - Z_0) / (Z_{in} + Z_0)| \quad (10)$$

, where  $Z_0$  is the impedance of free space,  $Z_{in}$  is the input impedance of the absorber, which can be expressed as:

$$Z_{in} = Z_0 \sqrt{\mu_r / \epsilon_r} \tanh\{j(2\pi ft / c) \sqrt{\mu_r \epsilon_r}\} \quad (11)$$

, where  $c$  is the speed of light. Fig. 7 is a 3-dimensions graph which clearly shows the electromagnetic wave absorption properties of carbon nanotube/paraffin composite with different thickness within the measurement frequency range of 1-14 GHz. The color bar in Fig. 7 indicates different RL values in dB unit corresponding to different colors. Fig. 7(a) and (b) have shown the electromagnetic wave absorption properties of CNTs absorbers with different thickness (1-10 mm) within 1 - 14 GHz. It is clearly shown that MWCNTs have better electromagnetic wave absorption properties in terms of RL value. The minimum RL value of MWCNTs is about -20 dB. However, the minimum value of RL is -9.5 dB for SWCNTs. For many applications, the RL value smaller than -10 dB is accepted. A two-dimension contour map can be used to clearly show the electromagnetic wave absorption properties of absorbers with different thickness, which are given in Fig. 8(a). Learned from this map, the RL value of MWCNTs is less than -10 dB except for the

frequency range of 6-10 GHz and with the thickness ranging from 2.2 mm -3 mm. Especially, a thin MWCNTs absorber with a thickness of 1.62 mm can have an excellent absorption property (RL = -20 dB). Several examples showing the reflection loss of MWCNTs have been given for thickness of 1.62 mm, 2 mm, 4 mm and 6 mm, please see Fig. 8(b). The bandwidth with the RL value less than -10dB is respectively 0.56 GHz, 0.73 GHz, 1.90 GHz and 2.85 GHz for the thickness (t) as 6 mm, 4 mm, 2 mm and 1.62 mm. Compared with Fig. 8(b), Fig. 8(a) can give a straightforward presentation on the bandwidth of an absorber with different thickness.

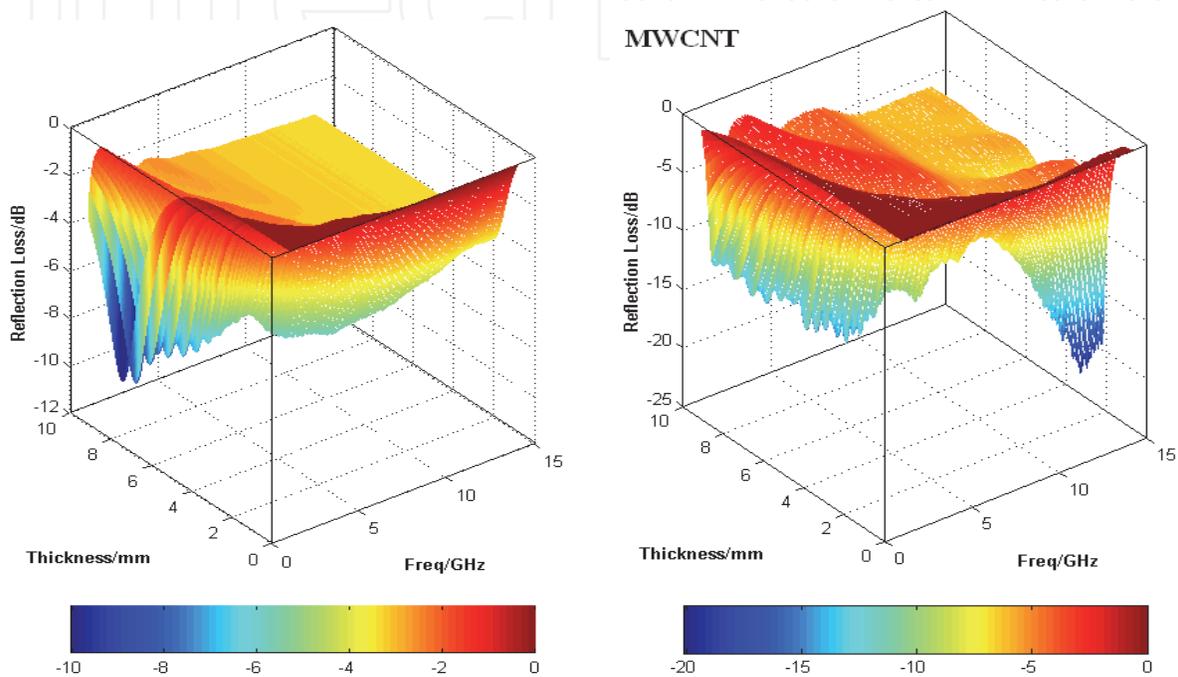


Fig. 7. (a) Reflection loss of SWCNTs (1:5 wax) ; (b) MWCNTs (1:5 wax)

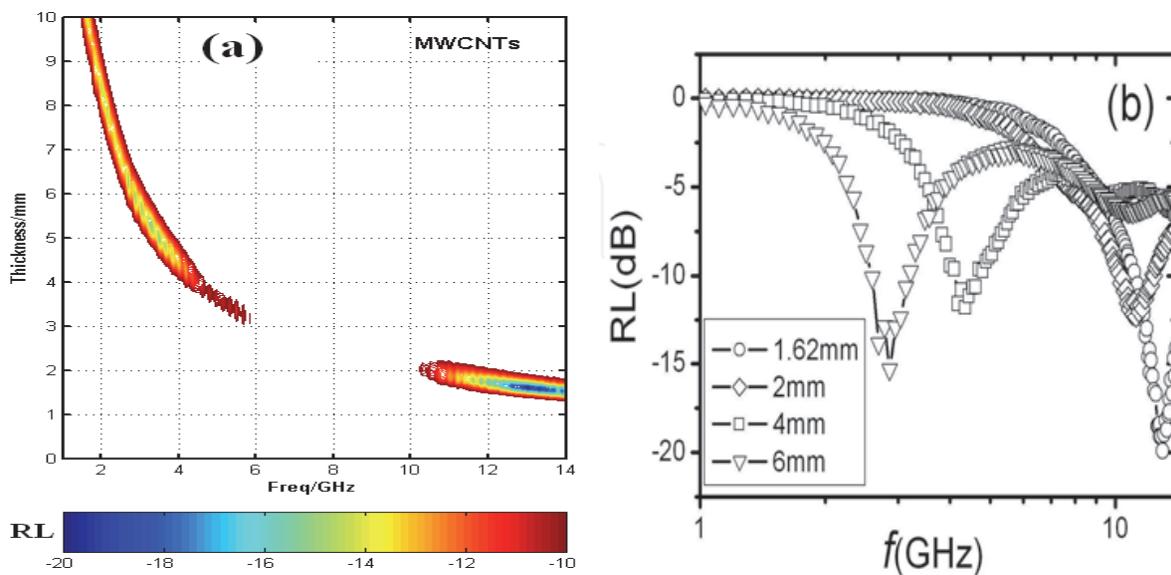


Fig. 8. (a) Contour map indicating the RL values less than -10 dB; (b) the RL~ f spectra of an absorber with different thickness values.

## 6. Doping effects of CNTs on electromagnetic properties of ferrites

Generally speaking, two well separated peaks are rarely observed in NiZn spinel ferrites in GHz range. However, as shown in Fig. 9(a), two peaks are found in the spectrum of  $\mu'' - f$  for sample 0 (NiCoZn ferrite / wax composite). For these two peaks, one is at 1.76 GHz, the other one is at 6.8 GHz. It is well accepted that there are two mechanisms responsible for the permeability dispersion of power ferrites. One is the domain wall motion, and the other one is the spin rotation. The resonance frequency of domain wall motion (i. e.  $f_{dw}$ ) is lower than the one for the spin rotation (i. e.  $f_r$ ). The permeability dispersion curve of pure MWCNT is shown in Fig. 6(b). The observed permeability of MWCNT is believed due to the residual nano-sized Ni particles on MWCNT. For sample 1, 2, the magnetic materials contributing to the permeability spectrum shape include ferrite and Ni. For such hybrid magnetic composites containing two ferromagnetic substances, the dispersion curve is extremely difficult to be expressed by superposing the magnetic spectrum of each magnetic material. Similar situation has been reported in permalloy-NiZn ferrite hybrid composite materials by Kasagi et al (Kasagi et al, 2004). Also, if the permeability dispersion curve is badly distorted, the dispersion curve would be very difficult to be fitted by assuming it comprises two component spectra: one is for domain wall movement; one is for spin resonance.

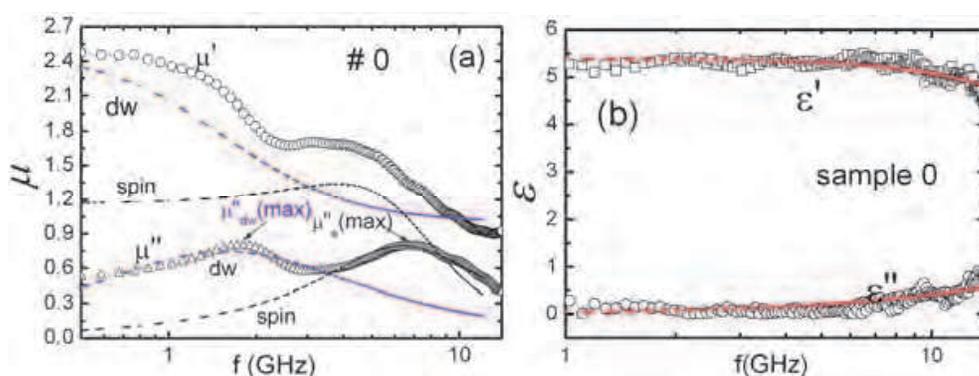


Fig. 9. High frequency properties of sample 0, (a) permeability spectra; (b) permittivity spectra, “*dw*” denotes the domain wall component. “*spin*” denotes the spin rotation component.

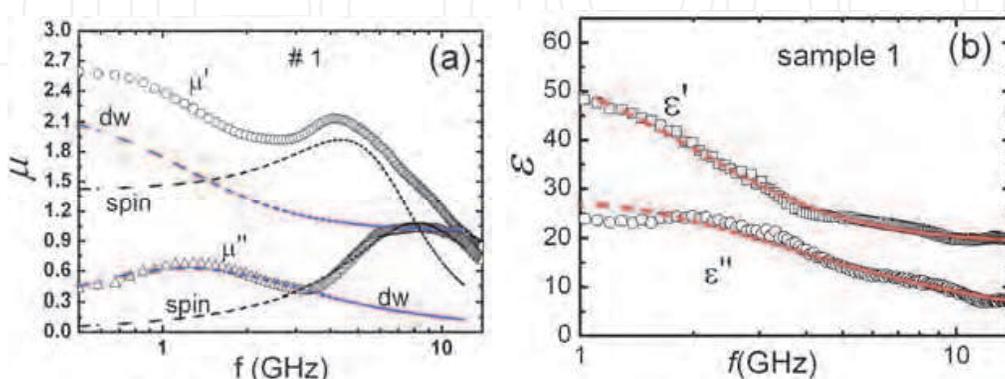


Fig. 10. High frequency properties of sample 1, (a) permeability spectra; (b) permittivity spectra.

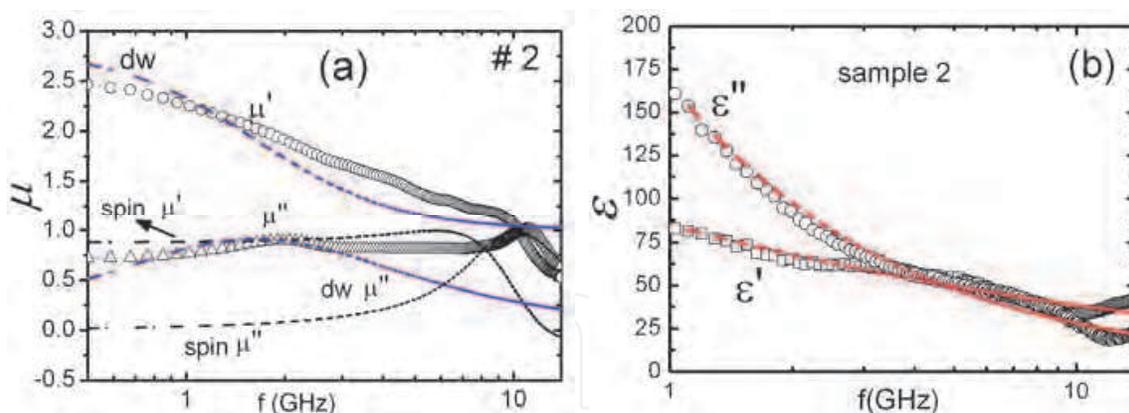


Fig. 11. High frequency properties of sample 2, (a) permeability spectra (b) permittivity spectra

Based on the spectrum shape, we learn that the spectra in Fig. 9(a) are the relaxation type because no negative  $\mu'$  values have been observed. Therefore, the resonances at lower frequencies in Fig. 9(a), 10(a) and 11(a) are due to the domain wall movement mechanism. The resonances at higher frequencies are due to the spin rotation mechanism. In Fig. 9(a),  $f_{dw}$  and  $f_s$  are 1.76 GHz, 6.80 GHz respectively for sample 0 without MWCNT. In Fig. 10 (a),  $f_{dw}$  and  $f_s$  are 1.28 GHz, 8.03 GHz for sample 1 with 5.2 wt% MWCNT. In Fig. 11 (a),  $f_{dw}$  and  $f_s$  are 1.98 GHz, 11.08 GHz for sample 2 with 10.4 wt% MWCNT. Clearly, the doping effect of MWCNT on the resonance frequency of spin rotation is much significant than that of domain wall movement. For the domain wall resonance with a relaxation type spectrum, the real part and imaginary part of permeability can be expressed by the Debye's dispersion law (Liao, 1988):

$$\mu'_{dw} = 1 + \chi_{d0} \frac{1}{[1 - (f / f_{dw})^2]} \tag{12}$$

$$\mu''_{dw} = \chi_{d0} \frac{f / f_{dw}}{[1 - (f / f_{dw})^2]} \tag{13}$$

, where  $\chi_{d0}$  is the static susceptibility for domain wall movement. The resonance frequency  $f_{dw}$  above can be expressed by physical parameters as (Liao, 1988):  $f_{dw} = (9 \pi \sigma_{wall} \rho) / (8 a \mu_0^2 Ms^2)$ , where  $\sigma_{wall}$  is the density of domain wall energy,  $a$  is the average distance of impurities in a ferrite particle which impede the domain wall movements.  $\rho$  is the electric resistivity of ferrite.  $Ms$  is the saturation magnetization. In our case, the ferrites in different composite samples are same. Hence,  $\sigma_{wall}$  can be considered as a constant. The particle size of ferrites is about 2~3  $\mu m$ . Multi domains can exist in each ferrite particle, so the parameter  $a$  can be assumed as a constant. Then, as shown in this equation,  $f_{dw}$  is dependent on  $\rho$  and  $Ms$ . It is well known that MWCNT is a substance with very low resistivity and weak ferromagnetism. Doped with MWCNT, both  $\rho$  and  $Ms$  of ferrite-wax composite will drop. Qualitatively speaking, for sample 1, we can conjure that the drop in  $\rho$  is stronger than the drop in  $Ms$ , hence  $f_{dw}$  is lower than the one for sample 0.

As for the spin resonance frequency  $f_s$ , it depends on (Rado,1953):  $f_s = (\gamma/2\pi) H_A$ , where  $\gamma$  is the gyromagnetic ratio,  $H_A$  is the effective magnetic anisotropic field. In our case, the

permeability spectrum due to the spin rotation mechanism is a relaxation type. Accordingly it can be expressed as (Tsutaoka 2003):

$$\mu'_s = 1 + \frac{\chi_{s0}\omega_s^2[(\omega_s^2 - \omega^2) + \omega^2\alpha^2]}{[\omega_s^2 - \omega^2(1 + \alpha^2)]^2 + 4\omega^2\omega_s^2\alpha^2} \quad (14)$$

$$\mu''_s = \frac{\chi_{s0}\omega_s\omega\alpha[\omega_s^2 + \omega^2(1 + \alpha^2)]}{[\omega_s^2 - \omega^2(1 + \alpha^2)]^2 + 4\omega^2\omega_s^2\alpha^2} \quad (15)$$

, where  $\chi_{s0}$  is the static susceptibility for spin rotation;  $\omega_s$  is the angular resonance frequency;  $\alpha$  is the damping factor for spin rotation.  $\omega$  is the angular frequency of alternating magnetic field.

Fitting the experimental curves in Fig. 9(a), 10(a) and 11(a) with combined equations for  $\mu'$ - $f$  (equations (12) + (14)) and  $\mu''$ - $f$  (equations (13) + (15)) respectively is found extremely difficult. Therefore, we have separately fitted the peaks for spin rotation component (designated as "spin") and domain wall component (designated as "dw") in the spectra: Firstly, we fit each peaks in the  $\mu''$ - $f$  curves. Secondly, we calculate the  $\mu'$ - $f$  curves by using the obtained fitting parameters and corresponding  $\mu'$  expressions (equation 12 or 14). The fitting results are shown in Fig. 9 (a), 10(a) and 11(a). The fitting parameters are listed in Table 2.  $f_s$  (exp) is the frequency at which the  $\mu''$  value of the spin rotation component reaches a maximum value on the experimental curve.  $f_{dw}$  (exp) has a similar definition for the domain wall component on the experimental curve. As shown clearly in Table 2, the damping factor  $\alpha$  of spin rotation mechanism is monotonously decreased with the increase of the MWCNT contents in the composites. For spin rotation resonance, the maximum frequency  $f_s(\max)$ , which is defined as the frequency at which  $\mu''$  reaches a maximum value, can be derived from the equation (8) as (Tsutaoka 2003):  $f_s(\max) = f_s / \sqrt{\alpha^2 + 1}$ . With the obtained fitting parameters, the calculated  $f_s(\max)$  values are also listed in Table 2. The small difference between the experimental  $f_s(\exp)$  values and the fitted  $f_s(\max)$  values indicate that our fitting method makes senses.

Sample	Domain wall motion			Spin rotation				
	$f_{dw}$	$f_{dw}(\exp)$	$\chi_{d0}$	$f_s$	$f_s(\exp)$	$f_s(\max)$	$\chi_{s0}$	$\alpha$
# 0	1.57(GHz)	1.76 (GHz)	1.49	9.42 GHz	6.80(GHz)	6.39(GHz)	1.167	1.08
# 1	1.20(GHz)	1.28 (GHz)	1.26	10.73 GHz	8.03(GHz)	8.12(GHz)	1.411	0.86
# 2	1.63(GHz)	1.98 (GHz)	1.83	11.37 GHz	11.08(GHz)	10.20(GHz)	0.881	0.49

Table 2. Fitting parameters for permeability dispersion curves, where  $f_{dw}(\exp)$  and  $f_s(\exp)$  denote the frequencies at which the  $\mu''$  values on experimental curves reach maximum for domain wall motion component and spin rotation component respectively.  $f_s(\max)$  is the calculated value, please see its definition in the text.  $f_{dw}$ ,  $f_s$ ,  $\chi_{d0}$ ,  $\chi_{s0}$  and  $\alpha$  are the fitting parameters.

Tsutaoka reported that in ferrite/resin composites, both  $f_{dw}$  and  $f_s$  increased with the decrease of the volume fraction of Ni-Zn ferrites (Tsutaoka 2003). They ascribed this to the gap parameter ( $\delta/D$ ) in ferrite composites. According to their theory, a sphere ferrite particle (or a particle cluster) with a diameter of  $D$  is supposed to be enclosed by a layer of nonmagnetic substance, such as epoxy resin. The thickness of nonmagnetic layer is  $\delta/2$ .  $f_{dw}$  and  $f_r$  can be expressed respectively as:  $f_{dw} = f_{B-dw}(1 + \chi_{B-dw} \frac{\delta}{D})^{1/2}$  and  $f_s = f_{B-s}(1 + \chi_{B-s} \frac{\delta}{D})$ , where  $\chi_{B-dw}$  ( $f_{B-dw}$ ) and  $\chi_{B-s}$  ( $f_{B-s}$ ) are the static susceptibility (resonance frequency) of bulk ferrite for domain wall motion and spin rotation mechanisms respectively. As the concentration of nonmagnetic material increases (i.e., the concentration of magnetic material decreases), the  $\delta/D$  value increases. According to Tsutaoka's theory (Tsutaoka 2003), it is clear that both  $f_{dw}$  and  $f_s$  increase with the decrease of ferrite content. However, in our case, the  $f_{dw}$  value is not monotonously increased with the decrease of the ferrite content (83 wt%, 78 wt%, and 72.9 wt% for sample 0, 1, 2 respectively). For instance,  $f_{dw}$  in sample 1 is lower than those in sample 0 and 2, see Table 2. Furthermore, we also found that with more MWCNT (15.6 wt %) added in the composite studied, the dispersion curve will be seriously distorted. In addition, the permeability behaviors of the MWCNT/NiCoZn ferrites/wax hybrid composites are different from the one reported by Tsutaoka (Tsutaoka 2003). Such an unusual behavior is believed due to the interactions between the MWCNTs and the NiCoZn ferrite particles. Interactions maybe occur between the residual nano-sized Ni particles in MWCNT and NiCoZn ferrites particles, or between the MWCNT and NiCoZn particles. Further investigation will be conducted in the future. The variation of permeability peak shapes also indirectly manifests the existence of such an interaction.

The permittivity spectra for NiCoZn/MWCNT/wax hybrid composites are shown in Fig. 9(b), 10(b) and 11(b). For sample without MWCNT (sample 0), the  $\epsilon$  values are almost constant within the measurement frequency range:  $\epsilon'$  is about 5 and  $\epsilon''$  is about 0, showing a typical feature of ferrite in GHz range. With increasing the MWCNT contents in composites (sample 1 and 2), both  $\epsilon'$  and  $\epsilon''$  increase within the frequency range, and both are gradually decreased with increasing frequency. Especially, for sample 2, there is a clear inverse proportion relationship between  $\epsilon''$  and frequency. This is a typical feature of a conducting media and shows that with more MWCNT added into the studied composite, it will become a conducting composite due to the high conductivity of MWCNT. At lower frequency range of 0.5 GHz - 6 GHz, sample 2 shows a significant dielectric loss, please see the  $\epsilon''/\epsilon'$  values in Fig. 12.

Samples	$\tau$ (s)	$\alpha$	$\epsilon_{rs}$	$\epsilon_{r\infty}$
#0	$3.89 \times 10^{-12}$	$1.39 \times 10^{-14}$	5.38	$1.18 \times 10^{-13}$
#1	$7.94 \times 10^{-11}$	$1.08 \times 10^{-16}$	57.38	18.89
#2	$6.70 \times 10^{-10}$	0.57	249.00	2.05

Table 3. Fitting results of permittivity spectra of sample 0, 1 and 2.

The permittivity dispersion spectra of these three samples have been studied based on the previous Cole-Cole dispersion law. The fitting spectra shown as the dotted lines are also presented in the above figures. The fitting results are given in Table 3. It is clear that both the static permittivity value ( $\epsilon_{rs}$ ) and the relaxation time ( $\tau$ ) increase with more MWCNTs added into the ferrites.

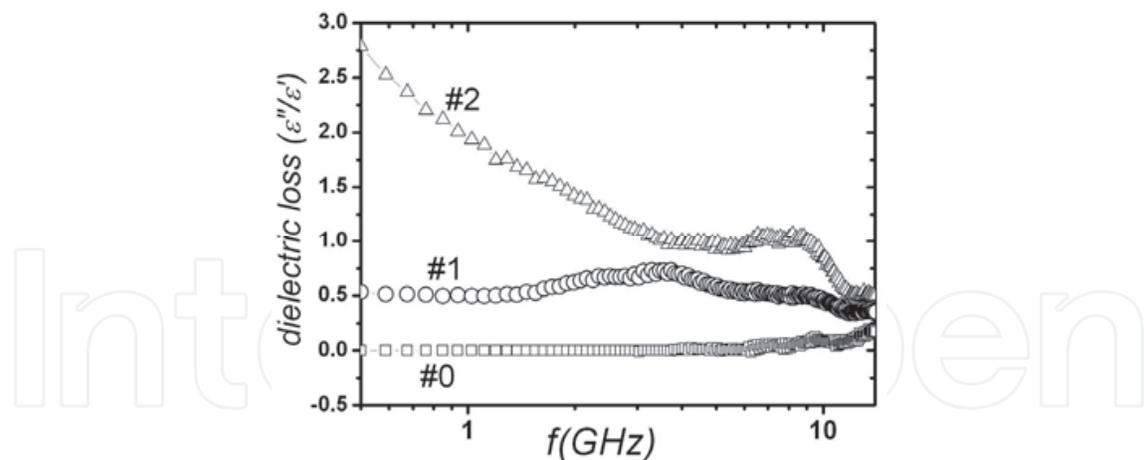


Fig. 12. Dielectric loss of ferrites doped with different amount of MWCNTs

## 7. Conclusions

In this chapter, we have studied the static properties of MWCNTs and SWCNTs. Our results show that the observed ferromagnetic properties are believed due to the residual ferromagnetic substances from the CNTs fabrication processes. Employing the Cole-Cole permittivity dispersion law, the high frequency permittivity properties have been explained for MWCNTs and SWCNTs. The observed high frequency permeability properties are also thought to be due to the ferromagnetic impurities in MWCNTs and SWCNTs. The microwave absorption properties of MWCNTs are better than those of SWCNTs in terms of reflection loss (RL) and absorption bandwidth. Our experimental results show that the high frequency properties of spinel ferrites can be effectively tuned by doping them with a small amount of MWCNTs. Especially, the permittivity of ferrites can be greatly altered by doping with MWCNTs, for instance, the dielectric loss. Both the resonance peaks of permeability spectra and values are seen to be altered by doping with MWCNTs.

## 8. Acknowledgments

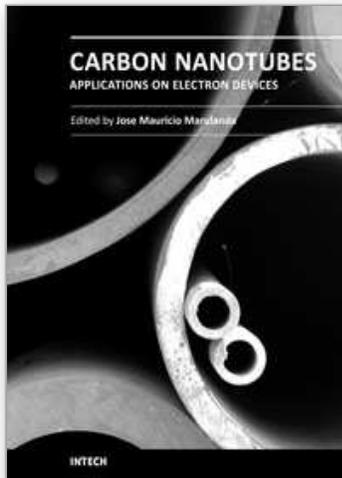
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## **Carbon Nanotubes Applications on Electron Devices**

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Carbon nanotubes (CNTs), discovered in 1991, have been a subject of intensive research for a wide range of applications. In the past decades, although carbon nanotubes have undergone massive research, considering the success of silicon, it has, nonetheless, been difficult to appreciate the potential influence of carbon nanotubes in current technology. The main objective of this book is therefore to give a wide variety of possible applications of carbon nanotubes in many industries related to electron device technology. This should allow the user to better appreciate the potential of these innovating nanometer sized materials. Readers of this book should have a good background on electron devices and semiconductor device physics as this book presents excellent results on possible device applications of carbon nanotubes. This book begins with an analysis on fabrication techniques, followed by a study on current models, and it presents a significant amount of work on different devices and applications available to current technology.

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Phone: +86-21-62489820  
Fax: +86-21-62489821

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